

Study of Electrical Properties of Dy³⁺ doped SrBi₂Nb₂O₉ Aurivillius ceramic for electronic applications

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We, **Taruna**, Roll No. **24/MSCPHY/37** and **Divyansh Ranjan Varshney**, Roll No. **24/MSCPHY/058** students of M.Sc. Physics here by declare that the project Dissertation titled "Study of Electrical Properties of Dy³⁺ doped SrBi₂Nb₂O₉ Aurivillius ceramic for electronic applications" which is submitted by us to the Department of Applied Physics, Delhi Technological University, Delhi in partial fulfilment of the requirement for the award of the degree of Master of Science, is original and not copied from any source without proper citation. This work has not previously formed the basis for the award of any Degree, Diploma Associateship, Fellowship or other similar title or recognition.

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ABSTRACT

The effect of Dy³⁺ substitution on the structural, dielectric and ferroelectric properties of SrBi_(2-x)Dy_xNb₂O₉ (SBN) (x=0.00,0.02,0.04,0.06) was systematically investigated, and the samples were synthesized using the conventional solid-state reaction method. X-ray diffraction (XRD) confirmed the formation of a single-phase orthorhombic structure, indicating successful incorporation of Dy³⁺ into the lattice. The substitution of Dy³⁺ caused systematic variations in lattice parameters and unit cell volume, suggesting localized structural distortion. Scanning electron microscopy (SEM) revealed a plate-like grain morphology with random orientation, characteristic of layered perovskite systems. FTIR spectroscopy confirms bands are present at 816 cm⁻¹, 612 cm⁻¹, and 454 cm⁻¹. Dielectric studies show well-defined ferroelectric characteristics with low dielectric loss. The dielectric constant (ϵ) was observed to be 160 for undoped SBN ceramic. The Polarization-Electric field (P-E) hysteresis measurements confirmed the ferroelectric behavior of all compositions. The remanent polarization was recorded as highest at 4% (x=0.04), and 2Pr was 7.95 $\mu\text{C}/\text{cm}^2$, whereas the coercive field 2Ec was 63.05 kV/cm. The variation in remanent polarization (Pr) and coercive field (Ec) with Dy³⁺ content indicates that dopant concentration significantly influences ferroelectric characteristics. Temperature-dependent PE loops indicate a stable ferroelectric response over a wide temperature range, 30°C to 180°C, with improved polarization. Good polarization retention with minute variations in remanent polarization (Pr) and coercive field (Ec) up to 10000 switching cycles, confirming good fatigue endurance and stability. Analysis of leakage current up to 10000 switching cycles reveals good insulating behaviour of the material with reduced defects and hence exhibits good electrical stability. Overall, Dy³⁺ substitution enhances structural stability, ferroelectric switching, and dielectric properties, making them useful in energy storage and sensing applications.

Keywords: SrBi₂Nb₂O₉ (SBN), Dy³⁺ doping, Solid-state reaction method, X-ray diffraction, Scanning electron microscopy, Ferroelectric properties, Dielectric properties.

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Abbreviations

SBN	$\text{SrBi}_2\text{Nb}_2\text{O}_9$
SBND2	$\text{SrBi}_{1.98}\text{Dy}_{0.02}\text{Nb}_2\text{O}_9$
SBND4	$\text{SrBi}_{1.96}\text{Dy}_{0.04}\text{Nb}_2\text{O}_9$
SBND6	$\text{SrBi}_{1.94}\text{Dy}_{0.06}\text{Nb}_2\text{O}_9$
SBND8	$\text{SrBi}_{1.92}\text{Dy}_{0.08}\text{Nb}_2\text{O}_9$
SBND10	$\text{SrBi}_{1.90}\text{Dy}_{0.10}\text{Nb}_2\text{O}_9$

CHAPTER-1

INTRODUCTION

1.1 Background and Historical Context

The history of Aurivillius Ceramics starts in the year 1949, when Swedish chemist Bengt Aurivillius synthesized a unique oxide and published a paper on it [1]. Before 1990, research focused on compounds such as PZT, which have low remnant polarization and high Curie temperature, but these materials are harmful to the environment. A significant breakthrough was achieved when researchers at the Symmetrix company, based in Seattle, discovered compounds such as Strontium Bismuth Niobate $\text{SrBi}_2\text{Nb}_2\text{O}_9$ (SBN) and Strontium Bismuth Tantalate $\text{SrBi}_2\text{Ta}_2\text{O}_9$ (SBT), which have high remnant polarization and less degradation over time than PZT. These materials exhibit fatigue-free behaviour under applied electric fields. This discovery paved the way for a non-volatile memory storage known as Fe-RAM.

Over the last two decades, environmental regulations have tightened, making the industrial use of compounds like PZT unsustainable, necessitating the shift to lead-free ceramics. The ability of Aurivillius ceramics to work under extreme conditions makes them suitable for applications in nuclear power plants and aerospace systems.

Today, research is focused on the spectroscopic characterization of these compounds, which helps to evaluate their structural integrity. Rare earth doping of elements has been investigated in host materials to enhance electrical and optical properties. Furthermore, A-site and B-site engineering have been explored to reduce leakage current and maximize remanent polarization to improve the quality of electrical components.

1.2 Aurivillius Ceramics

Aurivillius ceramics are an important class of ceramics that belong to Bismuth Layered Ferroelectric Structures (BLSF). They have excellent piezoelectric, ferroelectric, and dielectric properties. They have a high Curie temperature and high resistance to polarization fatigue under the applied electric field, making them ideal for high-temperature sensors and Ferroelectric Random Access Memory (Fe-RAM). The general formula for Aurivillius ceramics is $(\text{Bi}_2\text{O}_2)^{2+}(\text{A}_{m-1}\text{B}_m\text{O}_{3m+1})^{2-}$.

1.3 Ferroelectric ceramics

Ferroelectric ceramics have self-generated electric polarization which can be reversed by applying an electric field, this phenomenon is known as ferroelectricity. The perovskite structure is represented as ABO_3 , where A is the larger cation, for example, Ba^{2+} in $BaTiO_3$ and B is the smaller cation, for example, Nb^{5+} in KNN. Because of these traits, they are used in sensors, data storage, actuators and ultrasound devices

1.4 $SrBi_2Nb_2O_9$ (SBN) Ceramic

In this study, $SrBi_2Nb_2O_9$ (SBN) ceramic is selected as the host material and doped with Dy^{3+} [2]. Structural analysis of $SrBi_2Nb_2O_9$ (SBN) reveals a highly crystalline material consisting of two distinct structural blocks within the unit cell. The perovskite-like blocks comprise $(SrNb_2O_7)^{2-}$ layers, where strontium and niobium occupy the A- and B-sites of the NbO_6 octahedra, respectively. Interleaved $(Bi_2O_2)^{2+}$ sheets separate the perovskite blocks, acting as insulating layers between them [3].

SBN is a lead-free ceramic. Its eco-friendly nature makes it more suitable to use as compared to traditional materials like PZT. Due to the layered structure of SBN ceramic, the electrical and physical properties are highly directional, and there is a tiny shift that creates an electrical dipole without any external field being applied. This behaviour arises due to the spontaneous polarization inside the crystal lattice of SBN ceramic. Furthermore, doping this host material with rare earth elements or f-block elements like Er^{3+} , Dy^{3+} , and La^{3+} enhances its optical and electrical properties [4].

1.5 Rare earth doping strategy

Dy^{3+} was selected as a dopant due to its ionic radius of 0.912 Angstrom which is comparable to that of Bi^{3+} (1.03 Angstrom), facilitating the substitution of the bismuth without significant lattice distortion in the SBN ceramic. The present study aims to systematically investigate the effect of Dy^{3+} substitution on the structural, morphological, spectroscopic, ferroelectric, and dielectric properties of $SrBi_{(2-x)}Dy_xNb_2O_9$ ceramics.

Structural and electrical characterizations, including X-Ray Diffraction (XRD), Scanning Electron Microscopy (SEM), and Polarization-Electric field (PE) hysteresis loops, were performed. Frequency-dependent dielectric measurements were performed to investigate the

dielectric behaviour and defect dynamics at room temperature. Moreover, Fourier Transform Infrared Spectroscopy (FTIR) was employed to identify bond characteristics and vibrational modes of the ceramic lattice.

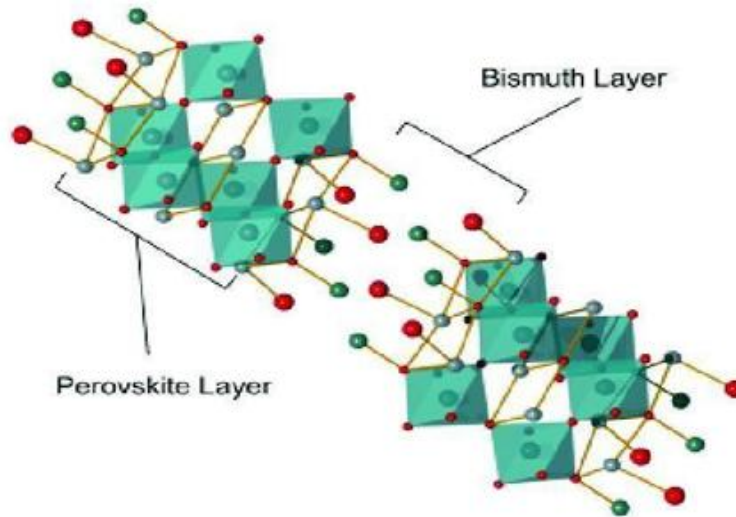


Figure 1: Bismuth Layered Structure Ferroelectric (BLSF) materials

CHAPTER -2

SYNTHESIS PROCESS

2.1 Solid State Reaction Method (SSR)

SSR method is the standard and widely used technique for the preparation of ceramic powders. The method involves weighing of all the precursors in the stoichiometric ratio, to ensure uniformity and then the precursors are well mixed or grinded for long durations with ethanol to attain uniformity. After this the powders are calcined at high temperatures to form the fine powder and polyvinyl alcohol (PVA) was added in the calcined powder and then it was converted to round pellets with the help of the hydraulic press the powder is converted into round pallets. These pallets are then sintered at a very high temperature for specific duration of time. Now these sintered pallets are being used for various characterizations, like XRD, SEM, PE, Dielectric, etc., with the help of the characterization results, the of study various electrical properties of the respective material can be done.

2.2 Synthesis of SBN with Dy³⁺

In the particular study, Dy³⁺ doped SrBi_(2-x)Dy_xNb₂O₉ (x=0.00,0.02,0.04,0.06), was prepared with process called the solid-state reaction (SSR) method. In this process, the precise weighing of the precursors of SBN, i.e., Bismuth Oxide (Bi₂O₃), Strontium Carbonate (SrCO₃), Dysprosium Oxide (Dy₂O₃), and Niobium (V) Oxide (Nb₂O₅) in the stoichiometric ratio for 15 g sample, respectively is done. Typically, the mixing is performed using a mortar and pestle for small quantities or a ball mill for larger amounts to ensure homogeneity. Here materials were then mixed with the help of an Agate mortar pestle with ethanol for 5 hours in order to get a uniform structure balance till they become a dry powder. Now this powder is calcined for 3 hours at a temperature of 950°C in a muffle furnace. 5wt% polyvinyl alcohol was added in the calcined powder and this powder is then converted to round pellets with a pelletizer at pressure around 50 kPa for 2 minutes. Then these round-shaped pallets are sintered at a temperature of 1050°C for 3 hours. Now these sintered pallets are being used for various characterizations, like XRD, SEM, PE, Dielectric, etc., with the help of the characterization results, one can study various electrical properties of the respective material.

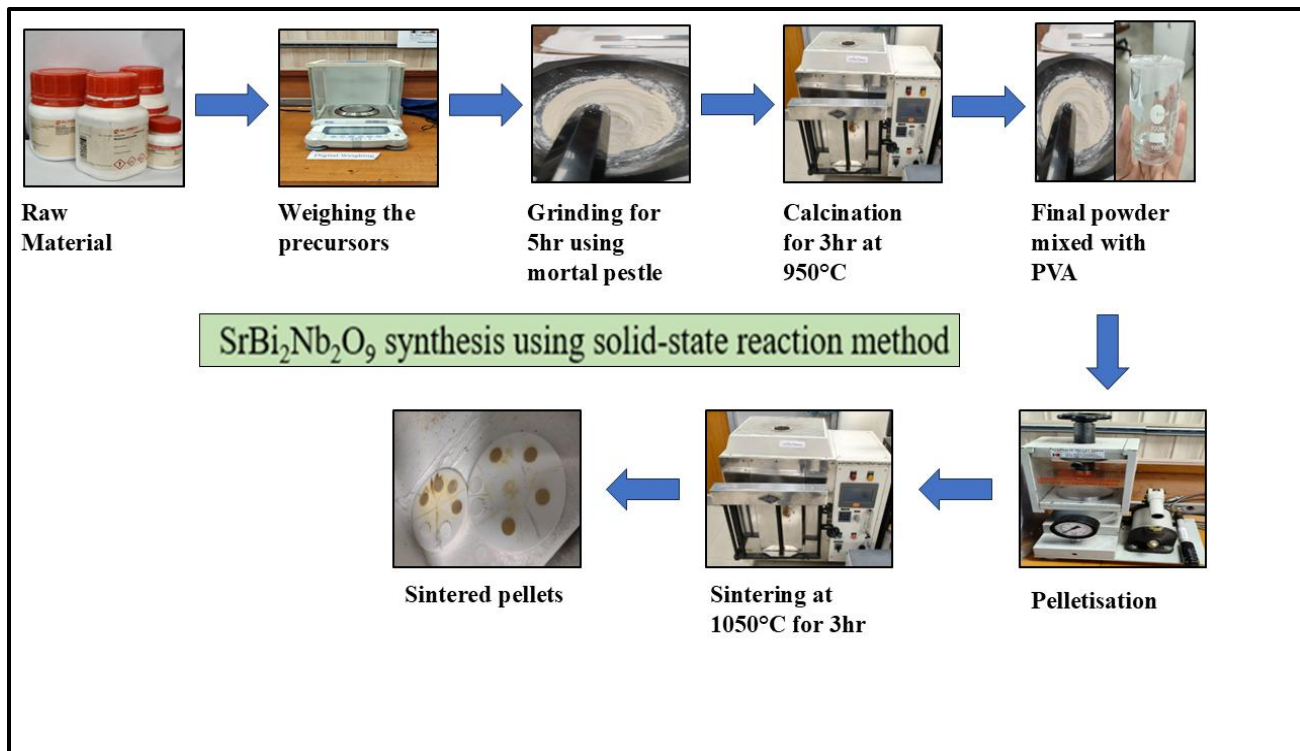


Figure 2: Schematic diagram for the synthesis of $SrBi_{(2-x)}Dy_xNb_2O_9$ ceramics using the solid-state reaction method for various dopant concentrations

CHAPTER -3

CHARACTERIZATION RESULTS AND DISCUSSIONS

3.1 Basis of X-ray diffraction (XRD) techniques

The phase identification and crystal structure analysis of Dy³⁺ doped SrBi_(2-x)Dy_xNb₂O₉ (x=0.00,0.02,0.04,0.06) ceramics were carried out using X-ray diffraction (XRD) techniques, which are essential for determining the crystalline phases present in the samples and providing insights into their structural characteristics. The XRD patterns were obtained using a Bruker D8 Discover X-Ray Diffractometer, a sophisticated instrument capable of delivering high-resolution diffraction data. The samples were prepared by grinding the sintered pellets into fine powders, which were then placed on a sample holder for analysis. The XRD measurements were conducted over a range of angles (typically 2θ) to capture the diffraction peaks associated with various crystalline phases.

The resulting XRD patterns allowed for the identification of the crystalline phases present in both pure SrBi₂Nb₂O₉ and the Dy³⁺ doped variants. By comparing the observed diffraction peaks with standard reference patterns from databases such as the Joint Committee on Powder Diffraction Standards (JCPDS), researchers could confirm the presence of specific phases and assess any changes in crystallinity due to doping.

The analysis provided valuable information regarding the lattice parameters, crystallite size, and overall structural integrity of the materials, which are crucial for understanding their ferroelectric and optical properties.

3.2 XRD Results

X-Ray Diffraction (XRD) of the Dy³⁺ doped SrBi_(2-x)Dy_xNb₂O₉ (SBN)(x = 0.02,0.04,0.06) ceramic crystal structure is shown in Figure 3. Successful incorporation of the Dy³⁺ in the crystal structure of the host SrBi_(2-x)Dy_xNb₂O₉(SBN) is confirmed.

Dy³⁺ ions were substituted at the Bi³⁺ ions site in the crystal structure of SrBi_(2-x)Dy_xNb₂O₉ (x= 0.02,0.04,0.06). The doped and undoped ceramics were then analyzed through the XRD pattern. The graph between Bragg's angle (2θ), ranging from 10°- 80°, and intensity for all the

compositions has been plotted and compared with the standard diffraction pattern of Strontium Bismuth Niobate $\text{SrBi}_2\text{Nb}_2\text{O}_9$ (SBN) that is the JCPDS data (#00-049-0607) as shown in Figure 2. X-ray diffraction confirms that the SBN formed is in a single phase with an A_{21} atom group, with no secondary peaks observed. Dy^{3+} ions doping in the crystal did not introduce defects as compared to the pure sample and successfully retained the orthorhombic structure. The most prominent peak shown in the XRD pattern is (115) for all the given compositions.

With increasing Dy^{3+} concentration, a slight shift is observed around 28° in left direction, attributed to the difference between the ionic radii of Dy^{3+} (0.912 Å) and Bi^{3+} ion (1.03 Å). This indicates lattice expansion of the crystal with unit cell volume increasing from 762.06 \AA^3 to 765.14 \AA^3 at ($x=0.04$) as shown in **Table 1**. Upon increasing the doping to ($x=0.06$) Dy^{3+} ions, lattice distortion is observed, resulting in a reduction in volume of the unit cell (761.47 \AA^3).

Regarding the distortion (b/a) as mentioned in **Table 1**, the crystal structure transitions from pseudo-tetragonal to orthorhombic structure. The density of the structures was calculated with Archimedes' principle as mentioned in **Table 1**. The structural parameters have been calculated using X-PowderX software as presented in **Table 1**. X-ray diffraction pattern results demonstrate enhancement in the properties. However, beyond optimal doping levels, distortion and degradation are observed [8,9,10,11].

Table1: - Structural parameters of $\text{SrBi}_{(2-x)}\text{Dy}_x\text{Nb}_2\text{O}_9$ (SBN)

Samples ($\text{SrBi}_{(2-x)}\text{Dy}_x\text{Nb}_2\text{O}_9$)				
Parameters	SBND0	SBND2	SBND4	SBND6
	$x=0.00$	$x=0.02$	$x=0.04$	$x=0.06$
a(Å)	5.5130	5.5272	5.5272	5.5112
b(Å)	5.5135	5.5189	5.5178	5.5101
c(Å)	25.074	25.0945	25.0970	25.0753
V(Å ³)	762.06	765.48	765.14	761.47
Distortion (b/a)	1.0009	0.9985	1.0090	0.9998
Density (g/cm ³)	12.8075	5.2111	10.7913	3.9153

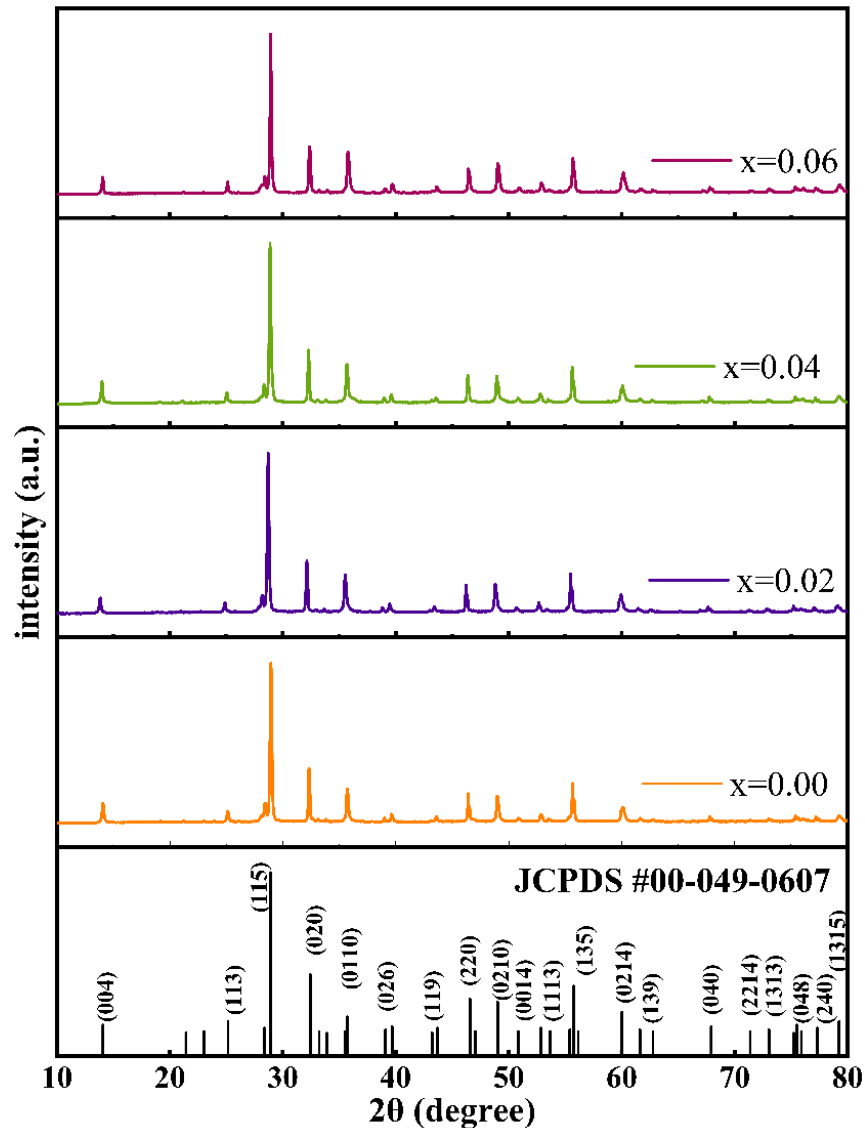


Figure 3: X-Ray Diffraction pattern of $\text{SrBi}_{(2-x)}\text{Dy}_x\text{Nb}_2\text{O}_9$ ($x=0.00, 0.02, 0.04, 0.06$)

3.3 Structural Morphology Analysis (SEM)

Structural morphology analysis is an important part of materials research because it helps to understand how the material is built at the microscopic level and how the internal structure influences its overall behaviour. By using techniques such as scanning electron microscopy (SEM), energy-dispersive X-ray spectroscopy (EDAX), and sometimes atomic force microscopy (AFM), one can observe the size, shape, and arrangement of grains or particles within a sample.

These features play a major role in deciding how a ceramic conducts electricity, responds to stress, or interacts with external fields.

In a typical ceramic material, grains are tightly packed together, and the boundaries between them can affect electrical and mechanical performance. Morphological analysis allows us to see whether the grains are uniform, well-developed, or porous. For example, larger grains often indicate better crystallization, while smaller or irregular grains may suggest incomplete reactions or the presence of defects. The presence of pores or cracks can also influence conductivity and dielectric properties by interrupting the continuity of the structure.

Doping changes grain growth behaviour, alter density, or introduce local strain, all of which can be visually confirmed through SEM images. By comparing undoped and doped samples, one can easily identify whether the modification improved the microstructure or introduced unwanted irregularities.

3.4 Results of SEM

The analysis of Scanning Electron Microscope (SEM) images reveals the surface morphology of sintered pellets of $\text{SrBi}_{(2-x)}\text{Dy}_x\text{Nb}_2\text{O}_9$ (SBN) ($x=0.00$). The SEM images are depicted below in Figure 3. $\text{SrBi}_{(2-x)}\text{Dy}_x\text{Nb}_2\text{O}_9$ (SBN) ceramic exhibits a plate-like structure with random orientation, characteristic of BLSF material. Grain size was calculated using Image J software and corresponding histograms were plotted for each concentration of Dy^{3+} , i.e., ($x=0.00, 0.02, 0.04, 0.06$) [12].

The grain size of each concentration is $1.440\ \mu\text{m}$ for no doping ($x=0.00$) of Dy^{3+} ion, $1.505\ \mu\text{m}$ for ($x=0.02$) concentration of Dy^{3+} ion, $1.547\ \mu\text{m}$ for ($x=0.04$) of Dy^{3+} ion, and $1.323\ \mu\text{m}$ for ($x=0.06$) of Dy^{3+} ion. The average grain size is $1.4535\ \mu\text{m}$. The highest value of grain size is at $x=0.04$ doping, indicating fewer defects in the crystal compared to other crystals due to its high grain size. Upon further increasing the concentration, distortion is observed due to the excessive doping, hence effective non-radiative energy transfer, resulting in a grain size of $1.323\ \mu\text{m}$. Therefore, $x=0.04$ doping or concentration of Dy^{3+} ion is identified as optimum for crystal growth and various electrical and optical properties of SBN ceramic.

SEM images confirm that the layered structures of SBN ceramic dominate the morphology of the crystal. At $x=0.04$, doping of the Dy^{3+} ion exhibits the highest structural enhancement of SBN ceramics [13,14,15].

Table 2: - Grain size (μm) for SBN ceramic.

Concentration	$x=0.00$	$x=0.02$	$x=0.04$	$x=0.06$
Grain size(μm)	1.440	1.505	1.547	1.323

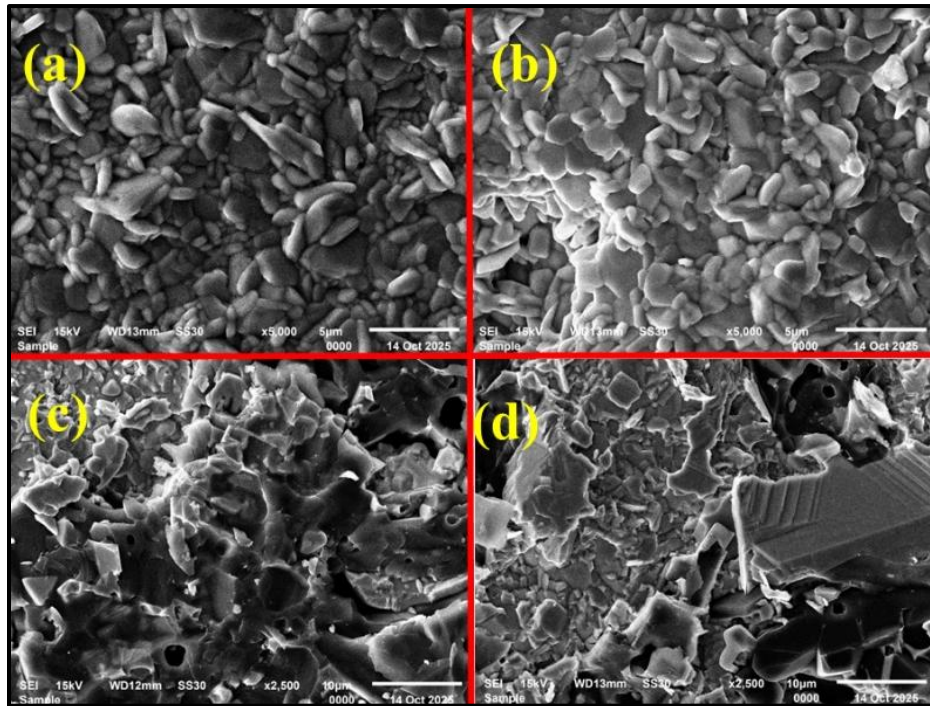


Figure 4(a-d): SEM images of $SrBi_{(2-x)}Dy_xNb_2O_9$ (SBN) where $x=0, 0.02, 0.04, 0.06$ respectively.

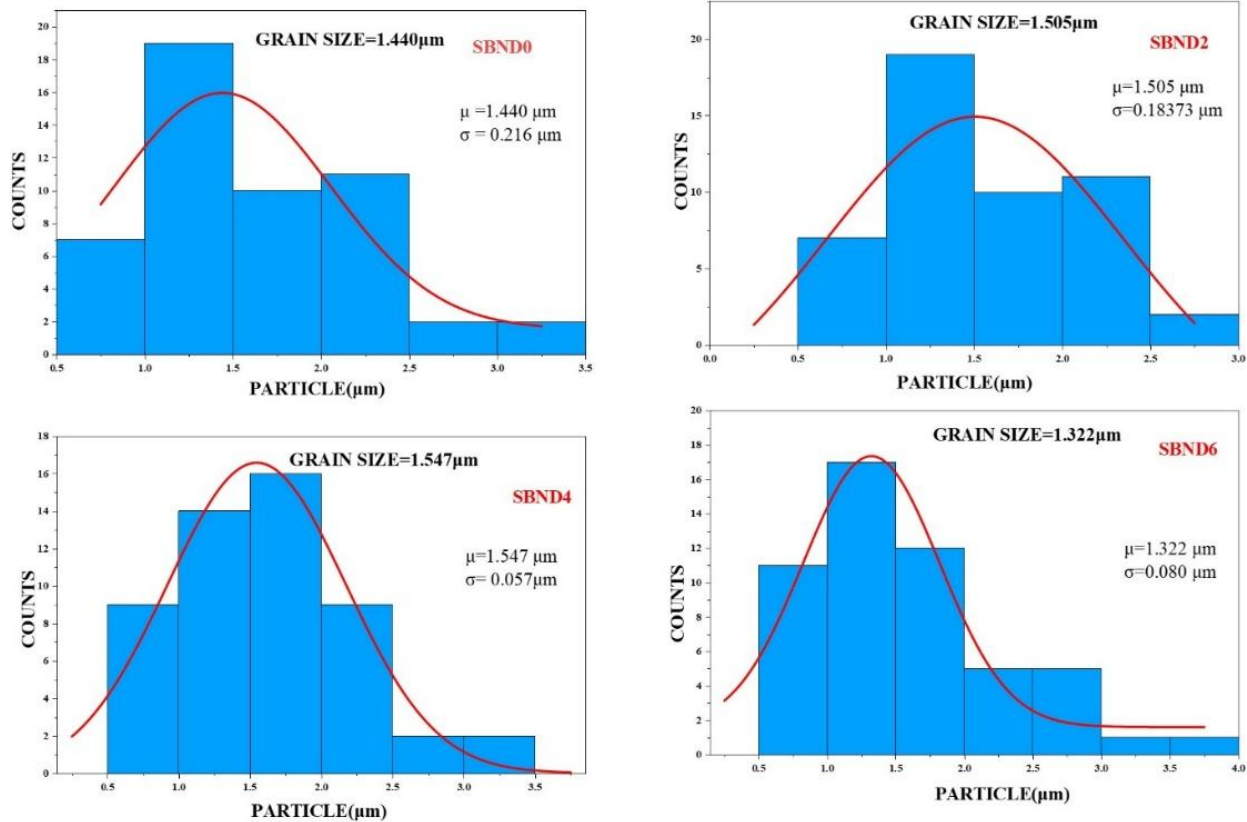


Figure 5(a-d). Histograms used for calculations of grain size from the SEM images for each Dy^{3+} doped of $SrBi_{(2-x)}Dy_xNb_2O_9$ (SBN) ceramics: (a) undoped ($x = 0.00$), (b) $x = 0.02$, (c) $x = 0.04$, and (d) $x = 0.06$,

3.5 FTIR (Fourier Transform Infra-Red spectroscopy)

FTIR spectroscopy involves recording the infrared spectrum of absorption of different types of solid, liquid, and gas materials. Various types of chemical bonds inside the structure absorb different types of frequencies. The resulting spectra are plotted on the transmission (%) versus wavenumber (cm^{-1}).

In our study of Dy^{3+} doped $SrBi_{(2-x)}Dy_xNb_2O_9$ (SBN) ($x=0.00, 0.02, 0.04, 0.06$), the following observations were recorded. The Fourier Transform InfraRed Spectrum Peak at 816 cm^{-1} represents a high frequency stretch of NbO_6 bonds, indicating that the bond present here has a shorter bond length, stronger and more rigid bonds are present in this region. Doping of Dy^{3+} ion in the Bravais lattice of $SrBi_{(2-x)}Dy_xNb_2O_9$ (SBN) ceramic confirms changes in the absorption in the FTIR spectra, with a sudden decrease in transmittance for $x=0.02$ concentration. Subsequently,

transmittance returns closer to its original position and a left shift is observed in the wavenumber(cm^{-1}) scale on the observed spectra of the doped ceramic of SBN [16].

The peak at 612 cm^{-1} represents the lower frequency region of NbO_6 , which indicates it has a longer bond length compared to the bonds present in the region of the 816 cm^{-1} . Consequently, a longer bond length is present in the region and the bond strength of the bonds available in the region is relatively lower. Regarding the effect of Dy^{3+} , transmittance follows a similar trend as observed in the 816 cm^{-1} region, with an increase observed at $x=0.04$. Upon increasing to $x=0.06$, left shift in the peak is observed [17]. The peak at 454 cm^{-1} indicates overlapping of $(\text{Bi}_2\text{O}_2)^{2+}$ structures and oxygen atoms flexing or moving relative to the central niobium atom in the Bravais lattice. Furthermore, $(\text{Bi}_2\text{O}_2)^{2+}$ is sandwiched between the perovskite layers of the $(\text{SrNb}_2\text{O}_7)^{2-}$ [18]. The absence of peaks between $2000\text{-}1000 \text{ cm}^{-1}$ confirms a high-quality ceramic or crystal has formed.

These observations confirm that NbO_6 has a distorted octahedral structure in the Bravais lattice of SBN ceramic.

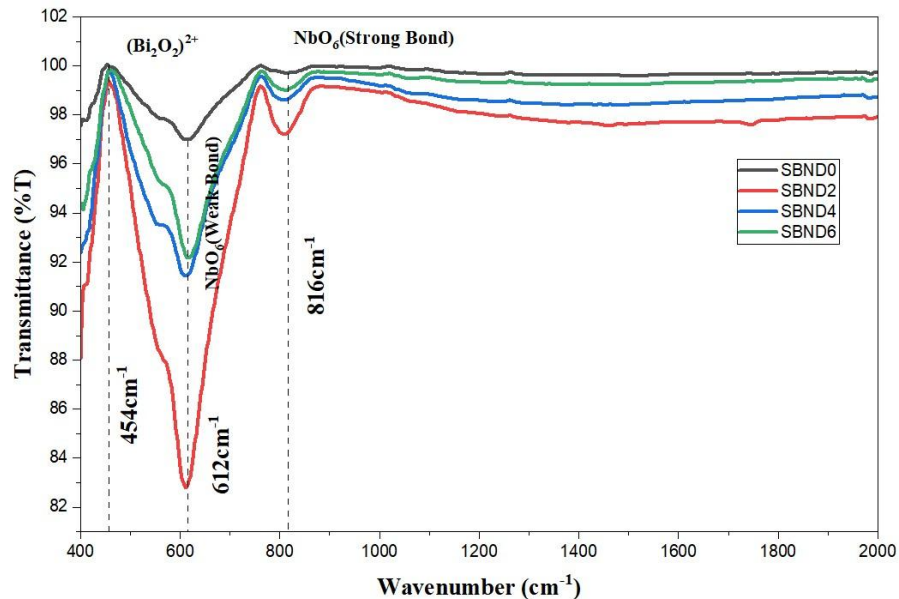


Figure 6: FTIR Spectra for the $\text{SrBi}_{(2-x)}\text{Dy}_x\text{Nb}_2\text{O}_9$ (SBN) ceramic, where $x=0.00$ (undoped), $x=0.02$, $x=0.04$, $x=0.06$

3.6 Polarization-Electric field (PE) hysteresis loop

The Polarization–Electric field (P–E) hysteresis loop is one of the most important techniques to understand the ferroelectric behaviour of a material. It demonstrates how the electric polarization of a sample responds to an externally applied electric field.

In ferroelectrics, their polarization does not immediately return to zero once the field is switched off. Hence, creates a characteristic loop shape when polarization is plotted against the electric field, known as the hysteresis loop. From this loop, several key properties of the material can be extracted, like the remnant polarization (P_r) and coercive field E_c .

The remnant polarization (P_r) tells how much the polarization remains after removing the electric field, signifying the strength and stability of the ferroelectric domains.

The coercive field E_c represents the field required to switch the polarization direction, revealing how easily the domains can be reoriented. A wider loop usually suggests stronger ferroelectric behaviour, while a slimmer loop may depict weak polarization.

3.7 PE Loop Results

Polarization-electric field (PE) hysteresis loop is a fundamental characteristic property of ferroelectric materials, which provides information about the remanent polarization (P_r) and coercive field (E_c).

The remanent polarization (P_r) is the polarization left in the ferroelectric materials after the removal of the external electric field, whereas the coercive field (E_c) measures the electric field required to reduce the polarization of a material. PE loops for the composition $\text{SrBi}_{(2-x)}\text{Dy}_x\text{Nb}_2\text{O}_9$ (SBN), where $x=0, 0.02, 0.04, 0.06$ are presented in Figure 6. Well-developed hysteresis loops confirm the ferroelectric nature of all compositions formed [19]. With increasing Dy^{3+} content, the remanent polarization ($2P_r$) increases up to an optimum level ($x=0.04$), as presented in **Table 3** leading to improved ferroelectric properties in the material. The undoped sample, SBND0 ($x=0.00$), exhibits a narrow and unsaturated loop with low remanent polarization ($2P_r=2.94\mu\text{C}/\text{cm}^2$). At $x=0.02$ (SBND2), the loop is open but not fully saturated compared to SBND0, with a slight increase in $2P_r$ value is observed ($2P_r=5.57\mu\text{C}/\text{cm}^2$). At $x=0.04$ (SBND4) results in a well-developed and saturated loop with a high $2P_r$ value ($7.95\mu\text{C}/\text{cm}^2$). This indicates that $x=0.04$ doping sample SBND4 has a strong ferroelectric response, and with the best balance

of strong polarization, this makes it suitable for designing ferroelectric devices like capacitors and sensors. At higher Dy^{3+} concentration ($x=0.06$), the loop of SBND6 becomes narrow and contracts near the origin. Defects are generated due to which the dipoles are misaligned, resulting in a reduction of the value of remanent polarization to $2P_r = 1.53\mu C/cm^2$, significantly lower compared to other compositions and further results in degradation of ferroelectric properties of the material.

The coercive field (E_c) indicates domain stability; the higher the value of E_c indicates more stable domains and harder domain switching, whereas a lower E_c indicates unstable domains, referring to domain motion and easier switching. With increasing content of Dy^{3+} , coercive field ($2E_c$) also increases as presented in Table 3 up to an optimum level ($x=0.04$). SBND0 exhibits a low $2E_c$ (36.82KV/cm), indicating easy switching of the domains with weak polarization, reflecting poor domain stability. At SBND2 ($x=0.02$), $2E_c=58.12KV/cm$, reflecting improved domain stability and enhanced ferroelectric configuration. Further, at SBND4 ($x=0.04$), $2E_c=63.05KV/cm$, indicating a balance between domain stability and switching, and thus is used for various applications. At $x=0.06$, (SBND6) $2E_c = 30.48 KV/cm$, domain switching becomes easier, contributing to internal defects, resulting in degradation of ferroelectric behaviour.

The variation in Dy^{3+} concentration considerably affects the ferroelectric behavior of the $SrBi_{(2-x)}Dy_xNb_2O_9$ (SBN) ceramic. Therefore, SBND4 achieves the most refined ferroelectric properties with a value of remanent polarization $2P_r = 7.95\mu C/cm^2$ and $2E_c = 63.05KV/cm$. Hence, SBND4 can be used for various applications, such as capacitors, sensors, where both high polarization and easy switching are important [20,21,22].

The value of coercive field ($2E_c$) for all the compositions is presented in the table and the graph for PE loops for the composition $SrBi_{(2-x)}Dy_xNb_2O_9$ (SBN) where $x=0, 0.02, 0.04, 0.06$, has been plotted.

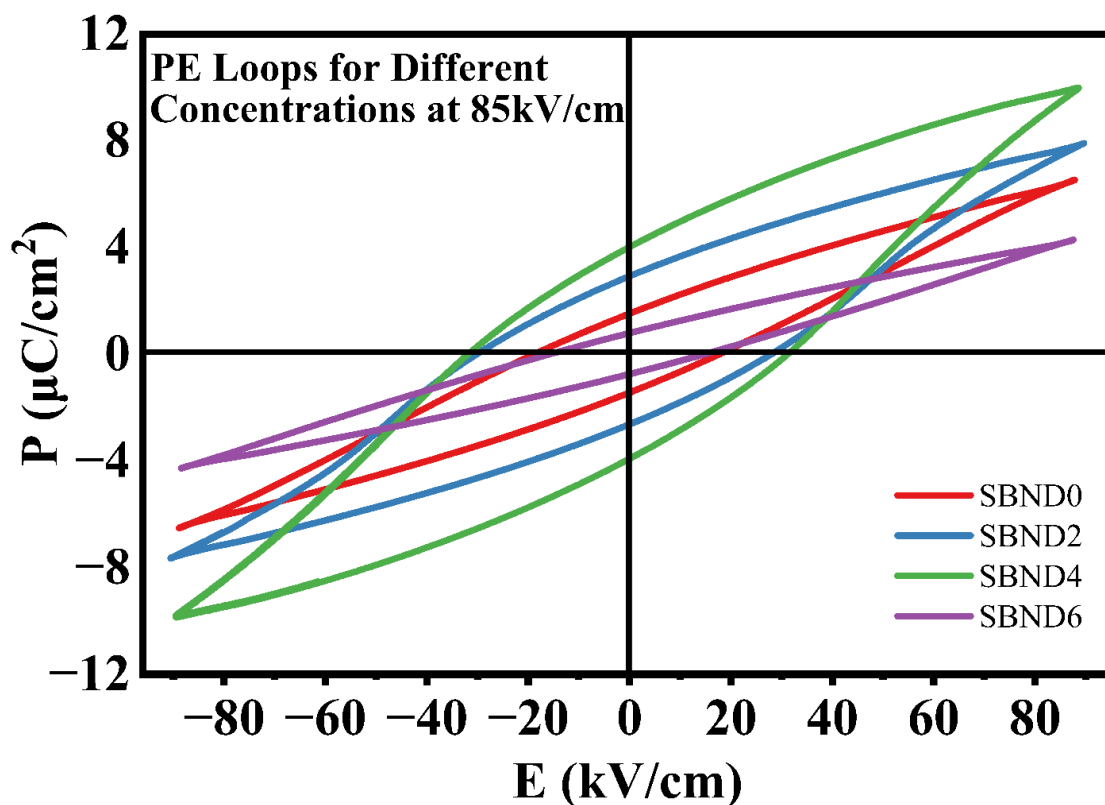


Figure 7: PE loops for the composition $SrBi_{(2-x)}Dy_xNb_2O_9$ (SBN) where $x=0,0.02,0.04,0.06$.

Table 3: Calculated values of remanent polarization ($2P_r$) and coercive field ($2E_c$)

Sno.	Polarization	SBND0	SBND2	SBND4	SBND6
1	$2P_r$ ($\mu\text{C}/\text{cm}^2$)	2.94	5.57	7.95	1.53
2	$2E_c$ (kv/cm)	36.82	58.12	63.05	30.48

3.7.1 Temperature-dependent PE Analysis

Temperature-dependent PE analysis reveals the material's macroscopic ferroelectric properties and thermal stability.

Temperature-dependent PE hysteresis loops were recorded for Dy³⁺ doped SrBi_(2-x)Dy_xNb₂O₉ compositions ($x = 0.00, 0.02, 0.04, 0.06$) over a temperature range of 30°C to 180°C. The remnant polarization (P_r), maximum polarization (P_m), Coercive field (E_c) and maximum applied field (E_{max}) were extracted from the dataset of PE hysteresis loop.

At $x=0.00$ (SBND0), the remnant polarization(P_r) is recorded as $0.368 \mu\text{C}/\text{cm}^2$, and the maximum polarization (P_m) is $3.309 \mu\text{C}/\text{cm}^2$ at 30°C. Both parameters increase steadily with rising temperature, attaining values of $4.386 \mu\text{C}/\text{cm}^2$ and $7.790 \mu\text{C}/\text{cm}^2$, respectively, at 180°C.

At $x=0.02$ (SBND2), the values of P_r and P_m exhibit a notable increase owing to enhanced dipole alignment induced by Dy³⁺ ions in the SBN ceramics. At 30°C P_r and P_m are recorded as $2.723 \mu\text{C}/\text{cm}^2$ and $7.310 \mu\text{C}/\text{cm}^2$, respectively. At 180°C, P_r and P_m further increase to $9.709 \mu\text{C}/\text{cm}^2, 14.969 \mu\text{C}/\text{cm}^2$, respectively. At $x=0.04$ (SBND4), the values of P_r and P_m further increase; there is better alignment of dipoles as compared to SBND2. The values of P_r and P_m at 30°C are $4.091 \mu\text{C}/\text{cm}^2$ and $8.091 \mu\text{C}/\text{cm}^2$, respectively, while at 180°C they reach $11.542 \mu\text{C}/\text{cm}^2$ and $15.095 \mu\text{C}/\text{cm}^2$. For $x=0.06$, the values decrease compared to SBND4, which is attributed to excessive lattice distortion induced by over-substitution of Dy³⁺ at the Bi-site. The values at 30°C are $1.312 \mu\text{C}/\text{cm}^2$ and $4.561 \mu\text{C}/\text{cm}^2$. At 180 °C, the respective values of P_r and P_m are $7.277 \mu\text{C}/\text{cm}^2$ and $11.213 \mu\text{C}/\text{cm}^2$. These results establish that SBND4 ($x = 0.04$) exhibits the most favourable ferroelectric properties and thermal stability among all investigated compositions, identifying it as the most promising candidate for non-volatile ferroelectric memory device applications.

Talking about the temperature-dependent coercive field (E_c) it tell us about the thermal agitation of the dipole inside the ceramic material. In general, it decreases with an increase in temperature.

In our study of Dy³⁺ doped SrBi_(2-x)Dy_xNb₂O₉ compositions ($x = 0.00, 0.02, 0.04, 0.06$) is recorded.

For $x = 0.00$ (SBND0), the coercive field (E_c) is $8.272 \text{ kV}/\text{cm}$ at 30°C. It increases steadily with rising temperature, attaining a value of $40.373 \text{ kV}/\text{cm}$ at 180°C.

For $x = 0.02$ (SBND2), the value of E_c exhibits a notable increase owing to enhanced dipole alignment induced by Dy^{3+} ions in the SBN ceramics. At 30°C , E_c is recorded as 30.931 kV/cm . At 180°C , E_c further increases to 40.206 kV/cm .

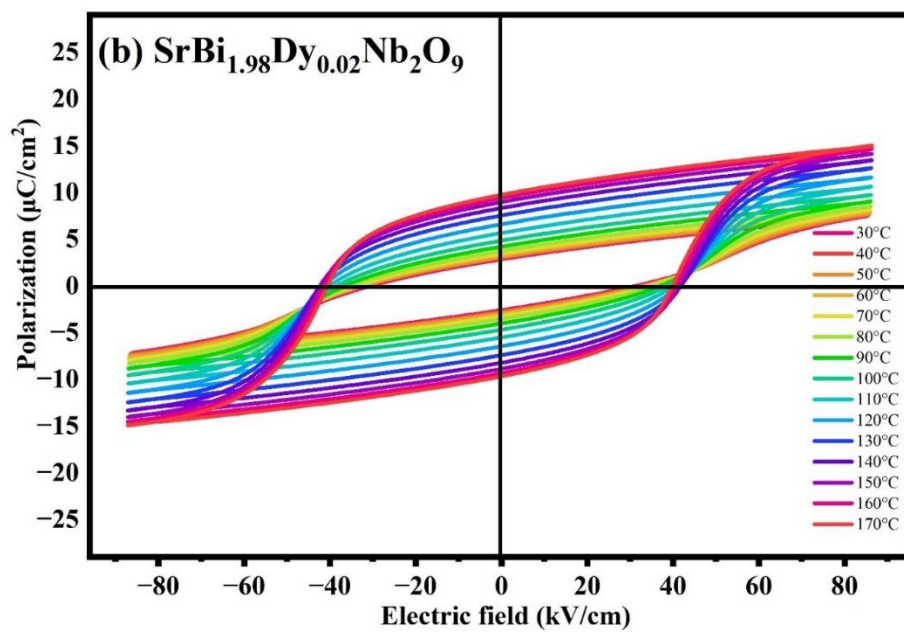
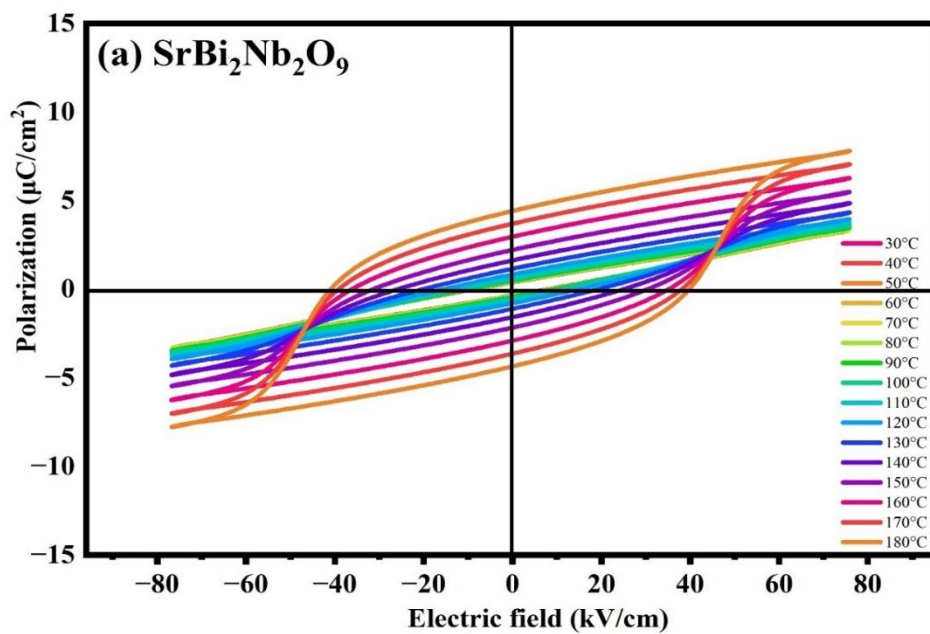
For $x = 0.04$ (SBND4), E_c decreases compared to SBND2, which is attributed to superior dipole ordering induced by Dy^{3+} substitution at the Bi-site. At 30°C , E_c is recorded as 18.161 kV/cm , while at 180°C it reaches 21.106 kV/cm .

For $x = 0.06$ (SBND6), E_c decreases compared to SBND4, which is attributed to excessive lattice distortion induced by over-substitution of Dy^{3+} at the Bi-site. At 30°C , E_c is recorded as 11.92 kV/cm . At 180°C , E_c attains a value of 21.741 kV/cm [23,24].

Based on these results, the SBND4 ceramic is the most suitable composition for non-volatile ferroelectric memory storage device. Data of the temperature run for the different compositions is given below in **Table 4**.

Table 4 :- Remanent Polarizations and coercive field values for different composition of $SrBi_{(2-x)}Dy_xNb_2O_9$ (SBN) where $x=0,0.02,0.04,0.06$ at different temperatures

Temperature(°C)	SBND0		SBND2		SBND4		SBND6	
	P_r ($\mu C/cm^2$)	E_c (kV/cm)	P_r ($\mu C/cm^2$)	E_c (kV/cm)	P_r ($\mu C/cm^2$)	E_c (kV/cm)	P_r ($\mu C/cm^2$)	E_c (kV/cm)
30	0.368	8.272	2.723	30.931	4.091	18.161	1.312	11.929
40	0.378	4.097	2.846	16.126	4.151	36.955	1.346	25.351
50	0.384	4.098	2.949	32.557	4.282	18.588	1.367	25.538
60	0.392	4.215	3.12	16.863	4.5	37.941	1.412	25.838
70	0.399	8.848	3.341	17.291	4.563	18.951	1.471	12.814
80	0.42	9.181	3.649	35.327	4.657	37.844	1.56	26.933
90	0.455	9.686	4.114	18.546	4.855	38.855	1.694	13.745
100	0.526	10.872	4.762	19.342	5.396	39.653	1.908	14.642
110	0.637	12.609	5.585	40.149	5.812	40.198	2.234	32.292
120	0.811	7.457	6.525	41.373	6.646	41.448	2.786	17.809
130	1.129	19.761	7.471	20.887	7.476	42.449	3.53	19.519
140	1.600	12.479	8.277	42.064	8.419	21.521	4.432	20.868
150	2.193	30.878	8.92	41.661	9.359	43.93	5.352	43.816
160	2.934	35.651	9.368	20.352	10.977	43.7	6.175	22.032
170	3.669	38.612	9.709	40.206	11.371	21.395	6.784	21.99
180	4.386	40.373	-----	-----	11.542	21.106	7.277	21.741



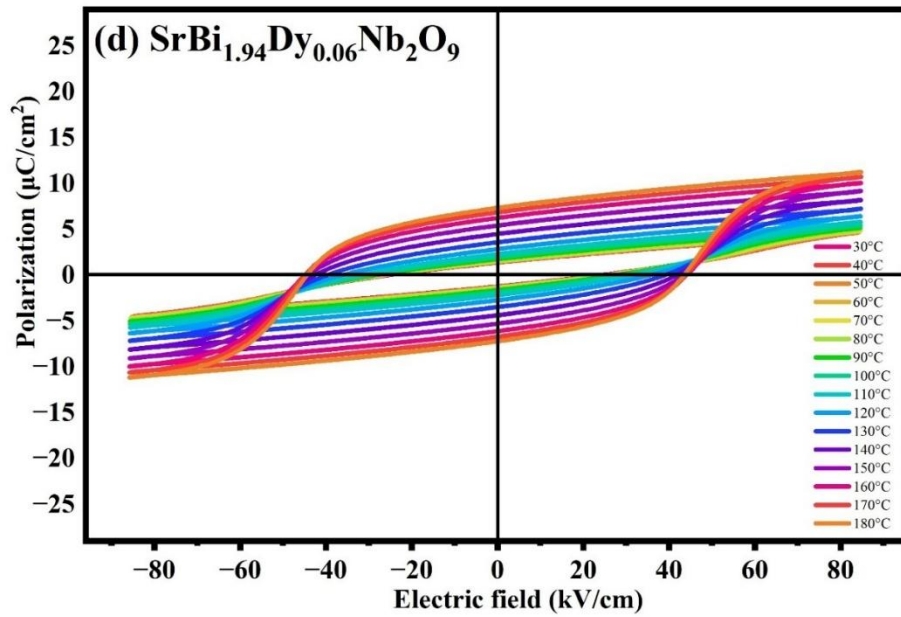
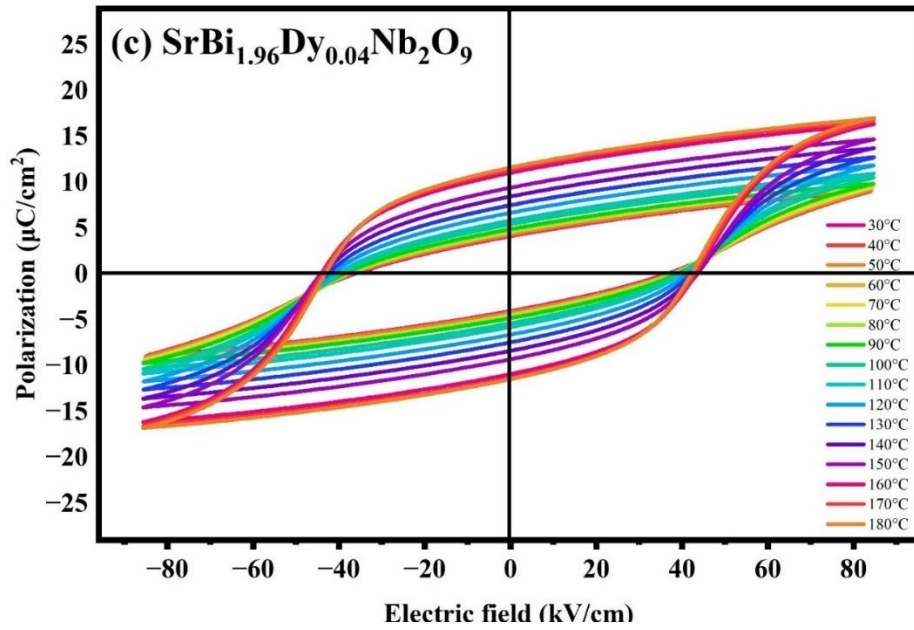


Figure 8(a-d): Plots of temperature-dependent PE loop for each Dy^{3+} doped $\text{SrBi}_{(2-x)}\text{Dy}_x\text{Nb}_2\text{O}_9$ (SBN) ceramics: (a) undoped ($x = 0.00$), (b) $x = 0.02$, (c) $x = 0.04$, and (d) $x = 0.06$.

3.7.2 Fatigue analysis of PE loop

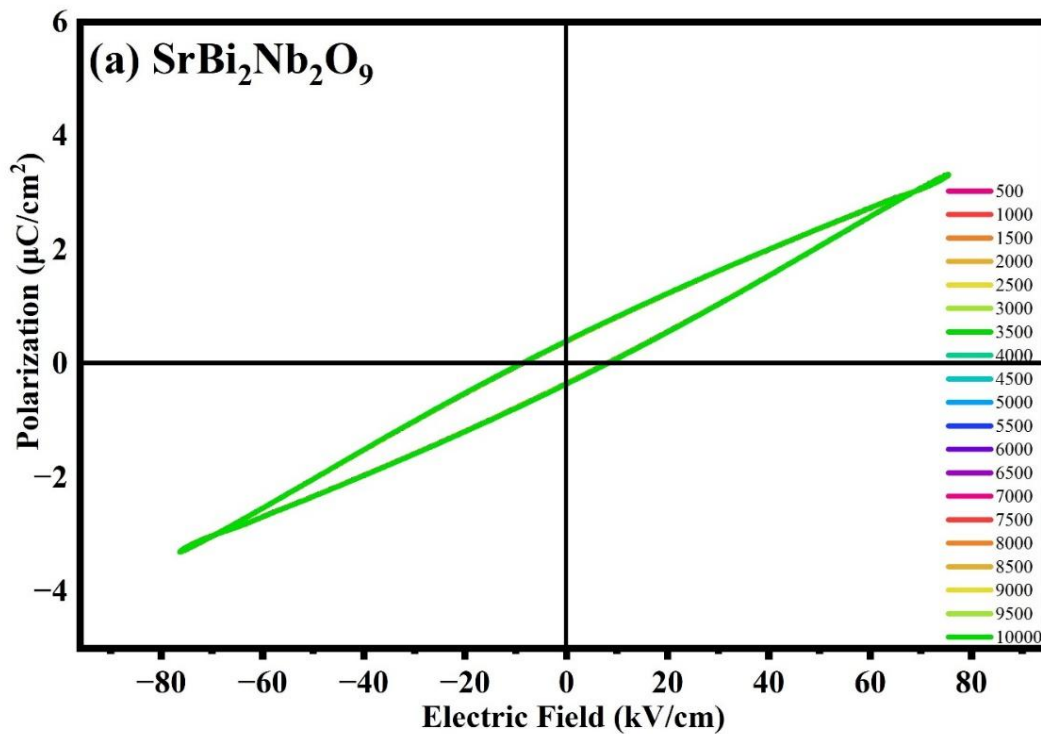
Fatigue test is a mechanism to be carried out to test the ferroelectric material to check how well it retains its remanent polarization after a million cycles of applied electric field. The gradual loss in remanent polarization is known as ferroelectric fatigue. The gradual loss in remanent polarization is known as ferroelectric fatigue. When a ferroelectric ceramic is subjected to a large number of alternating electric field cycles, the domain walls responsible for polarization switching become increasingly pinned by accumulated space charges, oxygen vacancies, and structural defects at grain boundaries and electrode interfaces. As a result, the fraction of switchable polarization volume decreases gradually with increasing cycle number, leading to a measurable reduction in P_r . This phenomenon directly limits the operational lifetime of ferroelectric-based devices such as non-volatile ferroelectric random access memory (FeRAM), actuators, and sensors. Therefore, fatigue characterization is essential to check the practical application of the ceramic before it can be used in Fe-RAM devices to avoid the breakdown before using it.

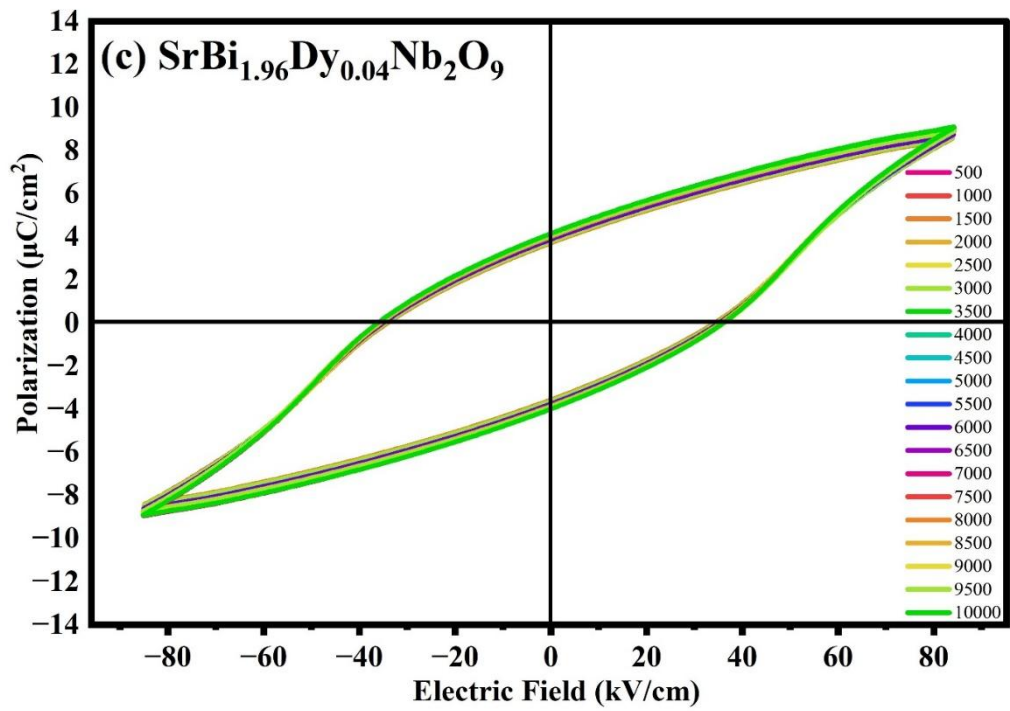
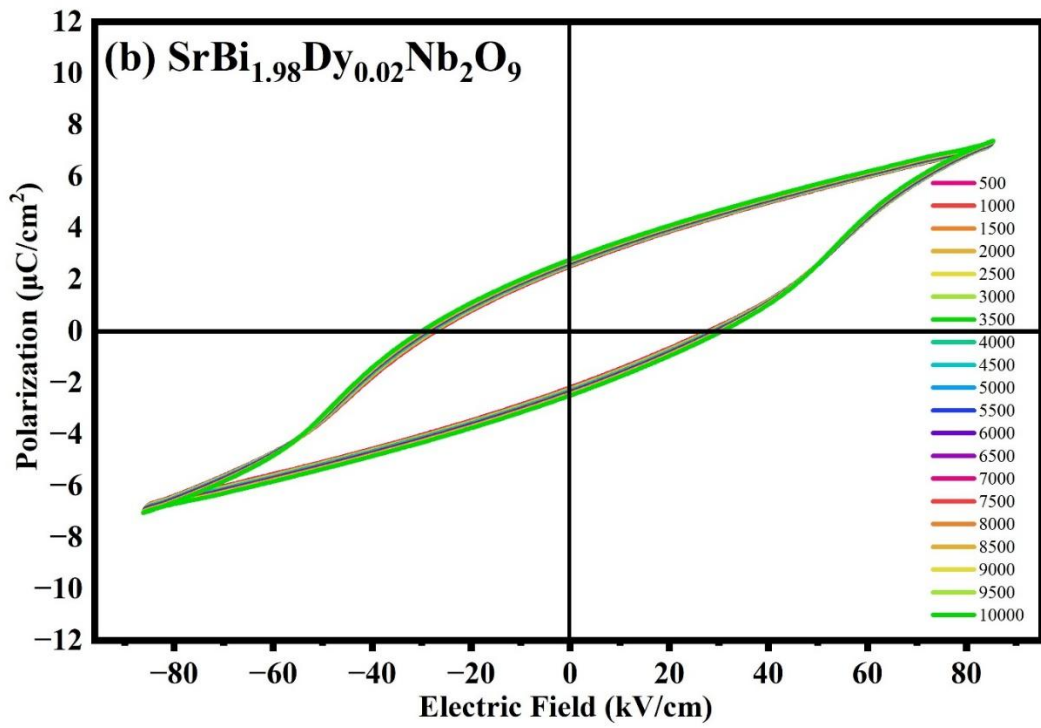
In our study of Dy^{3+} doped $SrBi_{(2-x)}Dy_xNb_2O_9$ compositions ($x = 0.00, 0.02, 0.04, 0.06$), a fatigue test was performed by applying a bipolar electric field over 10,000 cycles. The values of remanent polarization, coercive field and leakage current were recorded.

For SBND0($x=0.00$), the values of remnant polarization are relatively stable regarding all cycles in which it was performed. The range of value of P_r is $0.37 \mu C/cm^2$ to $0.377 \mu C/cm^2$. This confirms the fatigue-resistant nature of SBN ceramic. The coercive field also shows minimal variation during the scan and its range is 3.912 kV/cm to 4.059 kV/cm . Upon SBND2($x=0.02$) the value of remnant polarization shows slight variation with increasing cycles. The minor defects are shown due to the incorporation of Dy^{3+} in the SBN ceramic. The range values of remnant polarization and coercive field are $2.35\text{-}2.682 \mu C/cm^2$, $27.514\text{-}30.288 \text{ kV/cm}$. For SBND4($x=0.04$), the remnant polarization exhibits highly stable composition for all compositions throughout the entire cycling process. It confirms Dy^{3+} establishes the perfect domain wall structure, resulting in superior fatigue resistance. The coercive field also remains constant throughout the entire process. Upon increasing the concentration at $x=0.06$ (SBND6), a gradual decrease is observed during the cycles of the applied electric field. It happens due to the lattice distortion caused by the excessive

doping of Dy^{3+} . This confirms that the excessive doping of Dy at the Bi-site in the SBN ceramic also deteriorates the fatigue resistance of the ceramic.

Based on these results, SBND4 exhibits the most stable fatigue resistance behaviour among all other compositions, which makes it ideal for a non-volatile Fe-RAM memory storage device[25,26,27,28].





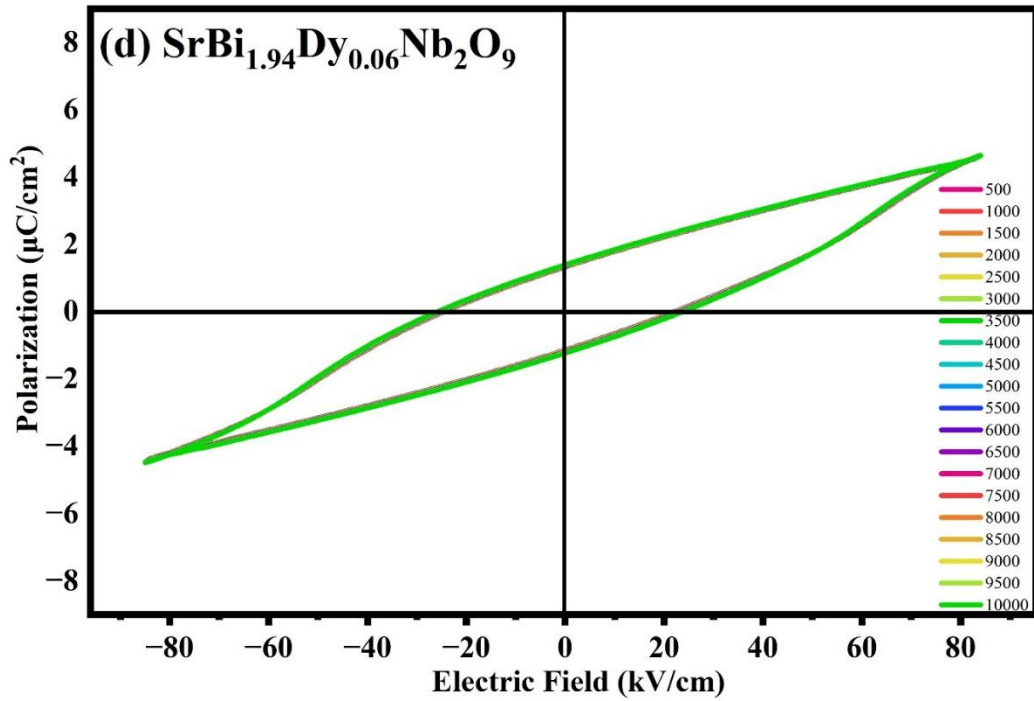


Figure 9(a-d): plots of fatigue analysis of PE loop for each Dy^{3+} doped $\text{SrBi}_{(2-x)}\text{Dy}_x\text{Nb}_2\text{O}_9$ (SBN) ceramics: (a) undoped ($x = 0.00$), (b) $x = 0.02$, (c) $x = 0.04$ and (d) $x = 0.06$.

3.7.3 Leakage current

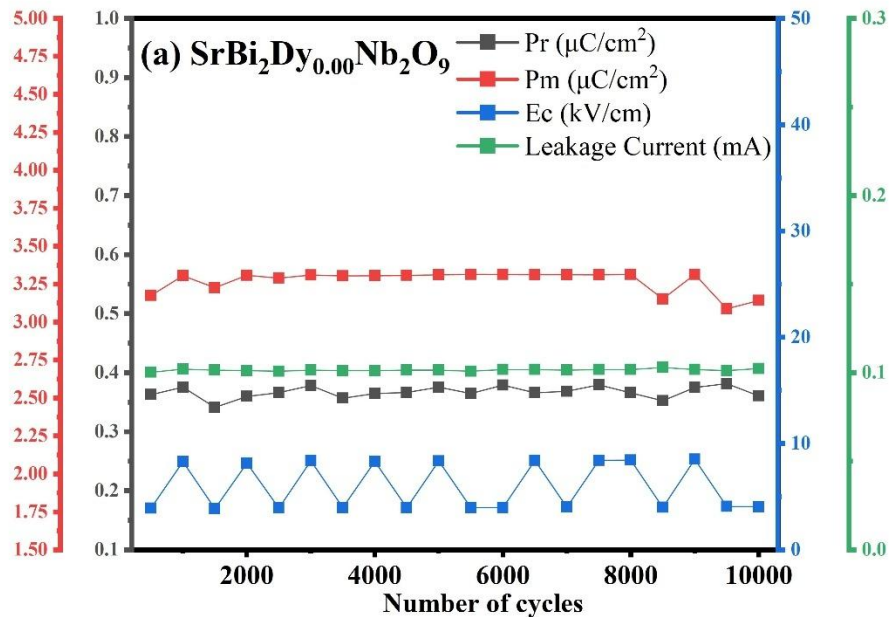
The unwanted flow of electric current inside a ferroelectric material under an applied electric field is due to the movement of charge carriers, such as oxygen vacancies, electrons, and holes, through the ceramic lattice is known as leakage current. If the leakage current is high, then it leads to a distorted PE loop for the ferroelectric materials. It indicates that poor insulating behaviour results in detorting the polarization switching of the ferroelectric materials. Therefore, low and stable leakage current is required for reliable ferroelectric device applications, like non-volatile Fe-RAM and other memory devices.

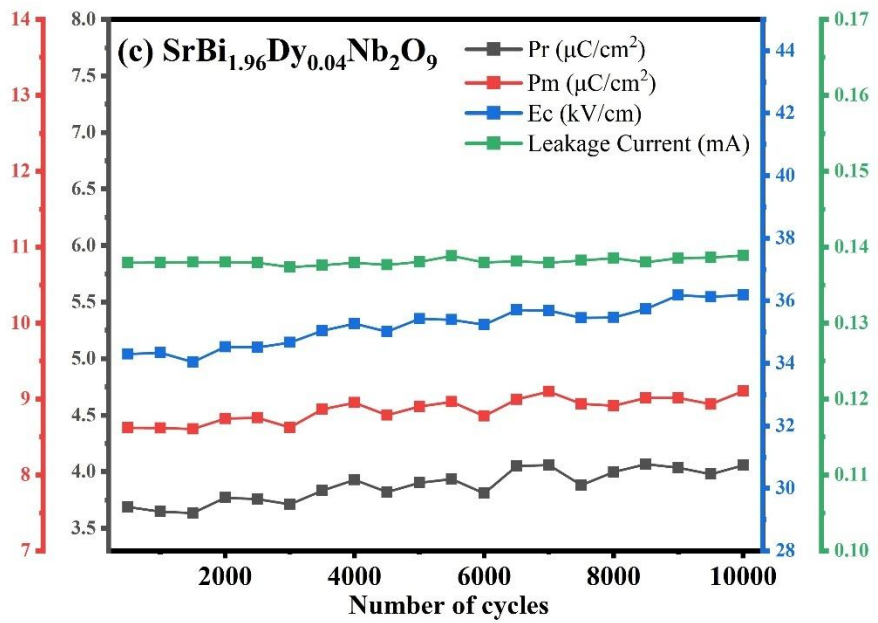
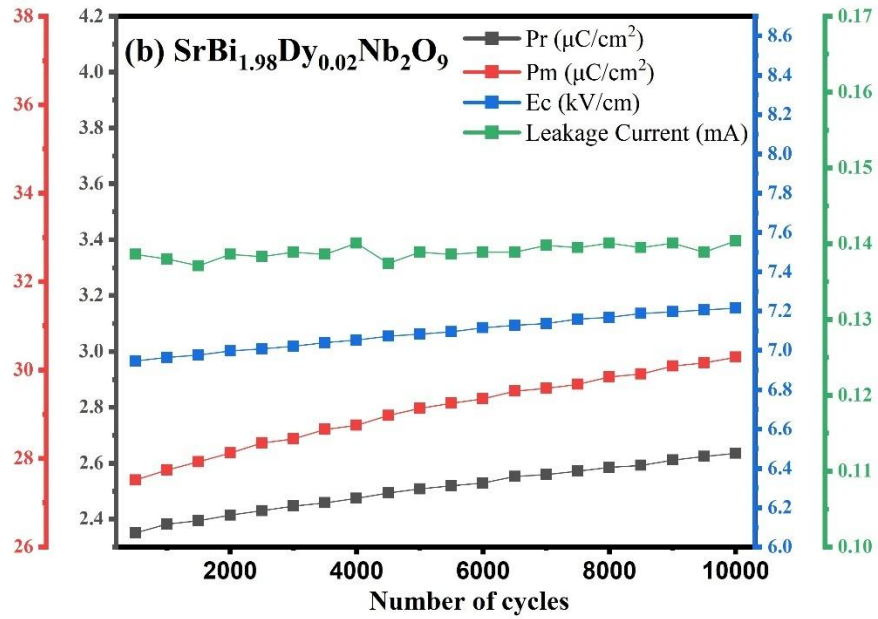
In the present study of Dy^{3+} doped $\text{SrBi}_{(2-x)}\text{Dy}_x\text{Nb}_2\text{O}_9$ compositions ($x = 0.00, 0.02, 0.04, 0.06$), the following observations were made in the domain of leakage current. At $x=0.00$ (SBND0) the value of leakage current is highly stable and lies between 0.100 mA to 0.102 mA over 10,000 cycles of applied electric field. The increase observed in leakage current is due to the accumulation of charges on the interface of electrodes and grain boundaries. For $x=0.02$ (SBND2), the range of

leakage current lies between 0.138 mA to 0.140 mA for the applied electric field over the electric field for 10,000 cycles. An increment in leakage happens due to the substitution of Dy^{3+} at the Bi-site in the lattice of $SrBi_{(2-x)}Dy_xNb_2O_9$ ceramic. Nevertheless, the variation in leakage current value is in an acceptable range, confirming the poor insulating behaviour for the compositions. Upon increasing the concentration to $x=0.04$ (SBND4), the leakage current is thoroughly uniform at 0.138 mA, which shows the ferroelectric behaviour of the sample does not deteriorate on applying bipolar cycles of electric field over 10,000 cycles. It confirms the good insulating behaviour of SBND4. This stable leakage current behaviour further supports the superior fatigue resistance of the SBND4 composition, consistent with the stable P_r observed throughout the cycling process. Further increment to $x=0.06$ (SBND6) of Dy^{3+} at the Bi-site in the lattice of $SrBi_{(2-x)}Dy_xNb_2O_9$ ceramic, the variation in the leakage current in the range of 0.104 mA to 0.105 mA over 10,000 bipolar switching cycles. This trend is observed due to the excessive doping of Dy^{3+} , which results in lattice distortion and distortion of the PE loop because of overaccumulation of the charge carriers[29,30,31].

Based on these results, SBND4 is the most suitable candidate for memory devices like Fe- RAM.

The combined plots for remnant polarization (P_r), Coercive Field (E_c), Maximum polarization (P_m) and leakage current (mA) are shown below in Figure 10





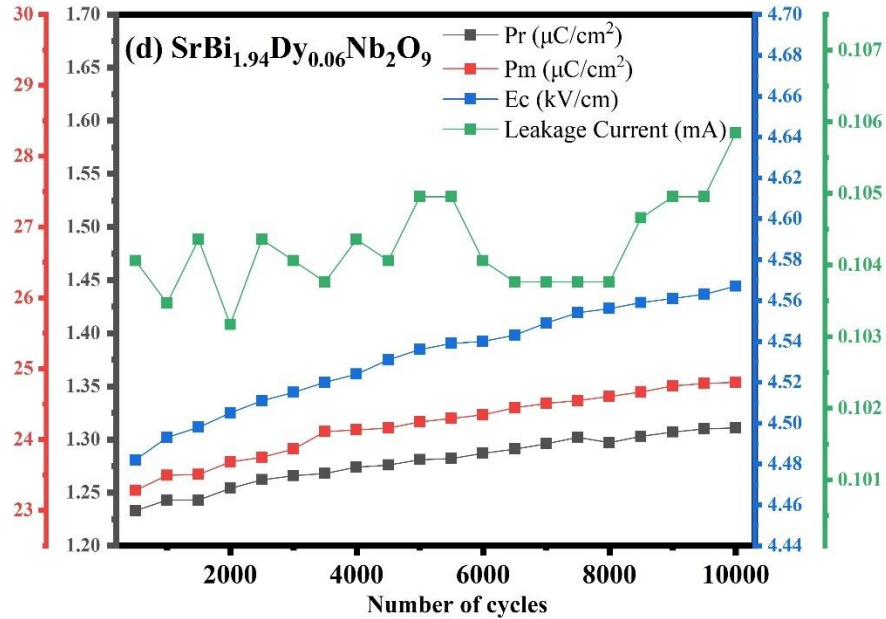


Figure 10(a-d): The combined plots for remnant polarization (P_r), Coercive Field (E_c), Maximum polarization (P_m) and leakage current (mA) for each Dy^{3+} doped of $SrBi_{(2-x)}Dy_xNb_2O_9$ (SBN) ceramics: (a) undoped ($x = 0.00$), (b) $x = 0.02$, (c) $x = 0.04$ and (d) $x = 0.06$.

3.8 Dielectric Study

Dielectric study is a prominent characterization technique for ferroelectric materials. It provides information on material inter-dipole interaction with the applied external field at different temperatures and frequencies.

In this study of Dy^{3+} doped $SrBi_{(2-x)}Dy_xNb_2O_9$ compositions ($x = 0.00, 0.02, 0.04, 0.06$), frequency-independent behaviour is observed at room temperature during the observations of SBN ceramic. The dielectric constant observed from the plot of dielectric constant (ϵ) and frequency (Hz) for pure SBN ($x=0.00$) is 159.94. The introduction of Dy^{3+} ion ($x=0.02$) shows a slight decrease in dielectric constant to 156.74. Further increasing the concentration of Dy^{3+} ion ($x=0.04$) results in an increase in the dielectric constant to 219.59. Beyond $x=0.04$, a decrease in the value of the dielectric constant to 180.56 is observed. $x=0.04$ is the optimal doping of the Dy^{3+} ion in SBN ceramics. The recorded decrease beyond $x=0.04$ is attributed to lattice strain [24], resulting in reduced ferroelectric behaviour.

A straight line in the plot of dielectric constant (ϵ) and frequency(Hz) confirms excellent dielectric response with frequency, confirming that the given SBN ceramic is a highly ferroelectric material. Among all compositions, $x=0.04$ composition is the best fit for high-frequency electronic devices and energy storage applications such as capacitors and sensors [32,33,34,35].

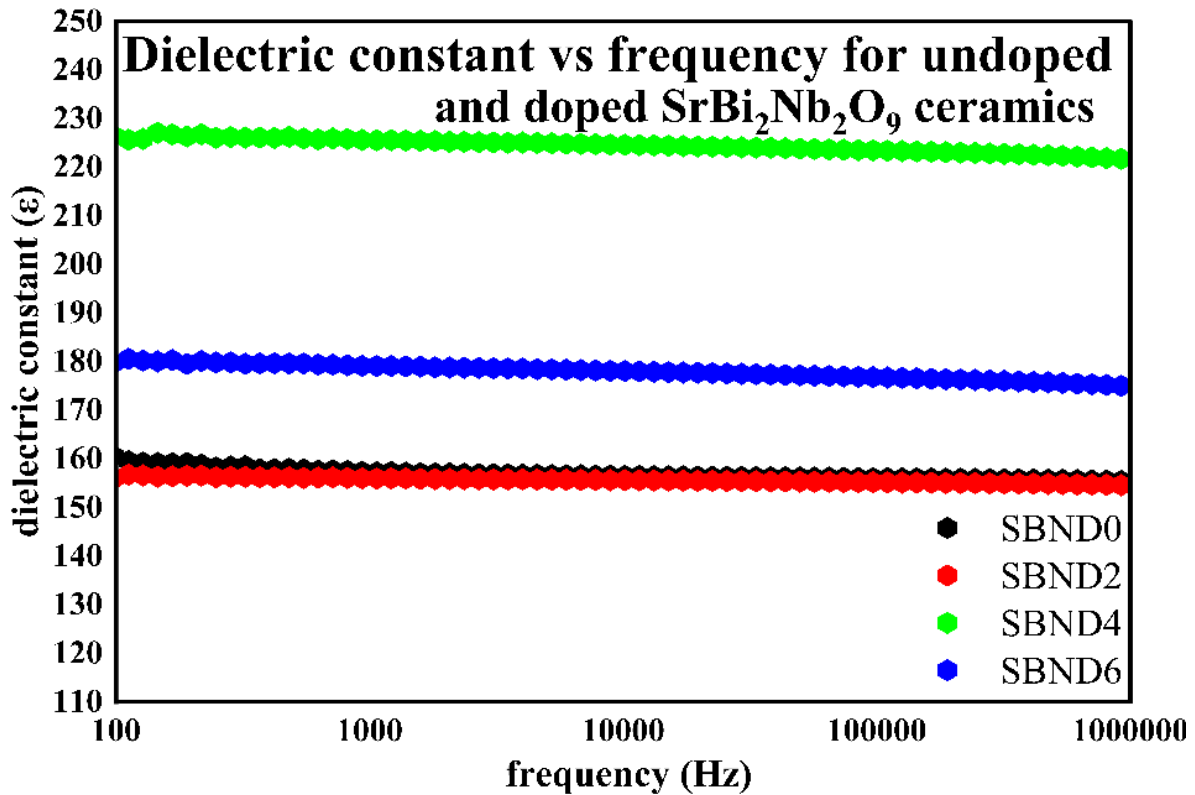


Figure 11: Frequency dependence of dielectric constant (ϵ) for different compositions of $\text{SrBi}_{(2-x)}\text{Dy}_x\text{Nb}_2\text{O}_9$ where $x=0, 0.02, 0.04, 0.06$.

3.9 Photoluminescence (PL)

Photoluminescence is an optical characterization technique in which a material absorbs photons of a specific wavelength to emit lower-energy photons of longer wavelengths. In this process, electrons absorb energy to move to their higher energy states and relax back to lower energy states, resulting in characteristic luminescence.

In this study, PL spectra of Dy³⁺ doped SrBi_(2-x)Dy_xNb₂O₉ (x=0.00, 0.02, 0.04, 0.06) ceramics were recorded under UV excitation at 387nm[36].

The peak of Dy³⁺ ions was observed in the region of 480-490 nm, and, according to previous research, it corresponds to the characteristic blue emission of Dy³⁺ ions, arising from the ⁴F_{9/2} → ⁶H_{15/2} transition due to magnetic dipole interaction in the Dy³⁺ doped SBN ceramic [37,38].

The peak was observed in the region 575-580 nm, corresponding to the ⁴F_{9/2} → ⁶H_{13/2} transition known as Hypersensitive yellow emission, attributed to the electric dipole interaction inside the SBN ceramic [39].

The peak around 660nm is expected, but it is not visible due to its lower intensity as compared to the blue and yellow emissions. The transition corresponds to the ⁴F_{9/2} → ⁶H_{11/2} of the Dy³⁺ ion. It is characteristic of the metal oxide ceramic [40]. It is known as red emission.

Based on these observations, it is concluded that the Dy³⁺ doped metal oxide ceramics are suitable candidates for the production of white LEDs[37,38,41,42].

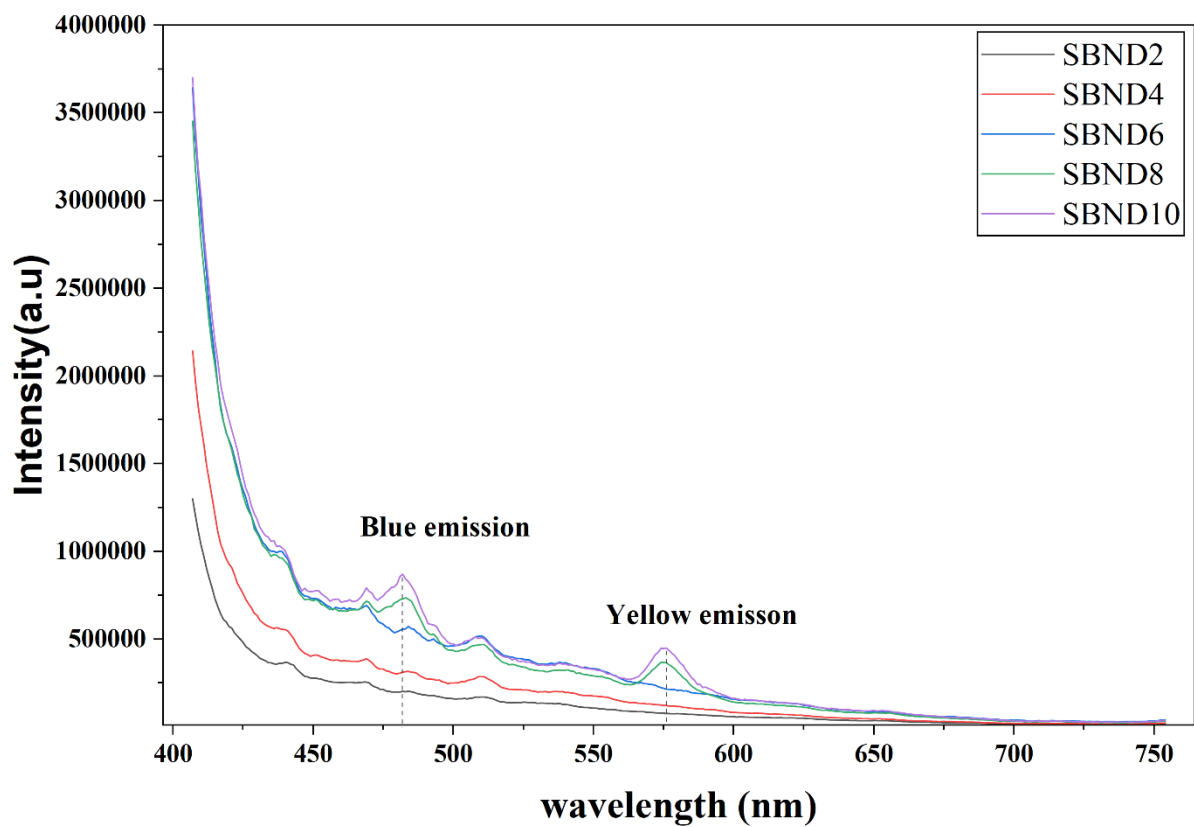


Figure 12: PL spectra for different compositions of $\text{SrBi}_{(2-x)}\text{Dy}_x\text{Nb}_2\text{O}_9$ (SBN) where $x=0,0.02,0.04,0.06,0.08,0.10$.

CHAPTER-4

CONCLUSION

In this study, the successful incorporation of Dy^{3+} ion in the $\text{SrBi}_2\text{Nb}_2\text{O}_9$ lattice with the X-ray diffraction patterns (XRD) is confirmed. The materials were synthesized using the solid-state reaction (SSR) method. The diffraction patterns indicate the single-phase structure with no extra or secondary peaks, although a slight peak shift was observed with the increasing content of Dy^{3+} ion. The shift suggests minor distortion due to the difference between the ionic radius of Dy^{3+} and the Bi^{3+} ion. SEM images revealed a plate-like grain morphology, characteristic of layered perovskite systems, with uniform distribution across all doped samples. FTIR analysis indicates a high-quality ceramic with well-defined NbO_6 vibrational modes. PE hysteresis loop at room temperature reveals strong ferroelectric behaviour. The remanent polarization (P_r) increased with Dy concentration, showing improved ferroelectric switching. The temperature-dependent PE loop reveals that as the temperature increases, dipoles within the crystal lattice undergo more favourable alignment than at room temperature, owing to thermally activated wall domain motion. The fatigue analysis of the P-E loop reveals that it can retain its memory after a huge number of cycles of bipolar applied fields. The leakage current also reveals the same. Dielectric studies further confirmed frequency-independent dielectric behaviour with low dielectric loss, demonstrating that Dy doping improves energy efficiency. The study of PL Spectra reveals its application in the production of white LEDs. Hence, Dy^{3+} doped $\text{SrBi}_{(2-x)}\text{Dy}_x\text{Nb}_2\text{O}_9$ ($x=0.02,0.04,0.06$) ceramics exhibited enhanced structural stability, ferroelectric, and dielectric properties, indicating their potential for high-frequency electronic and energy storage applications.

CHAPTER-5

APPLICATION AND FUTURE SCOPE

5.1 APPLICATIONS

Dy³⁺ doped SrBi_(2-x)Dy_xNb₂O₉ (SBN) belongs to the family of Aurivillius ferroelectric ceramics, which are widely recognized for their excellent electrical stability, low dielectric loss, high Curie temperature, and strong ferroelectric behaviour. Because of these characteristics, SBN and its modified forms hold significant potential for a wide range of advanced electronic applications.

One of the primary applications of Dy-doped SBN ceramics is in non-volatile ferroelectric random-access memory (FeRAM). The stable remnant polarization and low fatigue behaviour make these materials suitable for memory devices that require fast switching, low power consumption, and long data retention. Dy³⁺ doping enhances the structural stability and reduces defects, which helps improve the overall endurance of such memory elements. Another important area is high-frequency capacitors and dielectric components, where the dielectric constant and low dielectric loss play a crucial role. The improved dielectric performance observed in the Dy-substituted samples suggests their suitability for filters, resonators, and energy-storage capacitors in modern electronics. Additionally, Dy-modified SBN can be explored for energy-storage applications, where high breakdown strength and stable polarization behaviour are essential. The optimized composition ($x = 0.04$) shows improved ferroelectric loop characteristics that are beneficial for such devices. Overall, the enhanced electrical behavior of Dy³⁺ doped SrBi_(2-x)Dy_xNb₂O₉ (SBN) demonstrates that it is a strong candidate for next-generation electronic, electromechanical, and energy-storage applications.

5.2 FUTURE SCOPE

The study on Dy³⁺ doped SrBi_(2-x)Dy_xNb₂O₉ ($x=0.00, 0.02, 0.04, 0.06$) provides several promising directions for future research, particularly because Aurivillius ceramics continue to attract attention for advanced electronic and multifunctional applications. One potential area lies in optimizing dopant concentration beyond the limited range studied here. Since the 4% Dy³⁺ composition displayed the most favourable ferroelectric and morphological characteristics, it would be valuable to investigate finer doping steps around this concentration to identify the exact optimum level. Future studies can also focus on understanding the conduction mechanisms in these

materials using impedance spectroscopy, and temperature-dependent dielectric studies. These techniques can give clearer insights into how Dy^{3+} influences charge transport, domain wall motion, and relaxation processes within the SBN lattice. Additionally, the incorporation of co-dopants such as rare-earth ions (La^{3+} , Nd^{3+} , Gd^{3+}) may further tailor grain growth behaviour, reduce defects, and enhance energy-storage capability. Overall, the present study lays a strong foundation, and by extending it across new compositions, techniques, and device geometries, Dy-doped SBN ceramics hold immense potential for next-generation electronic and energy-storage technologies.

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CHAPTER-6
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APPENDIX

Conference Record

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Study of Electrical Properties of Dy³⁺-doped SrBi_{2-x}Dy_xNb₂O₉ aurivillius ceramic for advanced electronic applications

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ABSTRACT

A series of ferroelectric ceramics with the composition SrBi_{2-x}Dy_xNb₂O₉ (SBN) were synthesized using the conventional solid state reaction method. X-ray diffraction (XRD) confirmed the formation of a single-phase orthorhombic structure, indicating successful incorporation of Dy³⁺ ions into the lattice. The substitution of Dy³⁺ caused systematic variations in lattice parameters and unit cell volume, suggesting localized structural distortion. Scanning electron microscopy (SEM) revealed a plate-like grain morphology with random orientation, characteristic of layered perovskite systems. Dielectric studies shows well-defined ferroelectric characteristics with low dielectric loss, while PE hysteresis measurements confirmed the ferroelectric behaviour of all compositions. The variation in remanent polarization (P_r) and coercive field (E_c) with Dy content indicates that dopant concentration significantly influences ferroelectric switching characteristics. Overall, Dy³⁺ substitution in SBN ceramics enhances structural stability and dielectric performance, making these materials promising candidates for high-frequency electronic devices and energy storage applications.

Introduction

- This study focuses on developing scientific understanding of single-phase Aurivillius-type SrBi₂Nb₂O₉ (SBN) ceramics doped with Dy³⁺ ions.
- It belongs to the Bismuth Layered Structure Ferroelectrics (BLSF) family, known for its unique layered perovskite-like structure, which can be improved by doping with rare-earth elements like Dy³⁺, Nd³⁺, and Er³⁺.
- In this work, Dy³⁺-doped SrBi_{2-x}Dy_xNb₂O₉ (x = 0.02, 0.04, 0.06) ceramics were synthesized and analyzed to examine how Dy³⁺ substitution affects their structural and electrical properties.

Results and Discussions

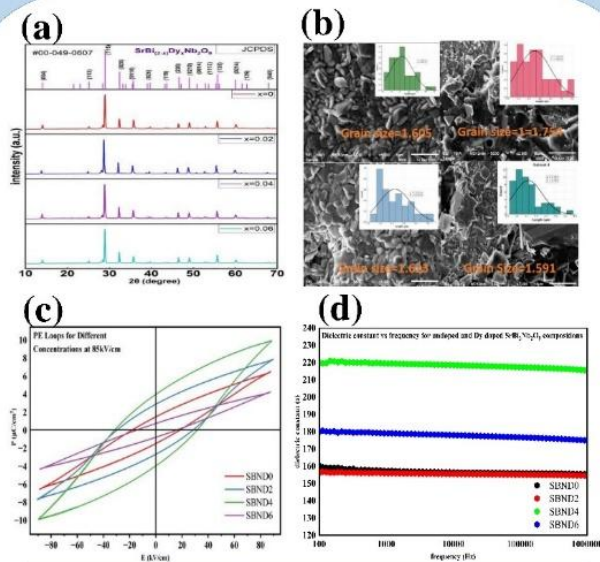
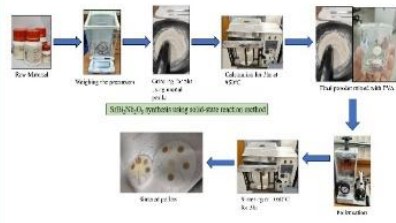


Fig.-(a) XRD spectra (b) SEM images (c) PE Loops (d) Dielectric spectra of undoped and Dy doped SBN.

Methodology



Conclusion

- The formation of a single-phase Aurivillius-type orthorhombic structure and the successful incorporation of Dy ions into the SBN lattice is confirmed from the XRD Spectra.
- The average grain size is calculated from the SEM microstructures of all the undoped and doped SBN compositions.
- The samples exhibited significant remanent polarisation (Pr) values for increasing Dy³⁺ concentrations.
- The frequency-independent Dielectric spectra confirm the ferroelectric characteristics of the prepared SBN compositions.

References

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- [2] Banwal, R. Bokolia, "Effect of Er³⁺ Ion Doping on Structural, Ferroelectric, and Up/down Conversion Luminescence in BaBi₂Nb₂O₉ Ceramic, vol. 47, 2021, pp. 4692-4695, 03.

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Participation Certificate

The certificate features logos for JECRC University (with the tagline 'BUILD YOUR WORLD'), JMCH (JECRC Medical College and Hospital), SMRS (Soft Materials Research Society, Jaipur, India), and IAPT (International Association of Physics Teachers). It also includes a '25 Years of Building Talents' anniversary logo. The main title is 'International Conference on Innovations, Advances in Material Science for Sustainable Goals (IAMSSG 2025)' held from October 29-31, 2025. The event is organized by the Department of Physics at JECRC University, Jaipur, Rajasthan, and sponsored by ANRF and DST - Rajasthan. The certificate is for a paper/poster presentation titled 'Electrical and Optical study of Bi^{3+} doped-- advanced electronic applications' by an author whose name is redacted. The presentation was made by 'Tannu'. The certificate is signed by Prof. Pranav Saxena (Convener), Prof. Widhi Dubey (General Chair), and Prof. Victor Gambhir (President, JECRC University).

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