

**INVESTIGATION OF STRUCTURAL AND OPTICAL
PROPERTIES OF PURE SnO_2 , ZNO AND SnO_2/ZNO
COMPOSITE NANORODS**

A Dissertation

SUBMITTED IN PARTIAL FULFILMENT OF THE REQUIREMENT

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Submitted by

Vineeta yadav 2K20/MSCPHY/36

Nirmal 2K20/MSCPHY/19

Under the supervision of

Dr. Deshraj Meena



Department of Applied Physics

Delhi Technological University

Bawana Road, Delhi -110042

MAY 2022

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VINEETA YADAV, NIRMAL

2K20/MSCPHY/36, 2K20/MSCPHY/19

M.Sc., Applied Physics

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Dr. DESHRAJ MEENA

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List of symbols and abbreviations

S.no.	List of Symbols	
	Symbol	Full Form
1.	SnO ₂	Tin Oxide
2.	WH Plot	Willianson Hall Plot
3.	SnCl ₄ .5H ₂ O	Tin Tetrachloride
4.	ZnCl ₂	Zinc Chloride
5.	ZnSO ₄ .7H ₂ O	Zinc Sulphate Heptahydrate
6.	ZnCl ₂	Zinc Chloride
7.	KOH	Potassium Hydroxide
8.	Na ₂ CO ₃	Sodium Carbonate

Abbreviations:

S.no.	Abbreviations
1.	3D = Three dimensional
2.	NWS=Nanowires
3.	NRS=Nanorods
4.	XRD=X Ray Diffraction

5.	ZnCl ₂ =Zinc Chloride
6.	DI=De-ionized water
7.	SMO=Semiconductor Metal Oxide

Abstract

Nanorods of SnO₂, ZnO and SnO₂/ZnO composites were successfully synthesised by the hydrothermal method in a one step process under specific conditions. The lattice strain and crystallite size of the nanorods were calculated using Williamson-Hall and Scherrer methods. TEM (Transmission Electron Microscopy) results confirm the synthesized samples are nanorods. The various crystalline planes of the nanorods were also identified using SAED (Selected Area Electron Diffraction) Pattern. The EDS (energy dispersive spectroscopy) result confirms the chemical purity of the nanorods. Furthermore, the optical band gap of SnO₂/ZnO nanocomposite was found to be lesser than the pure optical bandgaps of both SnO₂ and ZnO nanorods. Various functional groups were found to be on the surface of the synthesized nanorods, from the FTIR analysis.

CHAPTER 1

1. Introduction

In recent years, many applications, including [1] gas sensors [2], solar cells [3], lithium-ion batteries (LIBs)[3], and other opto-electronic nanodevices [6–9] are studied for metal oxide nano-wires, nano-tubes and nanofibers (Semi-conductor nanostructures), due to their appreciable electronic structural characteristics, optical properties and various chemical properties. The metal oxides that are formed between metal cations and oxide ions are the chemical compound. Moreover, due to their band gaps being wide and exciton binding energy being high, ZnO, TiO₂, SnO₂, CuO, Fe₂O₃ etc, are earlier reported to be some of best holders of semiconducting properties in metal oxides due to their wide band gap and high exciton binding energy. In ii and iv group of semiconductors Zn and Sn have attracted a lot of researchers due to their unique properties like band gap of 3.37 eV and also large exciton binding energy (60 Mev). Tin dioxide (SnO₂) and Zinc oxide (ZnO) have been widely used as photocatalyst for photodegradation of organic compounds, sensing material in resistive gas sensors, perovskite solar cells, liquid crystal display, biosensors, acoustic photodevices and photodiodes, electrochemical supercapacitors, catalysts and in the development of electrode materials for LIBs [3]. Both of these metal oxides have high melting point and large cohesive energy, excellent thermal and mechanical stability, good electrical conductivity, and high transparency in the visible spectrum. These properties make them obvious for the above application. Moreover, both are easily available and environment friendly. It has been observed that to obtain the desired materials with enhanced properties, the nanostructures of SnO₂ and ZnO composite has been used in many applications [4-5 both are easily available and environment friendly. The efficiency for the devices based on pure metals are lower than the devices based on SnO₂/ZnO nanocomposites as reported in previous literature [6]. Composites are advantageous as they have been found more porous in comparison with the pure metal oxides. Especially, SnO₂ can be made porous with the addition of ZnO in very small amounts [6].

In the present work, a comparative analysis of the structural and optical properties of SnO₂/ZnO nanocomposites with pure SnO₂ and ZnO has been done. The nanorods of the pure SnO₂, ZnO and SnO₂/ZnO composites were synthesized using hydrothermal method. The results reveal that the composites have lower optical band gap in comparison with pure SnO₂ and ZnO. The

efficiency of charge separation is greatly improved by threshold energy difference of SnO₂ and ZnO, thus improving the optical properties.

CHAPTER 2

2. Literature review

2.1 Paper 1

One-step synthesis of scions SnO₂/ZnO heterostructures and their improved gas sensing properties

This paper studied about branch heterostructure composite of the nanorods of zinc oxide which were backbones and the tin oxide were as a branch was prepared using the facile one-step hydrothermal method and the morphology and structure of the components of these composites were characterized by the SEM and x-ray powder diffraction and elemental mapping analysis the evolution process of the composites were observed by the SEM and it was studied the composites with the novel hydro structures were fabricated for a good sensor and other gases in properties were tested for various gases like ethanol methanol and acetone ammonia two words which give which gave different responses from 18 to 100 PPM.

Results

In the paper branched SnO₂ heterostructures were composed of ZnO on which SnO₂ was loaded by hydrothermal method. The structural and morphological properties were studied for this novel structure. Gas sensors were fabricated using this sample and gas sensing parameters were studied. It was found that the composites exhibited the better gas sensing properties. The reason for the better performance may be because of the formation of the N-N heterostructure which increased the resistance of the sensor compared pure zinc oxide sample.

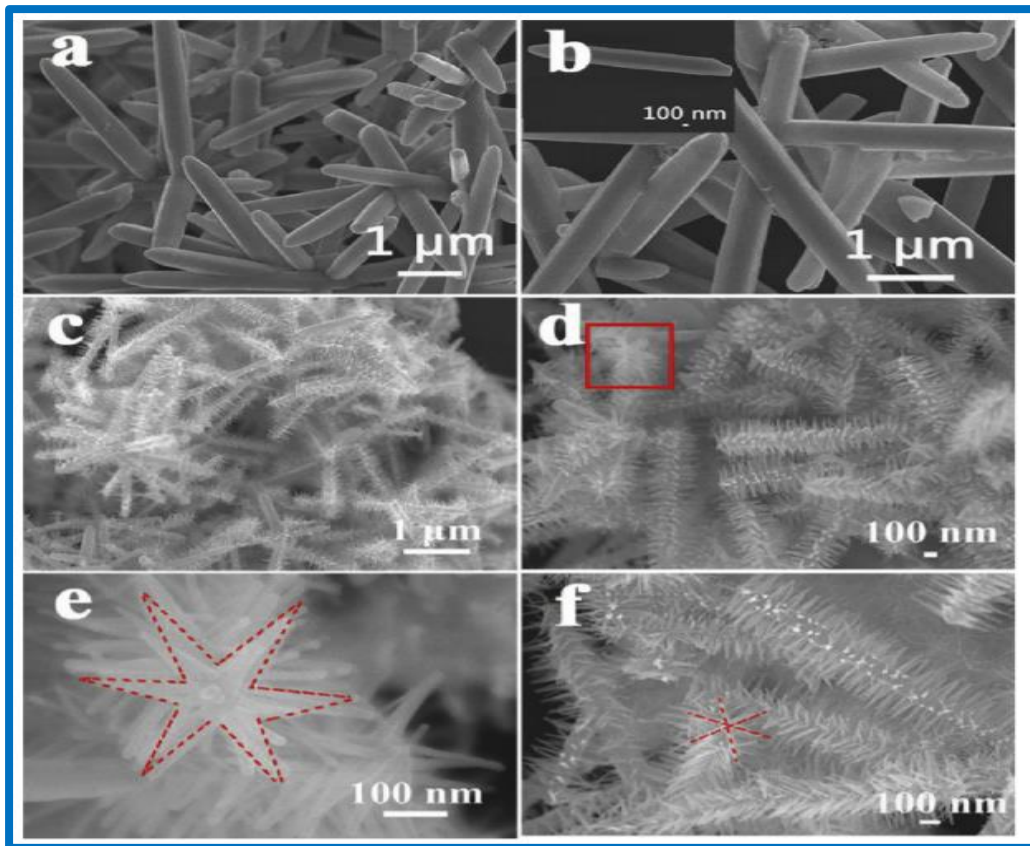


Figure 1 SEM images of the SnO₂/ZnO heterostructure.

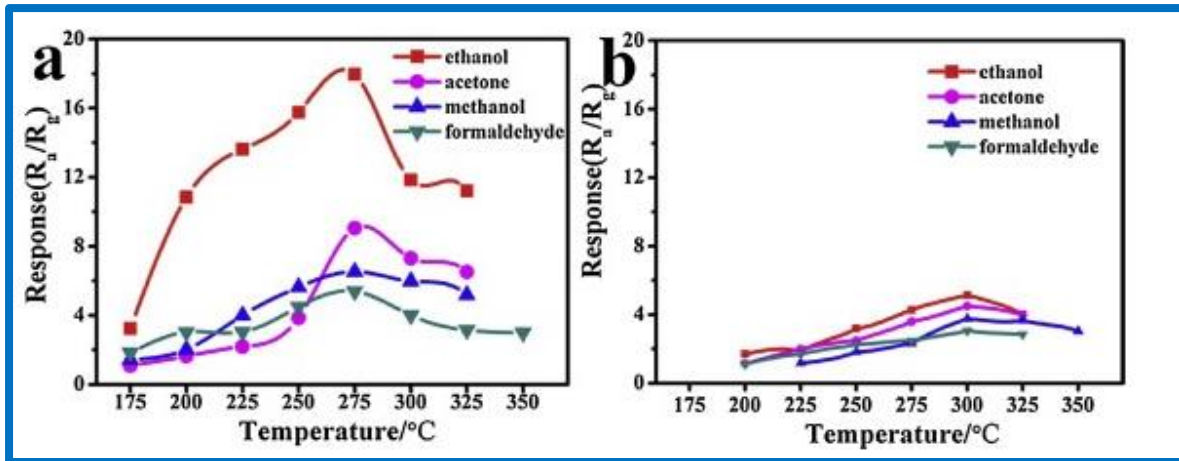


Figure 2 Response of the sample for a particular temperature for composite and for pure ZnO.

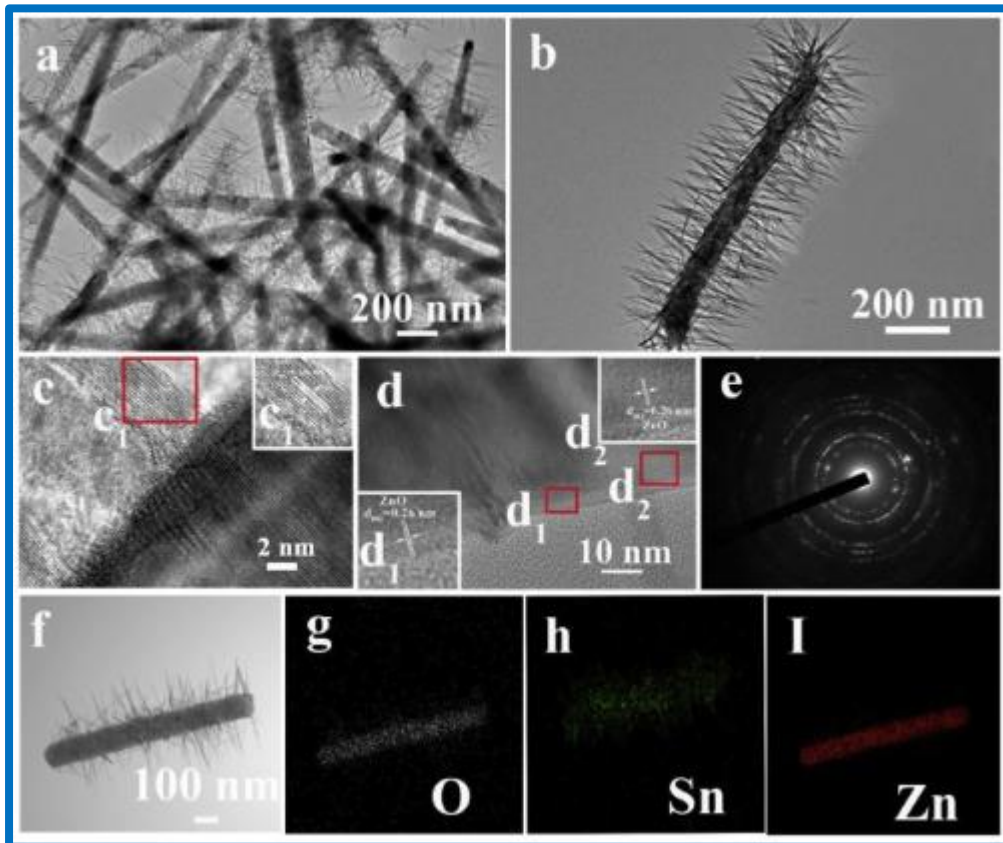


Figure 3 TEM and HR-TEM images of the SnO_2/ZnO sample.

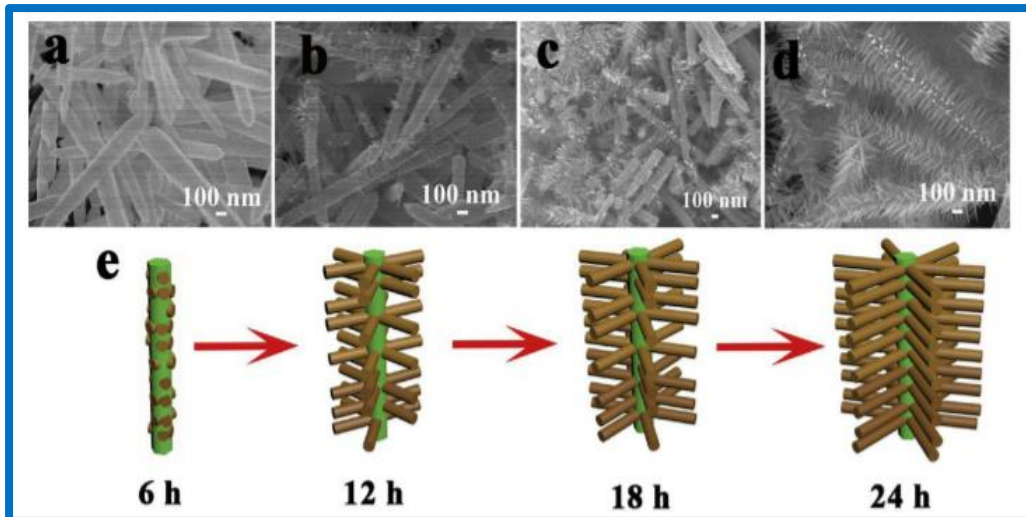


Figure 4 SEM images of (a and b) pure ZnO and (c–f) SnO₂/ZnO heterostructure

2.2 Paper 2

Room temperature highly toxic NO₂ gas sensors based on rootstock/scion nanowires of SnO₂/ZnO, ZnO/SnO₂, SnO₂/SnO₂ and, ZnO/ZnO” (2021).

In this paper composites of SnO₂ and ZnO rootstock/scion nanowires were grown using CVD method. And gas sensing parameters were studied for NO₂ gas for all the nanostructures.

In studies it was found that the SnO₂/ZnO has best sensing properties as compared to all other structures. The study was taken place at 100 ppb for NO₂.

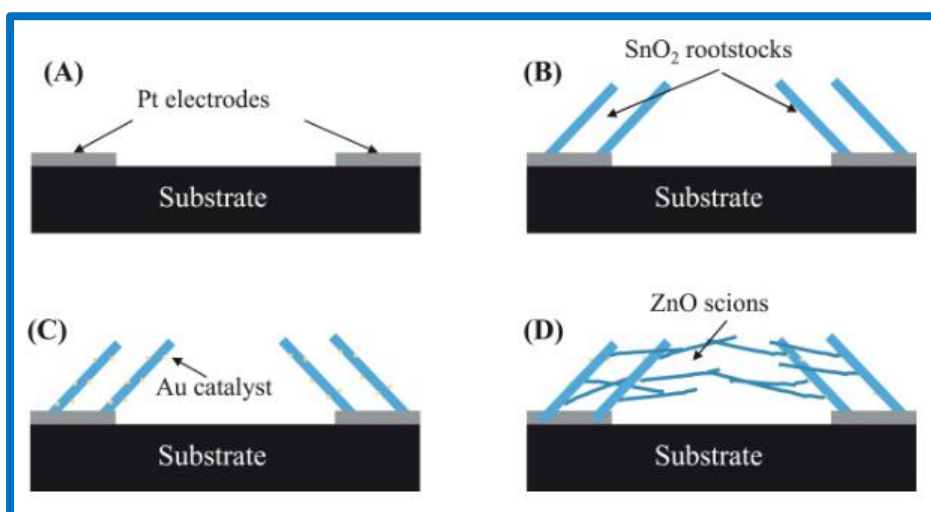


Figure 5 Fabrication process for the grafted nanowires of SnO₂/ZnO NWS.

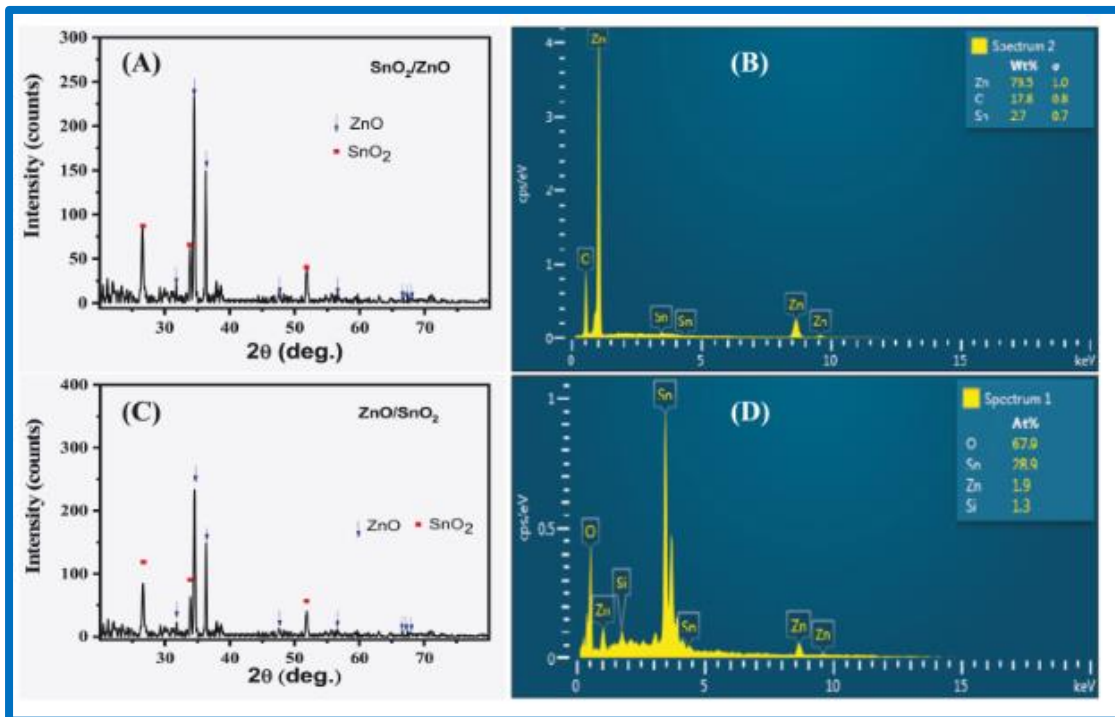


Figure 6 XRD pattern and EDS analysis of the composite samples.

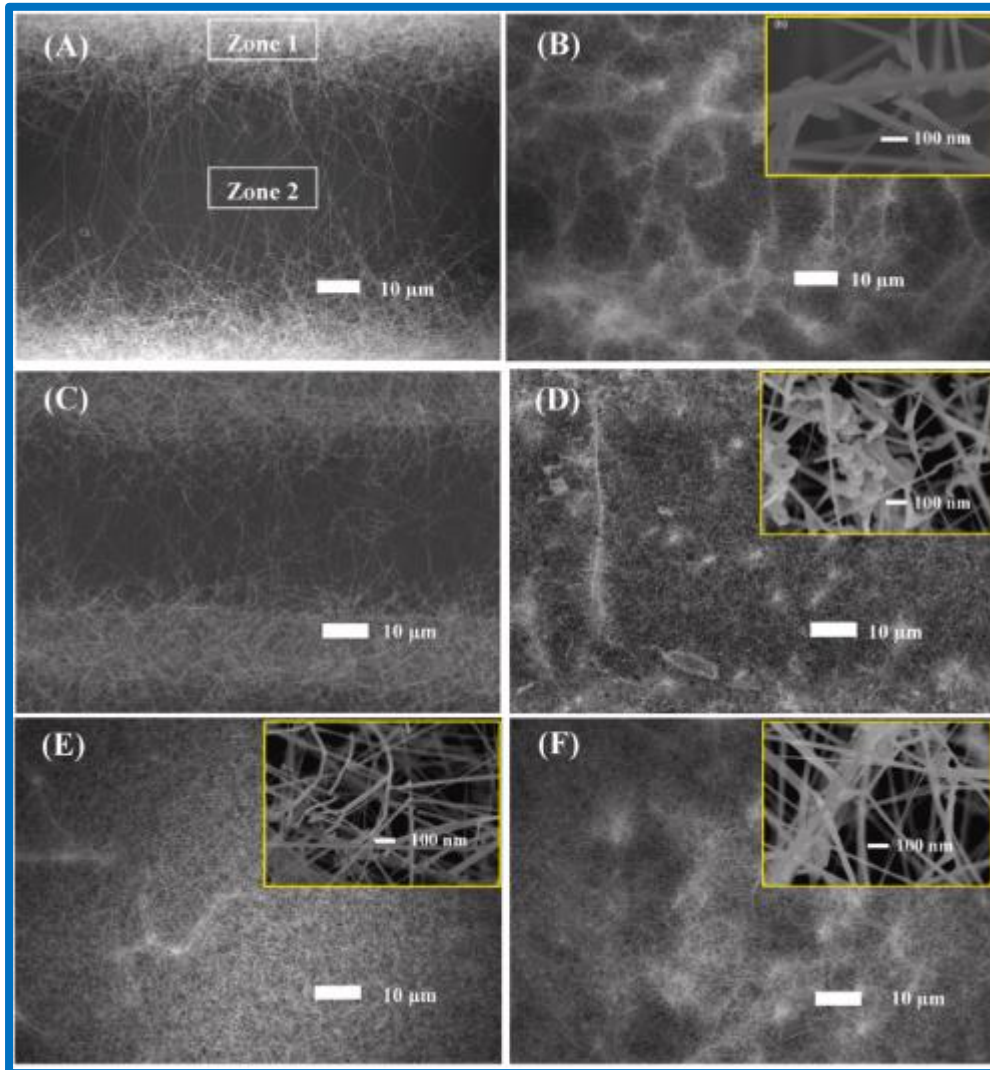


Figure 7 FESEM images for composites.

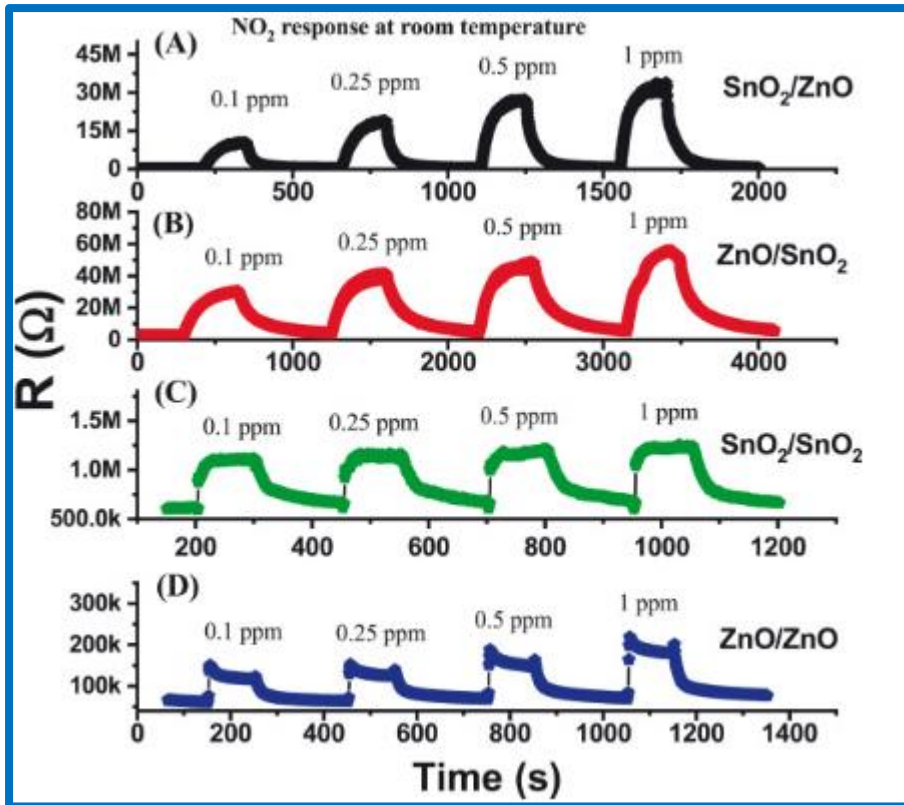


Figure 8 Gas sensing response parameter for different gases.

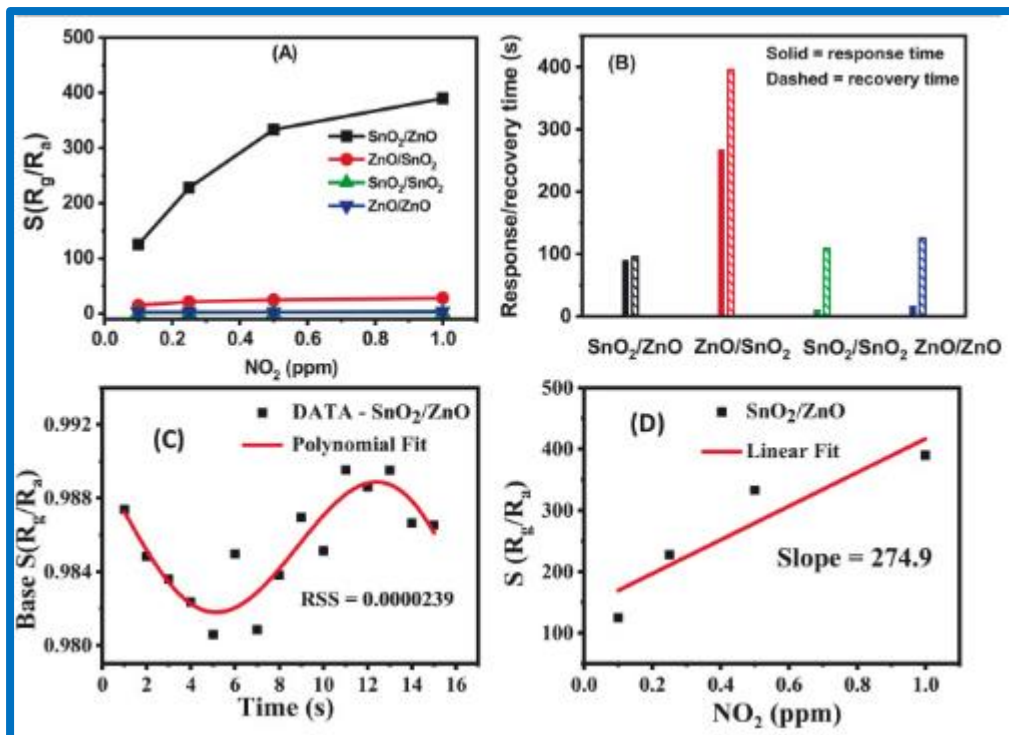


Figure 9 Comparison of gas sensing parameters for all 4 samples.

Result

Composite NWs were grafted using using two step synthesis process. In first step rootstocks were prepared and after that thermal evaporation technique was used to synthesis hetrostructures. TEM and SEM studies showed scions were sucessfully grown on the rootstocks. Gas sensing parameters were studied and it was observed that in concentration between 1 ppm to 100 ppm of NO₂ gas SnO₂ as rootstocks, homo or heterostructures both have similar enhanced properties than that of other two heterostructures (ZnO rootstocks) at 473 K temperature. And SnO₂/ZnO have best gas sensing parameters as comaperd to other three nanocomposites. And this composite is highly selective for the NO₂ gas.

CHAPTER 3

In this chapter complete process of synthesis of SnO₂ and ZnO is discussed with the proper quantity of precursors and other parameters that we have used and characterization of the produced sample using the XRD technique is also discussed.

3. Experimental

To synthesize the various nanostructures, a hydrothermal method was used and the schematic of the typical synthesis route is represented in Fig. 1. In the experimental process, analytical grades of chemical reagents were used as received without any modification.

3.1 Hydrothermal Method:

Compared with the “low-temperature” coprecipitation procedures that usually produce poor crystalline nanoparticles, the hydrothermal synthesis could get magnetic nanostructures with a very high crystalline nature due to their high-temperature and high-pressure reaction conditions. In this case, a water mixture of precursors is heated in a sealed stainless steel autoclave above the normal temperature, and consequently, the pressure within the reaction autoclave is dramatically increased above atmospheric pressure. This effect of high temperature and pressure provides a one-step process to produce highly crystalline materials and we don't need to do post annealing treatments. The reaction parameters, such as the type and concentration of the precursors, the solvent, the stabilizing agents, and the reaction temperature and time, present important effects on the products.

Hydrothermal synthesis refers to the heterogeneous reactions for synthesizing inorganic materials in water above room temperature and pressure. Hydrothermal strategies have been developed to synthesize a large range of nanostructures, including magnetic nanostructures.

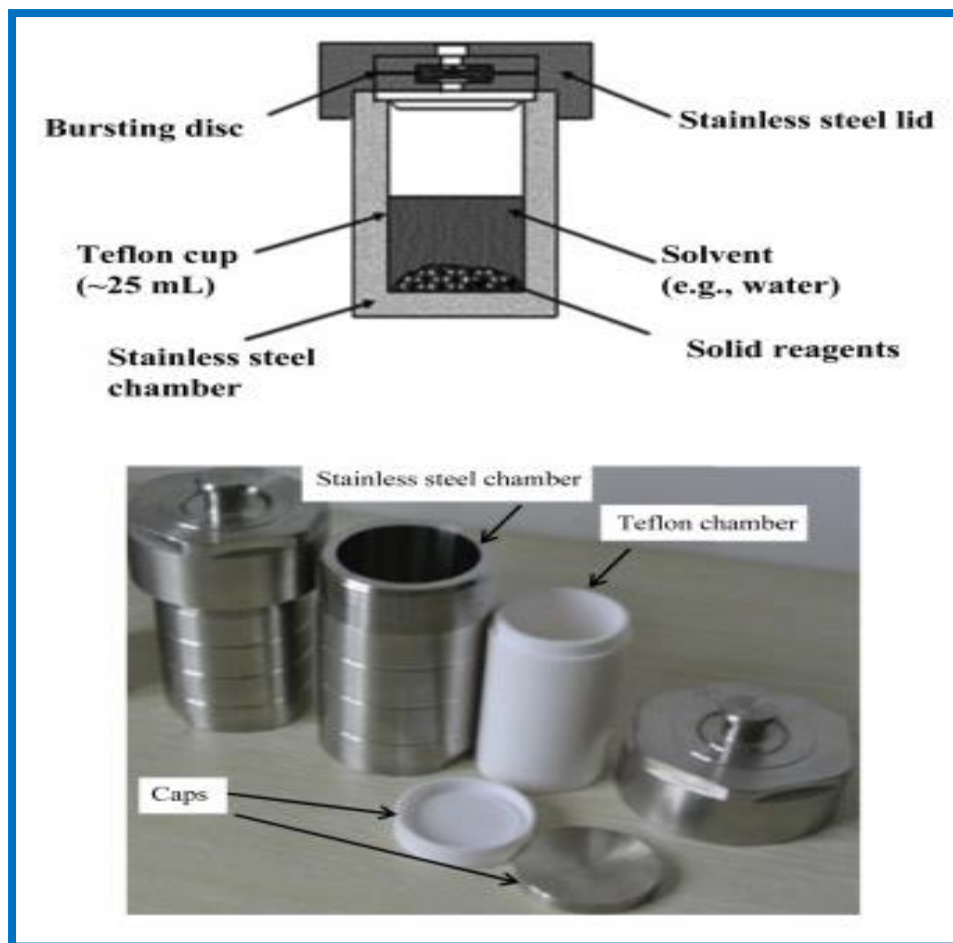


Figure 10 Schematic diagram of Autoclave

3.2 Synthesis of SnO₂ nanorods

Firstly, a solution in ethanol-water mixture, containing Tin tetrachloride (SnCl₄·5H₂O) (2 mM) and KOH (of 20 mM) was prepared. The solution was then stirred constantly for 50 minutes. The so formed solution was then kept in an oven for 24hrs at 180 °C, after transferring it to a hydrothermal container (Teflon lined stainless steel autoclave). After cooldown at a room temperature, the solution was then centrifuged for 10 minutes at 5000rpm. The white precipitate is collected and washed (three times, using DI) and followed by drying in an oven at 60°C for 24hrs.

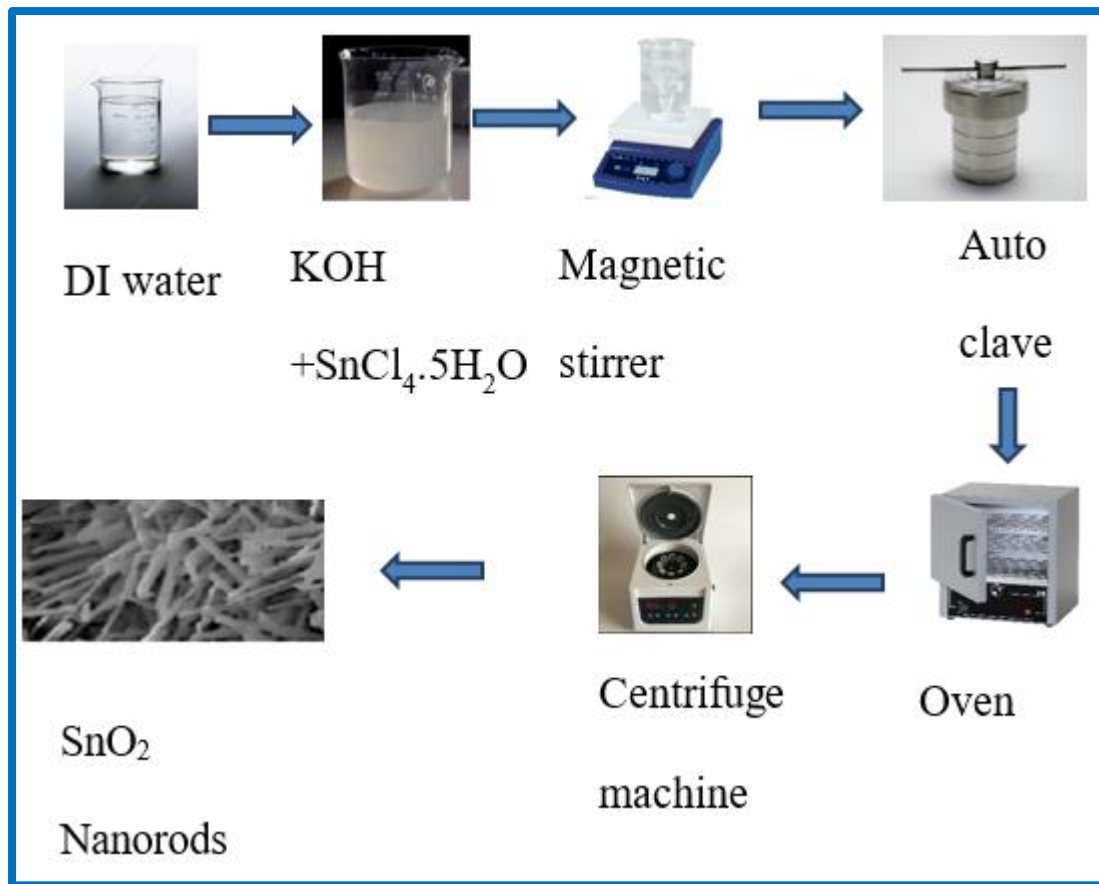


Figure 11 Hydrothermal method for SnO₂ nanorods Synthesis.

3.3 Synthesis of ZnO nanorods

An Exactly similar route (as in SnO₂ synthesis) was then used for the synthesis of ZnO nanorods except the concentration of salts and autoclave temperatures. Zinc chloride (ZnCl₂) 0.267 g (0.032 M), and sodium carbonate (Na₂CO₃) 26.67 g (4.193 M) were used to prepare the solution in ethanol-water mixture (1/1, V/V). The solution was heated at 140 °C in the Teflon autoclave for 24 hours [9].

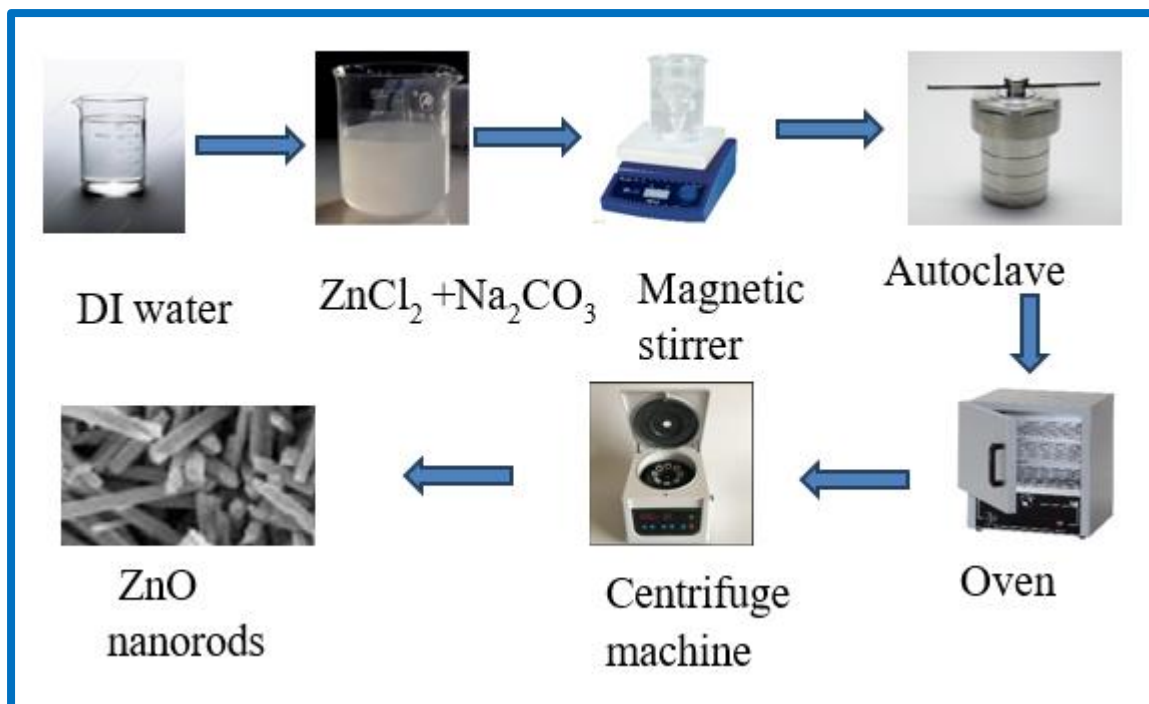


Figure 12 Synthesis of ZnO nanorods by Hydrothermal Method.

3.4 Synthesis of SnO₂/ZnO

2 mmol (2.104 g) of zinc sulfate heptahydrate (ZnSO₄·7H₂O) and of 1mmol (0.594 g) of tin tetrachloride pentahydrate (SnCl₄·5H₂O) were dissolved in an appropriate amount of DI water and then allowed to constantly stir for 20 min. Then to the solution formed above, an addition of 15 mmol of NaOH and then allowed to stir for 20 more minutes. The stirred solution was then kept in an oven for 24hrs at 180 °C, after successfully transferring it to a hydrothermal container (Teflon lined stainless steel autoclave). The after-processing (washing and drying) is carried out the same as was done for the synthesis of ZnO and SnO₂.

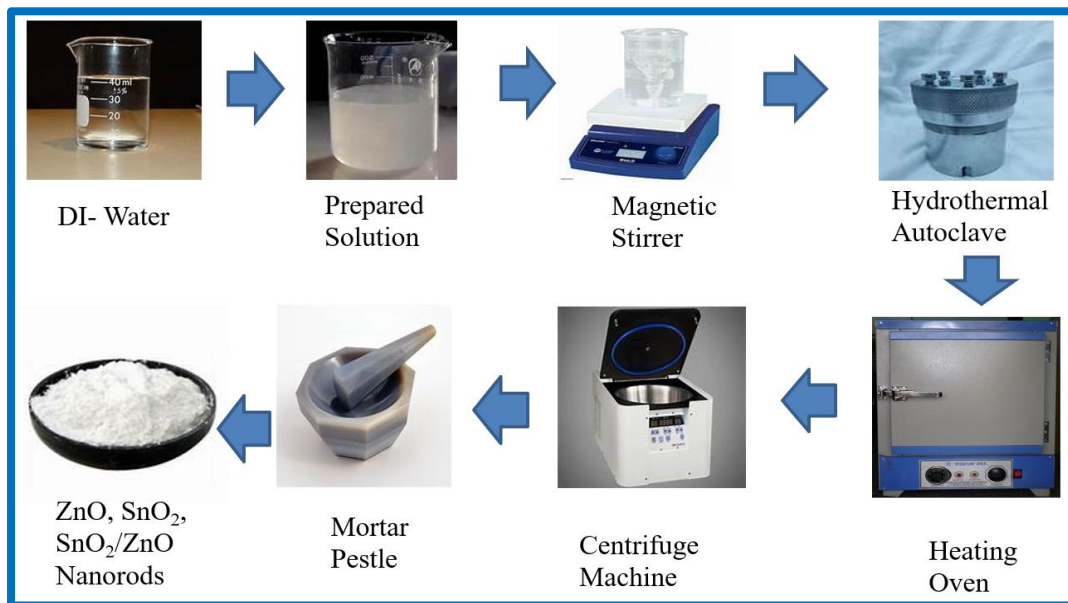


Figure 13 Schematic representation of synthesis process used for pure SnO₂, ZnO and SnO₂/ZnO composite.

CHAPTER 4

4. Results and Discussion

4.1 X-Ray Diffraction:

In the electromagnetic spectrum, the wavelength range of X-rays lies between 0.01 to 10 nm. X – rays were discovered by Wilhelm Conrad Rontgen in 1895. After their discovery, they are continuously being used in the science and technology domains.. And X-ray Diffraction from crystals was discovered by Laue in 1912. Since then, X – Rays are used by material scientists and crystallographers for the characterization of materials.

4.2 Bragg's Law:

Based on the phenomena of diffraction through a crystal, Bragg gave a simple formula using interplanar spacing 'd' of crystal planes and 'θ' is the angle of the incidence of X – rays demonstrated in figure13. The formula is written as:

For constructive interference, path length difference should be integer number of wavelengths
→ $n \lambda = 2 d \sin \theta$

Where λ - wavelength of X – rays

$n = 1, 2, 3, \dots$

Bragg's law is true for all $\lambda \leq 2d$.

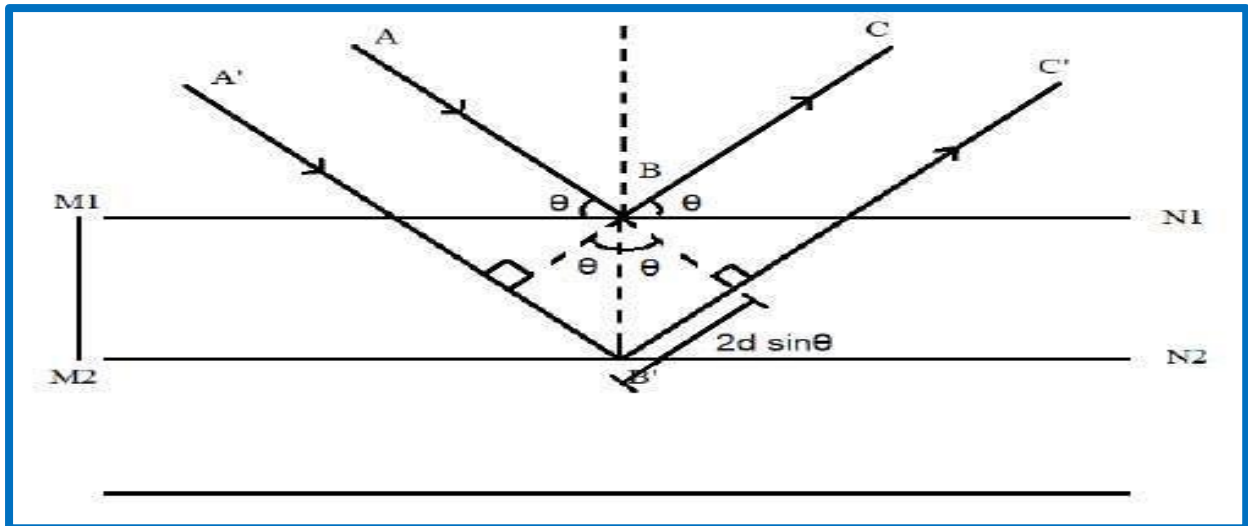


Figure 14 Bragg's Law.

The generated XRD pattern has three information of the sample; space group determined by the location of the peaks, information about the basis is given by the intensity of the peaks, and the determination of crystallite size and strain with the help of width of the peaks.

4.3 X-ray diffraction analysis:

The structural properties of the synthesized nanostructures were analysed using X-ray diffraction (XRD). The obtained XRD pattern of the synthesized samples were shown in Fig. 14. The sharp peaks at different 2θ values in XRD pattern indicates that synthesized nanostructures have good crystallinity and polycrystalline in nature. The 2θ position of all the diffraction peaks of SnO_2 and ZnO are well indexed and found in total agreement with the standard JCPDS [88-0287 and 00-036-0161] data. Which confirms the synthesized nanostructures of SnO_2 and ZnO have tetragonal rutile and hexagonal wurtzite structure, respectively. However, two additional peaks at $2\theta = 55.18^\circ$, 60.60° observed for ZnO nanostructures with very small intensity. These peaks may arise due to the presence of undissolved Na_2CO_3 which was used as a precursor during synthesis. Also, all the major XRD peaks of SnO_2/ZnO composite nanostructures were matched with the peaks of pure SnO_2 and ZnO . It has been observed that composite also has two additional peaks at $2\theta = 55.18^\circ$, 60.60° with very small intensities. The absence of any additional major diffraction peaks suggest that the synthesized SnO_2/ZnO composite has formed a heterojunction of SnO_2 and ZnO without going through any structural deformation and phase transformation.

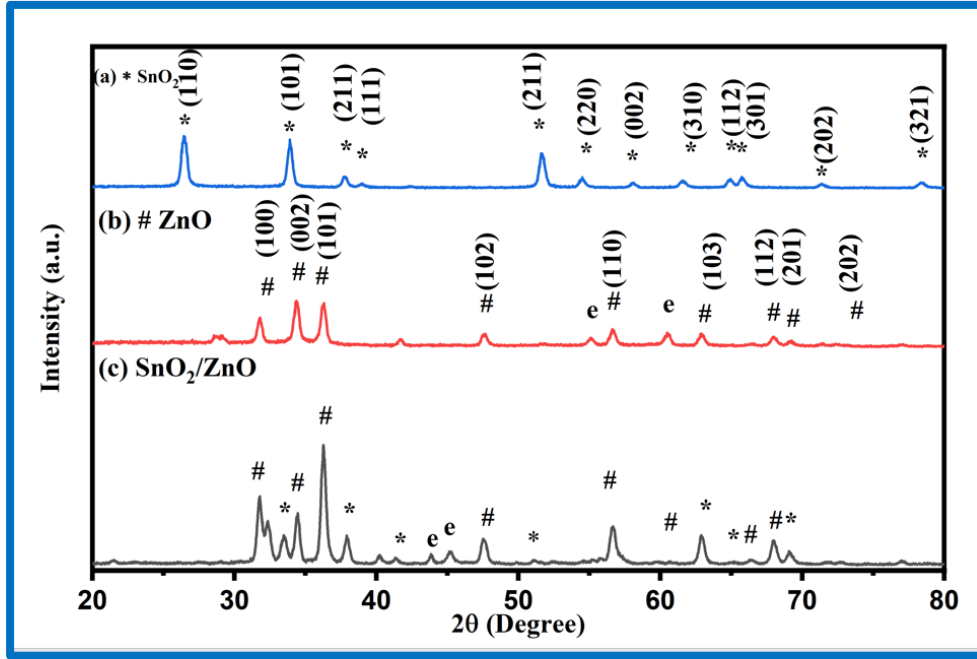


Figure 15 XRD patterns of synthesized (a) SnO₂, (b) ZnO and (c) SnO₂/ZnO composites

Furthermore, the crystallite size was calculated using Scherrer's formula (eq.1) and Williamson-Hall (eq.7) plot and presented in Table 1. It has been observed that crystallite size for all three nanostructures are almost similar. The crystallite size calculated using the W-H method has smaller values as compared to the values calculated using Scherrer formula. It might be because of the Scherrer equation only considers the influence of crystallite size on XRD peak broadening and excludes microstructures of the lattice, such as intrinsic strain, which develops in nanocrystals through point defects, grain boundaries, triple junctions, and stacking faults. However, the variation in crystallite size of the synthesized nanostructures are similarly obtained by both the methods. Also, the other structural parameters of the synthesized nanostructures were calculated using equations 2-4 and tabulated in Table 1. The calculated values of Stacking Fault (SF) and macrostrain (ϵ) are comparable to the reported data [19]. Smaller values of SF and macrostrain further strengthen that synthesized nanostructures have proper periodic arrangement of lattice planes and almost negligible lattice defects.

(1)

$$D = \frac{K\lambda}{\beta \cos \theta}$$

(2)

$$\delta = \frac{1}{D^2}$$

(3)

$$SF = \left[\frac{[2\pi^2]}{45\sqrt{\tan\theta}} \right] \beta$$

(4)

$$\varepsilon = \frac{\beta}{4\tan\theta}$$

Where λ is wavelength of radiation (1.5406 Å), 'D' is crystallite size, 'k' is constant (0.94), θ is Bragg's angle in radian and β is the full width at half maximum (FWHM) in radians. In accordance to W-H method the overall broadening (FWHM) of the XRD peaks occurs due to crystallite size (D) and lattice strain (ε) of the nanostructures (eq. 5). Uniform deformation model (UDM) has been considered to obtain the lattice strain and crystallite size. This model assumes the induced lattice strain due to lattice defects and imperfections in the nanomaterials is isotropic in nature [1,2].

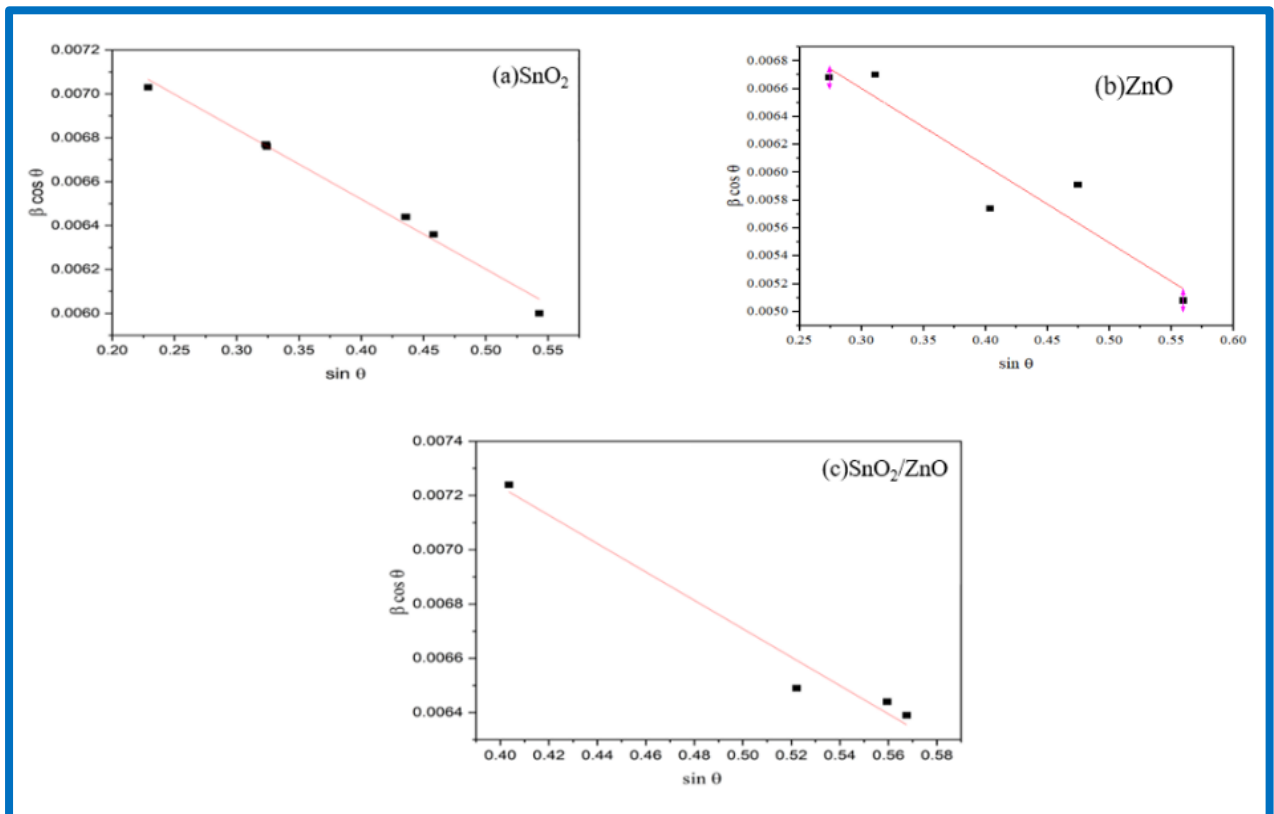


Figure 16 W-H analysis of synthesized (a) Pure SnO₂, (b) Pure ZnO and (c) SnO₂/ZnO composites.

$$\beta_{strain} = 4\epsilon \tan\theta \quad (5)$$

$$\beta_{hkl} = \frac{k\lambda}{D} \frac{1}{\cos\theta} + 4\epsilon \tan\theta \quad (6)$$

$$\beta_{hkl} \cos\theta = \frac{k\lambda}{D} + 4\epsilon \cdot \sin\theta \quad (7)$$

The y-intercept and slope of the linear fitted line plot of eq. 7 for various crystallographic planes gives the values of crystallite size and lattice strain, respectively. The W-H plot of the synthesized samples are shown in Fig. 15.

Table 1 Calculated values of various structural parameters by using Scherrer equation and W-H Methods from the XRD data.

Sample	Crystallite Size(D) (nm)		Dislocation density(δ)	Stacking Fault (SF)	Micro Strain (ϵ)	Lattice strain (W-H plot)
	Scherrer's Equation	W-H plot				
Pure SnO ₂	32.83	21.40	9.2x10 ⁺¹⁸	0.006309	0.006450	-0.00319
Pure ZnO	35.99	23.85	7.9x10 ⁺¹⁸	0.005423	0.005540	-0.00509
Composite SnO ₂ /ZnO	34.47	21.63	9.1x10 ⁺¹⁸	0.005517	0.004830	-0.00523

4.4 EDS analysis

The Energy Dispersive X-ray spectroscopy (EDS) analysis of pure SnO₂, pure ZnO and SnO₂/ZnO composite is shown in Figure 16 (a-c). The EDS spectrum confirms the chemical composition of the synthesized nanostructures and absence of any additional major peaks in

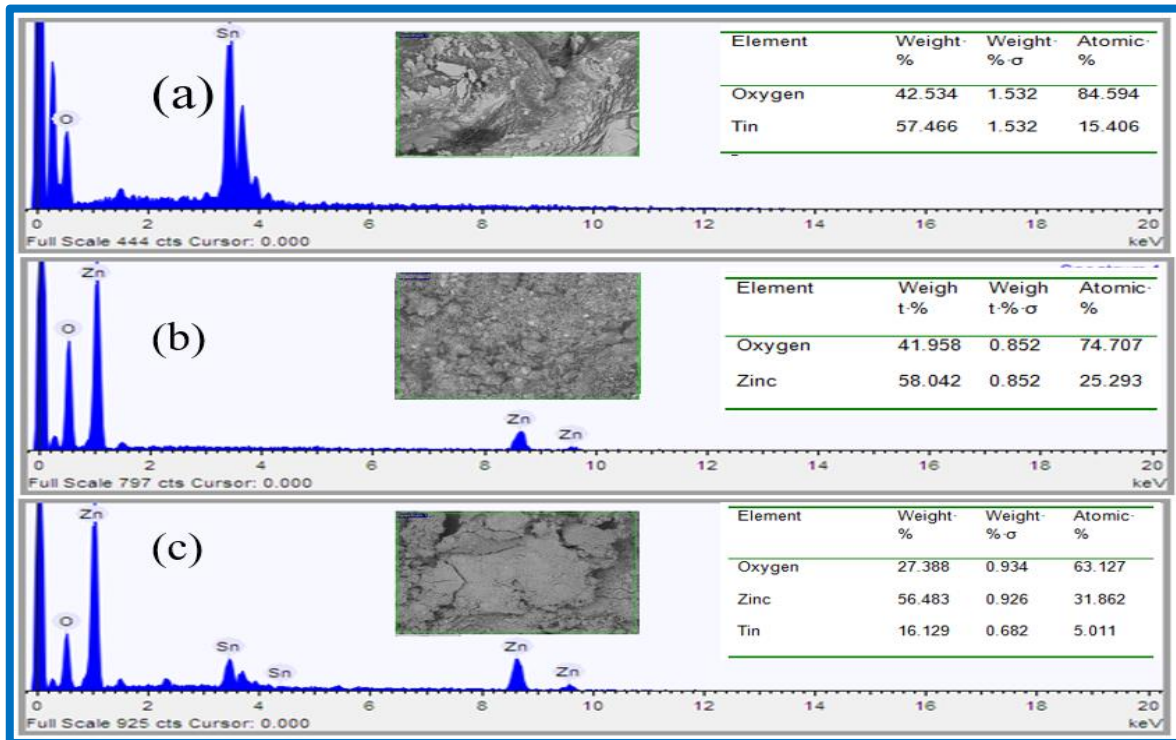


Figure 17 EDS analysis of a) pure SnO₂ nanorods b) pure ZnO nanowires C) SnO₂/ZnO.

spectra suggest that synthesized nanostructures have no impurity [18] [19].

4.5 UV-Visible spectroscopy:

UV-Vis spectroscopy is an expository strategy that measures the sum of discrete wavelengths of UV or unmistakable light that are ingested by or transmitted through a test in comparison to a reference or clear test. This property is affected by the test composition, possibly giving data on what is within the test and at what concentration. Since this spectroscopy method depends on the utilization of light.

How does a UV-Vis spectrophotometer work?

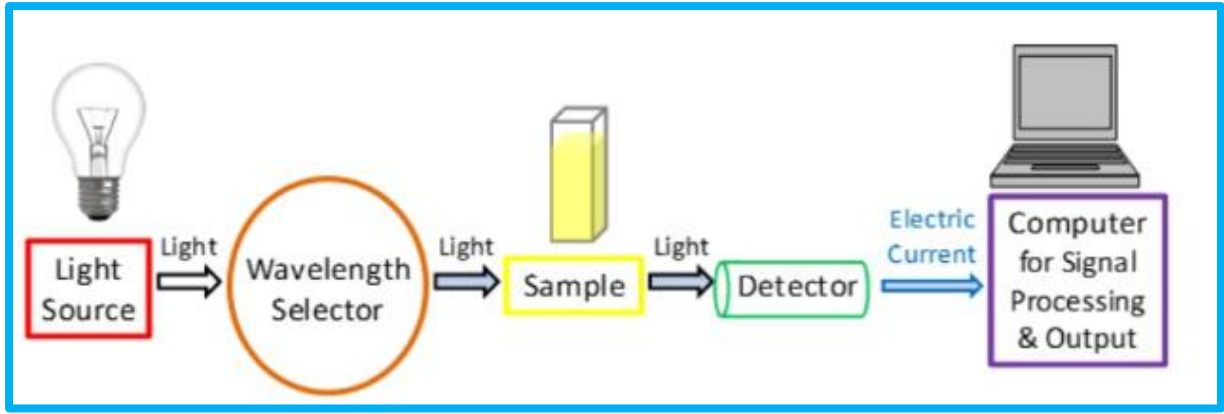


Figure 18 UV-Visible spectroscopy.

The strategy is non-destructive, permitting the test to be reused or continue to advance handling or analyses. Measurements can be made rapidly, permitting simple integration into test protocols. Instruments are simple to utilize, requiring small client preparing earlier to use. Data examination by and large requires negligible preparing, once more meaning small client preparing is required. The instrument is for the most part cheap to secure and work, making it open for numerous laboratories.

The absorbance spectra of pure SnO₂, ZnO, and SnO₂/ZnO composites nanostructure was recorded in the range of 200–800 nm using UV–Visible spectrophotometer at 300 K temperature and shown in Figure 5 (a-c). UV–Visible spectrum shows the information about the excitonic and inter transition of nanostructures. The absorption coefficient (α) has been calculated using Beer-Lambert's law by equation 8.

$$\alpha(\lambda) = \frac{\ln \ln \left(\frac{I}{I_0} \right)}{x} = \frac{\ln T}{x} = \frac{2.303A}{x} \quad (8)$$

Where I and I_0 are instantaneous and initial value photon intensity, respectively. The thickness of the cuvette is represented by x and A represents the absorbance, which is a unit less quantity [13] [19]. To obtain the optical band gap of synthesized nanostructures we have plotted the Tauc plot between $(\alpha h\nu)^2$ versus photon energy ($h\nu$) as shown in Figure 5 (a-c) by using equation 9.

$$(\alpha h\nu)^2 = A(h\nu - E_g)^n \quad (9)$$

where A is a constant (energy independent), α is the absorption coefficient. E_g represents optical band gap while n is another constant which depends on the type of transition.

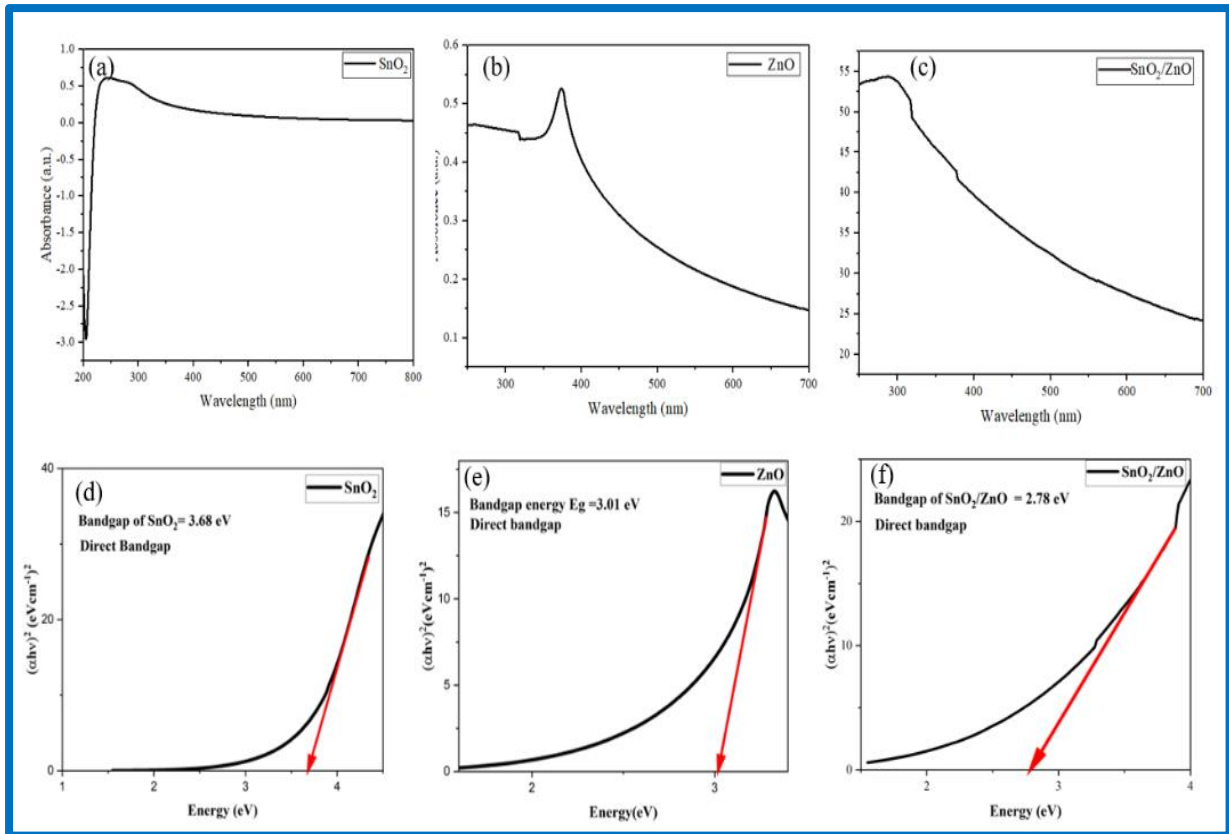


Figure 19 UV-Visible absorption spectrum of pure SnO₂, ZnO and SnO₂/ZnO and their Tauc plot.

To calculate band gap from the Tauc plot an intercept on x axis (energy axis) obtained by extrapolating the linear portion of the Tauc plot. The obtained values for direct band gap of pure SnO₂, ZnO and SnO₂/ZnO composite are 3.68 eV, 3.01 eV and 2.78 eV respectively. The obtained bandgap results are in accordance with the reported literature [12]. SnO₂/ZnO composite have lower direct band gap than pure ZnO and SnO₂. The reduction in bandgap of composite might be because of the generation of new energy levels of ZnO and SnO₂ on formation of composite nanostructure [12].

4.6 FTIR Analysis

Fourier change infrared (FTIR) spectroscopy tests the vibrational properties of amino acids and cofactors, which are delicate to miniature auxiliary changes. The need of specificity of this strategy, on the one hand, licenses us to test specifically the vibrational properties of nearly all the cofactors, amino corrosive side chains, and of water particles. On the other hand, we will

utilize reaction-induced FTIR distinction spectroscopy to choose vibrations comparing to single chemical bunches included in a particular response. Different strategies are utilized to distinguish the IR marks of each buildup of intrigued within the coming about reaction-induced FTIR distinction spectra.

(Particular) Isotope labeling, site-directed mutagenesis, hydrogen/deuterium trade are regularly utilized to distinguish the chemical bunches. Thinks about on show compounds and the expanding utilize of hypothetical chemistry for typical modes calculations permit us to decipher the IR frequencies in terms of particular auxiliary characteristics of the chemical bunch or atom of intrigued. This audit presents essentials of FTIR spectroscopy strategy and gives particular critical auxiliary and utilitarian data gotten from the examination of the information from the photosystems, utilizing this method.

The FTIR spectra of using ATR (Attenuated total reflection) mode, for ZnO, SnO₂, SnO₂/ZnO composites, the FTIR was recorded and various functional groups were obtained which can be confirmed using different absorption peaks as shown in the plots in Fig. 19. Absorbance band around at 607.6 cm⁻¹ could be for O-Sn-O, Sn-O lattice which is in accordance with the results obtained by [23]. Further, the absorptions at 3433 cm⁻¹ of infrared spectra assigned to hydroxyl groups of water molecules adsorbed on the outer surface of SnO₂ nanorods. Similarly, the absorption peak appearing at 3478 cm⁻¹ in the IR spectrum of ZnO nanorods could be assigned to the O-H vibrations, which confirms the presence of hydroxyl (OH) groups on the external surface of ZnO nanorods. Additionally, the appearance of the absorbance band at 498.5 cm⁻¹ confirms the formation of ZnO nanorods as reported in literature [22]. The absorbance band at 3423 cm⁻¹ appears in the IR spectrum of SnO₂/ZnO composite [21].

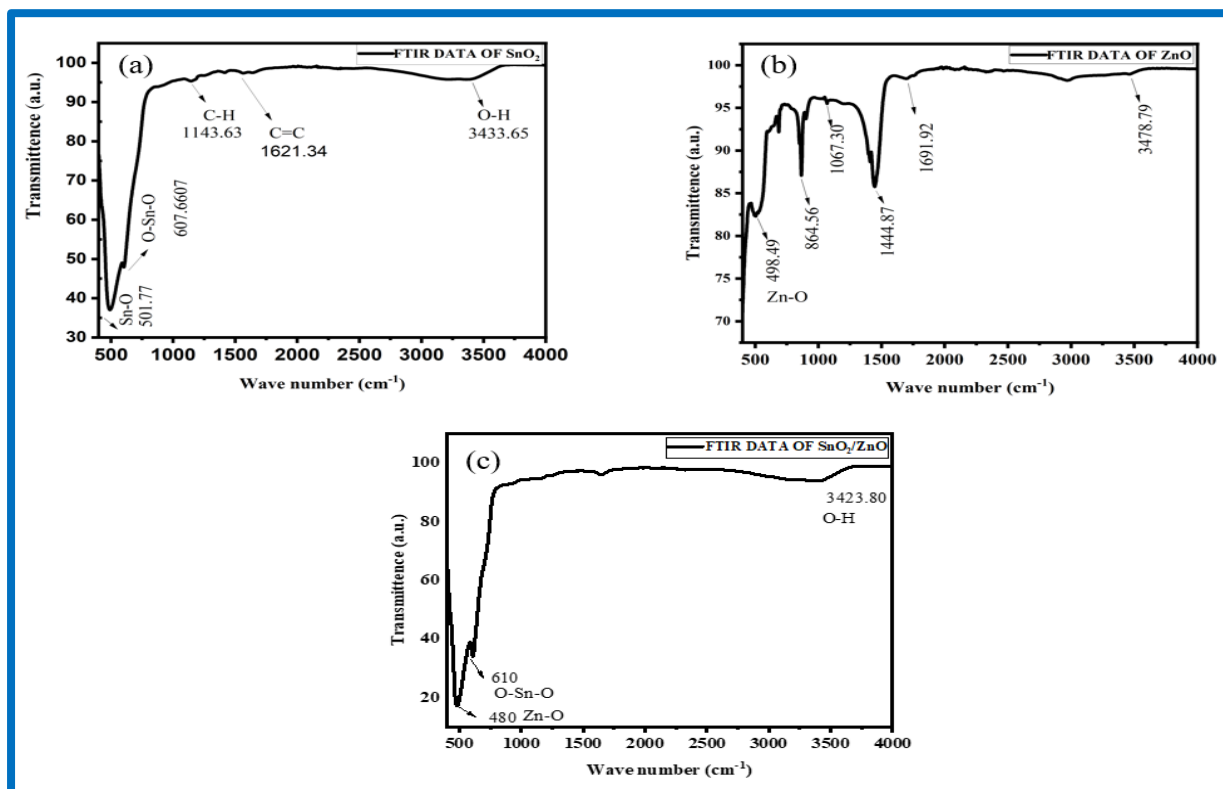


Figure 20 FTIR graphs of pure SnO₂, ZnO and SnO₂/ZnO composite.

4.7 TEM Analysis

Transmission electron microscopy (TEM) is a method in which a pillar of electrons are transmitted through sample and picture is created by these transmitted electrons. The sample is an ultrathin segment of size less than 100 nm or suspension on the framework. Picture is formed from the electrons which are transmitted through the sample. The image is amplified at that point and centered to the imaging gadget, like photographic film, fluorescent screen or scintillators like sensors joined by charged coupled devices.

TEM magnifying lenses are competent of taking highly resolved images owing to de broglie wavelength of electrons. It helps the instrument to capture small details which are thousands times smaller than a resolvable sample seen in the light magnifying instrument. TEM can be a major strategy to explain physical, chemical and natural sciences. It can help in investigation of cancer cells, virology and material science. It can be also used to investigate contamination, nanotechnology and semiconductor devices.

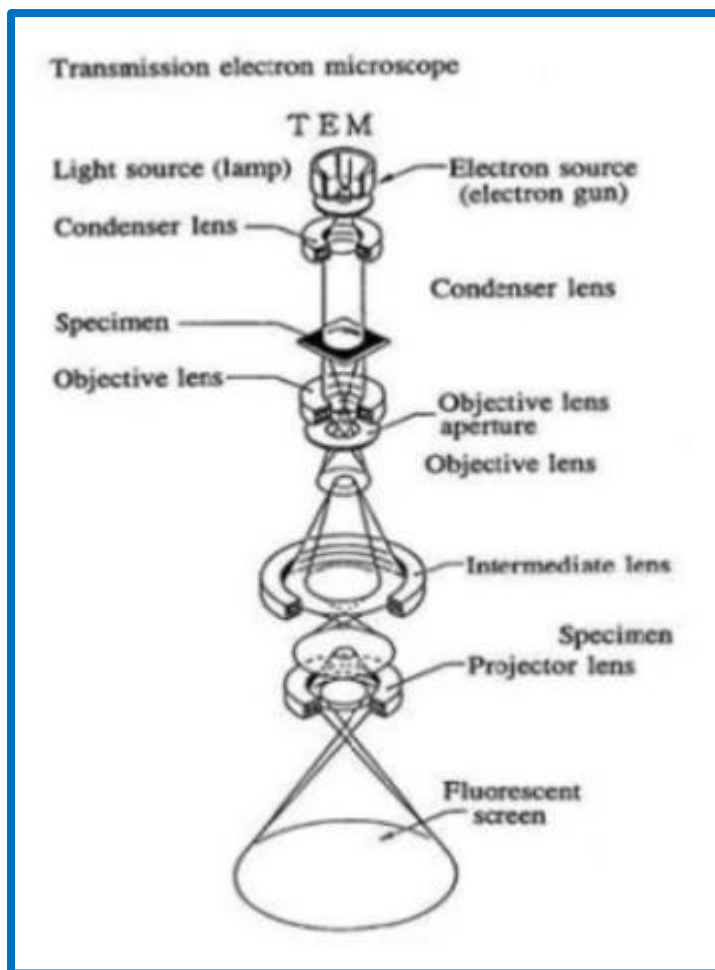


Figure 21 Schematic diagram of TEM.

The obtained Transmission Electron Microscopic (TEM) images of the synthesized samples are shown in Fig. 21 (a-i) at three different magnifications/scales. It can be observed that the morphology of the synthesized SnO_2 , and ZnO are nanobelts and nanorods, respectively. While the nanostructure of SnO_2/ZnO composites is a combination of nanobelts and nanorods as shown in Fig. 21 (g-i). Lattice resolved TEM fringes of surface of the SnO_2 , ZnO and SnO_2/ZnO composites are represented in Fig. 22 along with their SAED pattern. The spacing between two neighbouring fringes has been calculated from TEM image (e-f). The obtained average spacing for SnO_2 and ZnO fringes are 0.33 nm and 0.25 nm which correspond to (110) of SnO_2 and ZnO (002) crystal planes. The lattice fringes of SnO_2/ZnO composite with average interplanar spacing of 0.33 nm and 0.25 nm are also matched with (110) and (002) planes of SnO_2 and ZnO , respectively. This further confirms that the SnO_2/ZnO composite is a two-phase composite material composed of SnO_2 and ZnO .

The SAED, Figure 8 (a-c), clearly shows the continuous ring diffraction pattern due to the SnO_2 containing a large number of small randomly distributed crystallites for different planes like

(101), (211) and (401). ZnO have SAED pattern showing planes (002) and (103). In composite (002) and (101) planes of SnO₂ and ZnO are overlapping with each other while (110) and (103) planes of ZnO are clearly visible.

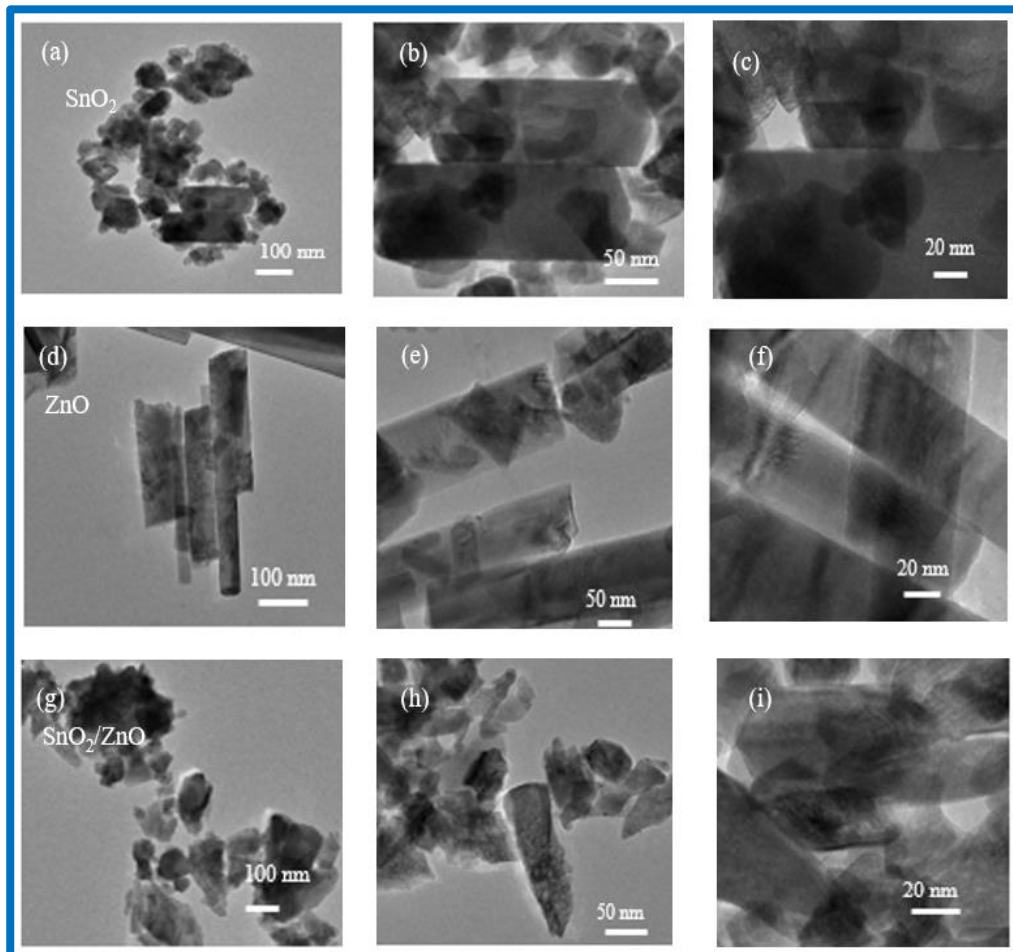


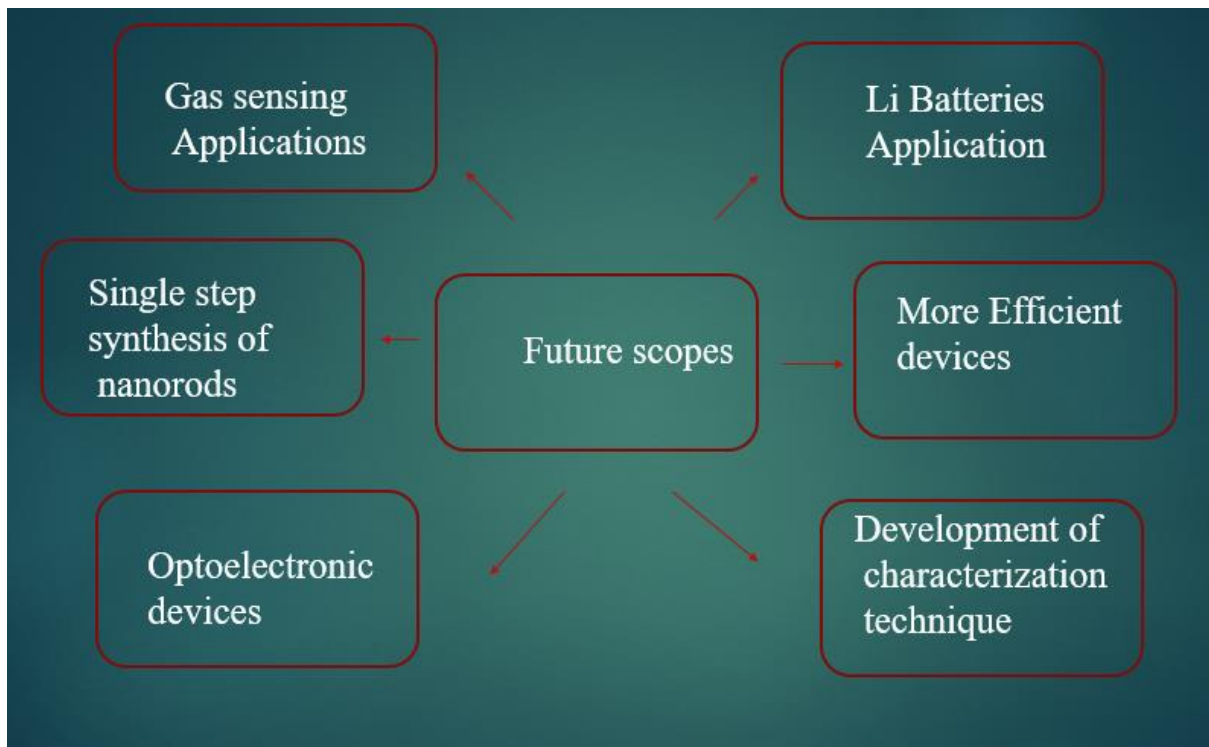
Figure 22 A typical magnification TEM image of (a-c) SnO₂ (d-f) ZnO (g-i) SnO₂/ZnO.

CHAPTER 5

5. Conclusion and Future scope

Pure SnO₂, ZnO and SnO₂/ZnO composite nanostructures were prepared by using hydrothermal methods. The various structural parameters of synthesized samples were obtained by using Scherrer and W-H Methods. The results indicate that the composites structure has very few additional defects and strain as compared to pure SnO₂ and ZnO. While the observed optical band gap of SnO₂/ZnO composite is smaller than pure SnO₂, and ZnO. Which might be due to generation of new energy levels in the composite structure. The FTIR spectrum further confirms the synthesis of pure SnO₂, pure ZnO and SnO₂/ZnO composites. This study suggests that on synthesis of SnO₂/ZnO composite the structural properties such as phase, lattice strain, defects did not change significantly while the UV-Visible and FTIR absorption peaks get shifted as compared to pure SnO₂ and ZnO. These results are found to be interesting for the application of SnO₂/ZnO composite in the optoelectronic and gas sensing devices.

5.1 Future Scope



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Investigation of structural and optical properties of pure SnO₂, ZnO and SnO₂/ZnO composite nanorods

Vineeta Yadav*, Nirmal Singh, Deshraj Meena

Department of Applied Physics, Delhi Technological University, New Delhi 110042, India

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ABSTRACT

Nanorods of SnO₂, ZnO and SnO₂/ZnO composites were successfully synthesised by the hydrothermal method in a one step process under specific conditions. The lattice strain and crystallite size of the nanorods were calculated using Williamson-Hall and Scherrer method. TEM (Transmission Electron Microscopy) results confirm the synthesized samples are nanorods. The various crystalline planes of the nanorods were also identified using SAED (Selected Area Electron Diffraction) Pattern. The EDS (energy dispersive spectroscopy) result confirms the chemical purity of the nanorods. Furthermore, the optical band gap of SnO₂/ZnO nanocomposite was found to be lesser than the pure optical bandgaps of either SnO₂ and ZnO nanorods. Various functional groups were found to be on the surface of the synthesized nanorods, from the FTIR analysis.

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1. Introduction

In recent years, many applications, including gas sensors [1], solar cells [2], lithium-ion batteries (LIBs), and other optoelectronic nanodevices [5–8] are studied for metal oxide nanowires, nano-tubes and nanofibers (Semi-conductor nanostructures), due to their appreciable electronic structural characteristics, optical properties and various chemical properties. The metal oxides that are formed between metal cations and oxide ions are the chemical compound. Moreover, due to their band gaps being wide and exciton binding energy being high, ZnO, TiO₂, SnO₂, CuO, Fe₂O₃ etc, are earlier reported to be some of the best holders of semiconducting properties in metal oxides due to their wide band gap and high exciton binding energy. Tin dioxide (SnO₂) and Zinc oxide (ZnO) have been widely used as photocatalyst for photodegradation of organic compounds, sensing material in resistive gas sensors, perovskite solar cells, liquid crystal display, biosensors, acoustic photodevices and photodiodes, electrochemical supercapacitors, catalysts and in the development of electrode materials for LIBs. Both of these metal oxides have high melting point and

large cohesive energy, excellent thermal and mechanical stability, good electrical conductivity, and high transparency in the visible spectrum. These properties make them obvious for the above application. Moreover, both are easily available and environment friendly. It has been observed that to obtain the desired materials with enhanced properties, the nanostructures of SnO₂ and ZnO composite has been used in many applications [3,4]. Due to creation of more electron and efficient transport of charge carriers in the composites of nanostructure have better properties. The efficiency for the devices based on pure metals are lower than the devices based on SnO₂/ZnO nanocomposites as reported in previous literature. Composites are advantageous as they have been found more porous in comparison with the pure metal oxides. Especially, SnO₂ can be made porous with the addition of ZnO in very small amounts [9–11].

In the present work, a comparative analysis of the structural and optical properties of SnO₂/ZnO nanocomposites with pure SnO₂ and ZnO has been done. The nanorods of the pure SnO₂, ZnO and SnO₂/ZnO composites were synthesized using hydrothermal method. The results reveal that the composites have lower optical band gap in comparison with pure SnO₂ and ZnO. The efficiency of charge separation is greatly improved by threshold energy difference of SnO₂ and ZnO, thus improving the optical properties.

Abbreviations: WH-Method, Williamson-Hall Method.

* Corresponding author. Tel.: +91-9140171561.

E-mail address: yadvavineeta38@gmail.com (V. Yadav).

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2. Experimental

To synthesize the various nanostructures, a hydrothermal method was used and the schematic of the typical synthesis route is represented in Fig. 1. In the experimental process, analytical grades of chemical reagents were used as received without any modification.

2.1. Synthesis of SnO₂ nanorods

Firstly, A solution in ethanol–water mixture, containing Tin tetrachloride (SnCl₄·5H₂O) (2 mM) and KOH (of 20 mM) was prepared. The solution was then stirred constantly for 50 min. The so formed solution was then kept in an oven for 24 hrs at 180 °C, after transferring it to a hydrothermal container (Teflon lined stainless steel autoclave). After cooldown at a room temperature, the solution was then centrifuged for 10 min at 5000 rpm. The white precipitate is collected and washed (three times, using DI) and followed by drying in an oven at 60 °C for 24hrs.

2.2. Synthesis of ZnO nanorods

An Exactly similar route (as in SnO₂ synthesis) was then used for the synthesis of ZnO nanorods except the concentration of salts and autoclave temperatures. Zinc chloride (ZnCl₂) 0.267 g (0.032 M), and sodium carbonate (Na₂CO₃) 26.67 g (4.193 M) were used to prepare the solution in ethanol–water mixture (1/1, V/V). The solution was heated at 140 °C in the Teflon autoclave for 24 hrs.

2.3. Synthesis of SnO₂/ZnO

2 mmol (2.104 g) of zinc sulfate heptahydrate (ZnSO₄·7H₂O) and of 1 mmol (0.594 g) of tin tetrachloride pentahydrate (SnCl₄·5H₂O) were dissolved in an appropriate amount of DI water and then allowed to constantly stir for 20 min. Then to the solution

formed above, an addition of 15 mmol of NaOH and then allowed to stir for 20 more minutes. The stirred solution was then kept in an oven for 24 hrs at 180 °C, after successfully transferring it to a hydrothermal container (Teflon lined stainless steel autoclave). The after-processing (washing and drying) is carried out the same as was done for the synthesis of ZnO and SnO₂.

3. Results and discussion

3.1. X-ray diffraction analysis:

The structural properties of the synthesized nanostructures were analysed using X- ray diffraction (XRD). The obtained XRD pattern of the synthesized samples were shown in Fig. 2. The sharp peaks at different 2θ values in XRD pattern indicates that synthesized nanostructures have good crystallinity and polycrystalline in nature. The 2θ position of all the diffraction peaks of SnO₂ and ZnO are well indexed and found in total agreement with the standard JCPDS [88–0287 and 00–036-0161] data. Which confirms the synthesized nanostructures of SnO₂ and ZnO have tetragonal rutile and hexagonal wurtzite structure, respectively. However, two additional peaks at 2θ = 55.18°, 60.60° observed for ZnO nanostructures with very small intensity. These peaks may arise due to the presence of undissolved Na₂CO₃ which was used as a precursor during synthesis. Also, all the major XRD peaks of SnO₂/ZnO composite nanostructures were matched with the peaks of pure SnO₂ and ZnO. It has been observed that composite also has two additional peaks at 2θ = 55.18°, 60.60° with very small intensities. The absence of any additional major diffraction peaks suggest that the synthesized SnO₂/ZnO composite has formed a heterojunction of SnO₂ and ZnO without going through any structural deformation and phase transformation.

Furthermore, the crystallite size was calculated using Scherrer's formula (eq.1) and Williamson-Hall (eq.7) plot and presented in Table 1. It has been observed that crystallite size for all three nanostructures are almost similar. The crystallite size calculated

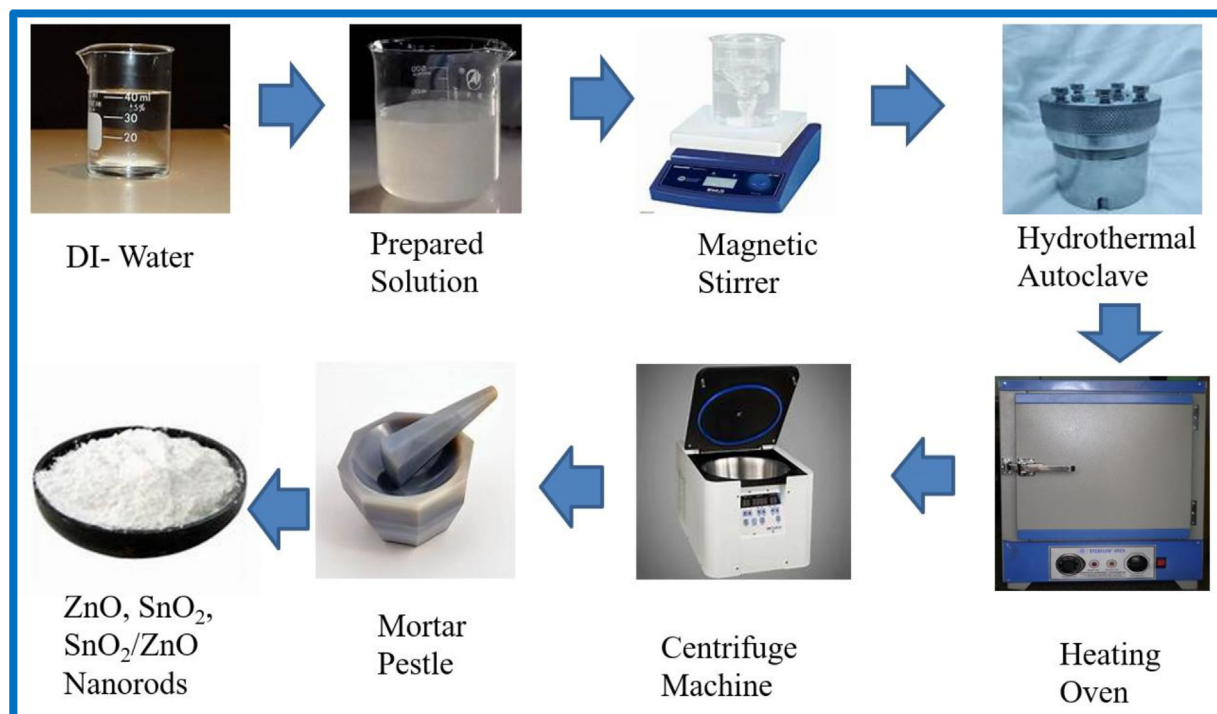


Fig. 1. Schematic representation of synthesis process used for pure SnO₂, ZnO and SnO₂/ZnO composite.

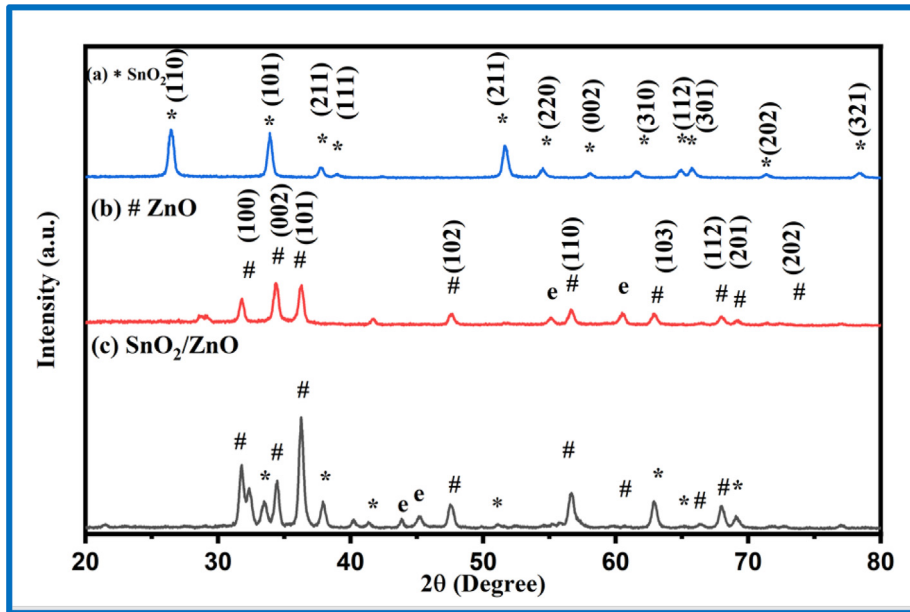


Fig. 2. XRD patterns of synthesized (a) SnO₂, (b) ZnO and (c) SnO₂/ZnO composites.

Table 1

Calculated values of various structural parameters by using Scherrer equation and W-H Methods from the XRD data.

Sample	Crystallite Size(D) (nm)		Dislocation density(δ)	Stacking Fault (SF)	Micro Strain (ϵ)	Lattice strain (W-H plot)
	Scherrer's Equation	W-H plot				
Pure SnO ₂	32.83	21.40	$9.2 \times 10^{+18}$	0.006309	0.006450	-0.00319
Pure ZnO	35.99	23.85	$7.9 \times 10^{+18}$	0.005423	0.005540	-0.00509
Composite SnO ₂ /ZnO	34.47	21.63	$9.1 \times 10^{+18}$	0.005517	0.004830	-0.00523

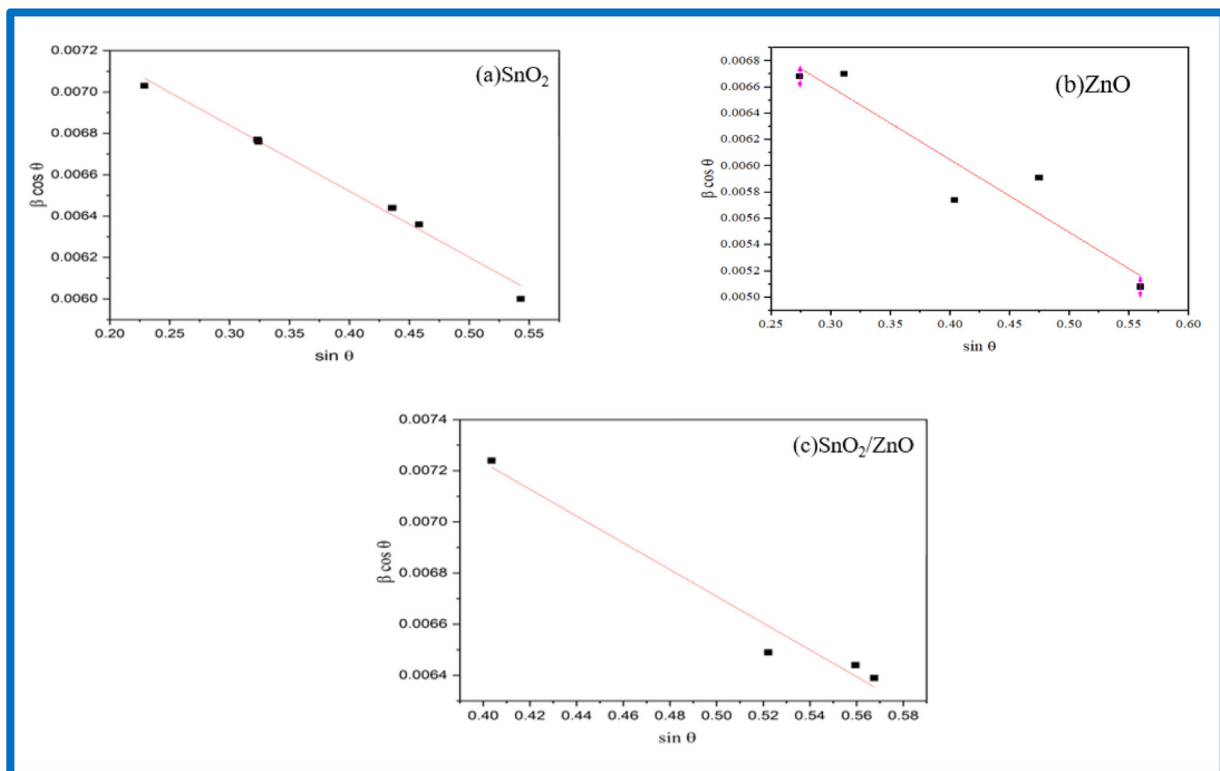


Fig. 3. W-H analysis of synthesized (a) SnO₂, (b) ZnO and (c) SnO₂/ZnO composites.

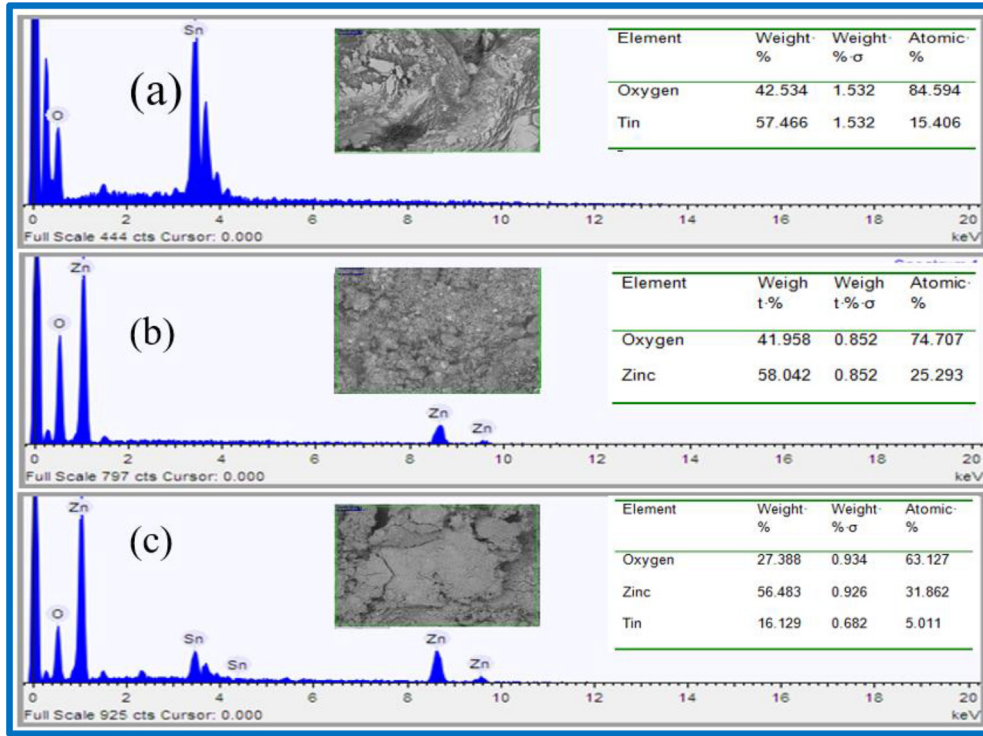


Fig. 4. EDS analysis of a) pure SnO₂ nanorods b) pure ZnO nanowires C) SnO₂/ZnO.

using the W-H method has smaller values as compared to the values calculated using Scherrer formula. It might be because of the Scherrer equation only considers the influence of crystallite size on XRD peak broadening and excludes microstructures of the lattice, such as intrinsic strain, which develops in nanocrystals through point defects, grain boundaries, triple junctions, and stacking faults. However, the variation in crystallite size of the synthesized nanostructures are similarly obtained by both the methods. Also, the other structural parameters of the synthesized nanostructures were calculated using equations 2–4 and tabulated in Table 1. The calculated values of Stacking Fault (SF) and macrostrain (ϵ) are comparable to the reported data [12]. Smaller values of SF and macrostrain further strengthen that synthesized nanostructures have proper periodic arrangement of lattice planes and almost negligible lattice defects.

$$D = \frac{K\lambda}{\beta \cos\theta} \quad (1)$$

$$\delta = \frac{1}{D^2} \quad (2)$$

$$SF = \left[\frac{2\pi^2}{45\sqrt{\tan\theta}} \right] \beta \quad (3)$$

$$\epsilon = \frac{\beta}{4\tan\theta} \quad (4)$$

Where λ is wavelength of radiation (1.5406 Å), 'k' is constant (0.94), 'D' is crystallite size, θ is Bragg's angle and β is the full width at half maximum (FWHM) in radians. In accordance to W-H method the overall broadening (FWHM) of the XRD peaks occurs due to crystallite size (D) and lattice strain (ϵ) of the nanostructures (eq. (5)). Uniform deformation model (UDM) has been considered to obtain the lattice strain and crystallite size. This model assumes the induced lattice strain due to lattice defects and imperfections in the nanomaterials is isotropic in nature.

$$\beta_{strain} = 4\epsilon \cdot \tan\theta \quad (5)$$

$$\beta_{hkl} = \frac{k\lambda}{D} \frac{1}{\cos\theta} + 4\epsilon \tan\theta \quad (6)$$

$$\beta_{hkl} \cos\theta = \frac{k\lambda}{D} + 4\epsilon \sin\theta \quad (7)$$

The y-intercept and slope of the linear fitted line plot of eq. (7) for various crystallographic planes gives the values of crystallite size and lattice strain, respectively. The W-H plot of the synthesized samples are shown in Fig. 3[12].

3.2. EDS analysis

The Energy Dispersive X-ray spectroscopy (EDS) analysis of pure SnO₂, pure ZnO and SnO₂/ZnO composite is shown in Fig. 4 (a-c). The EDS spectrum confirms the chemical composition of the synthesized nanostructures and absence of any additional major peaks in spectra suggest that synthesized nanostructures have no impurity [12].

3.3. the UV-Visible spectroscopy:

2.4. The absorbance spectra of SnO₂, ZnO, and SnO₂/ZnO composites nanostructure was recorded in the range of 200–800 nm using UV-Visible spectrophotometer at 300 K temperature and shown in Fig. 5 (a-c). UV-Visible spectrum shows the information about the excitonic and inter transition of nanostructures. The absorption coefficient (α) has been calculated using Beer-Lambert's law by equation (8).

$$\alpha(\lambda) = \frac{\ln\left(\frac{I}{I_0}\right)}{x} = \frac{\ln T}{x} = \frac{2.303A}{x} \quad (8)$$

Where I and I₀ are instantaneous and initial value photon intensity, respectively. The thickness of the cuvette is represented

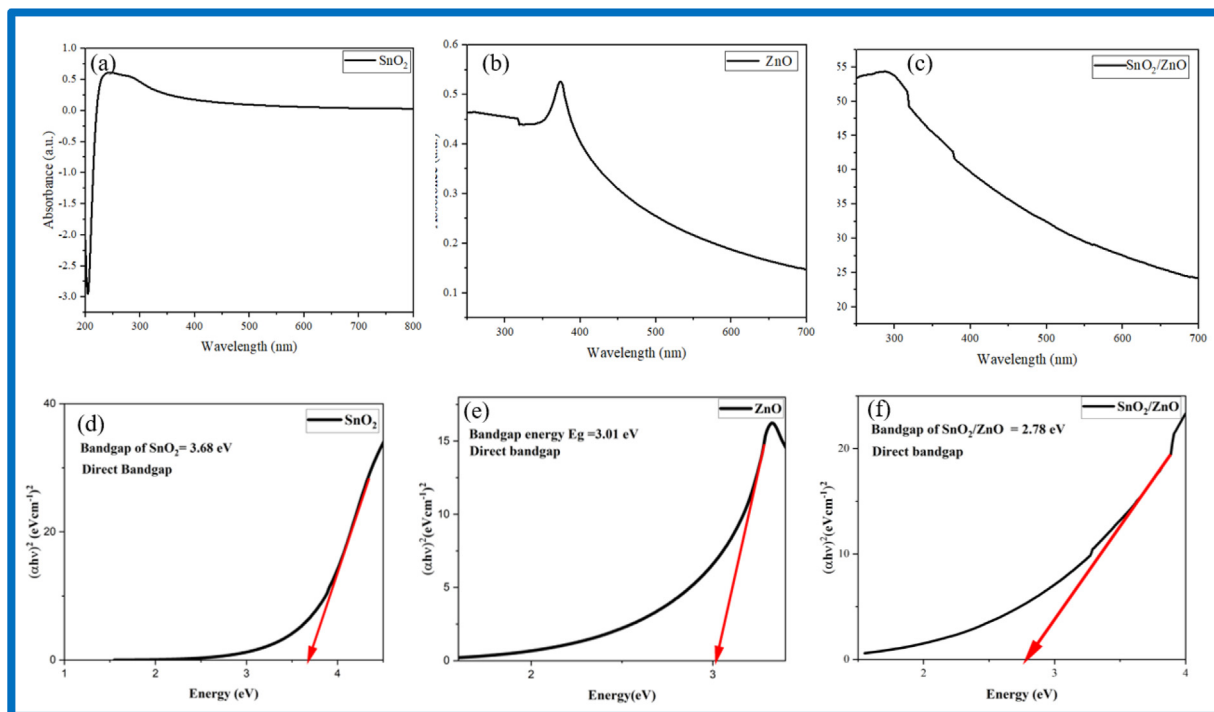


Fig. 5. UV-Visible absorption spectrum of pure SnO₂, ZnO and SnO₂/ZnO and their Tauc plot.

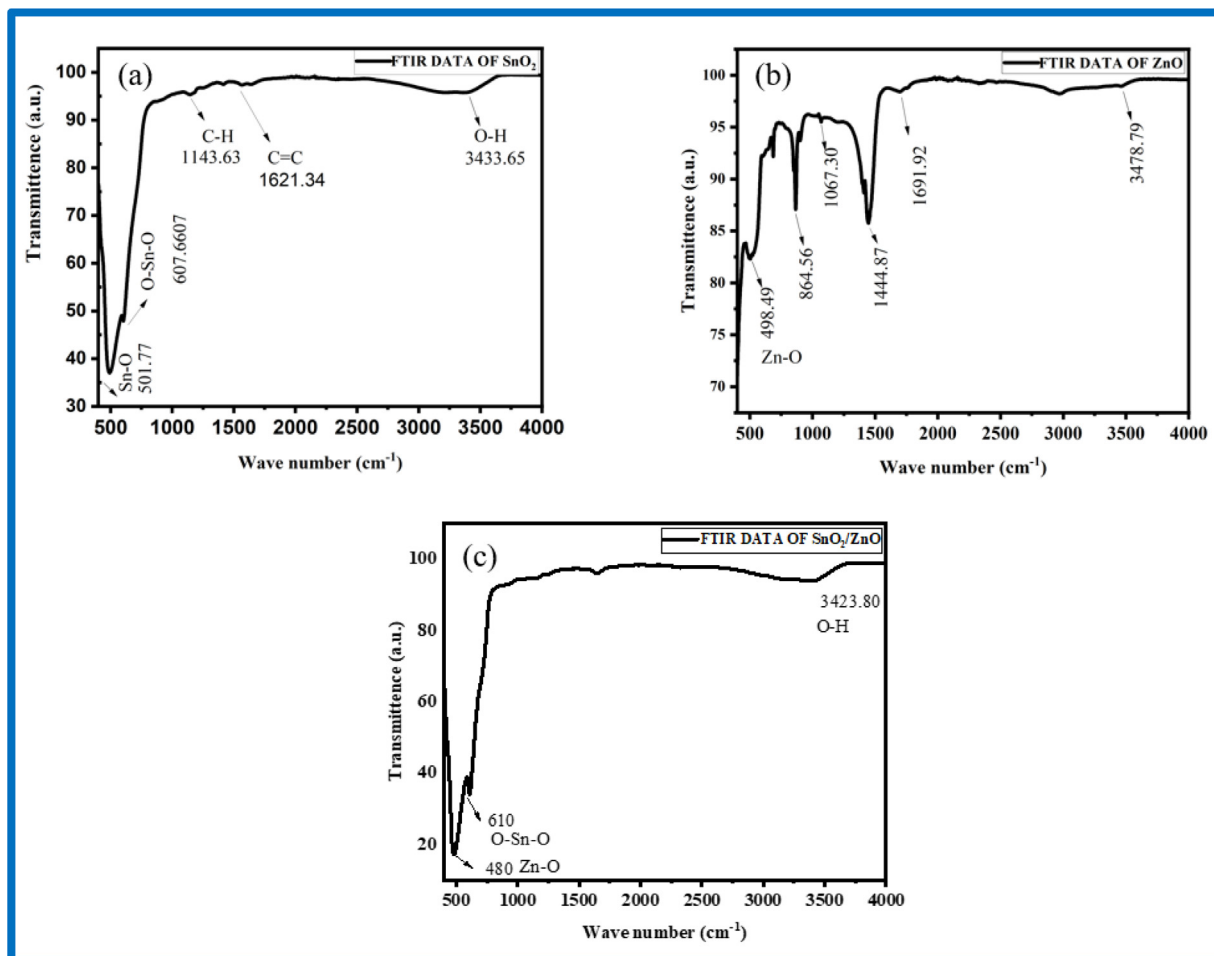


Fig. 6. FTIR graphs of pure SnO₂, ZnO and SnO₂/ZnO composite.

by \times and A represents the absorbance, which is a unit less quantity. To obtain the optical band gap of synthesized nanostructures we have plotted the Tauc plot between $(\alpha hv)^2$ versus photon energy (hv) as shown in Fig. 5 (a-c) by using equation (9).

$$(\alpha hv)^2 = A(hv - E_g)^n \quad (9)$$

where A is a constant (energy independent), α is the absorption coefficient. E_g represents optical band gap while n is another constant which depends on the type of transition.

To calculate band gap from the Tauc plot an intercept on \times axis (energy axis) obtained by extrapolating the linear portion of the Tauc plot. The obtained values for direct band gap of pure SnO_2 , ZnO and SnO_2/ZnO composite are 3.68 eV, 3.01 eV and 2.78 eV respectively. The obtained bandgap results are in accordance with the reported literature. SnO_2/ZnO composite have lower direct band gap than pure ZnO and SnO_2 . The reduction in bandgap of composite might be because of the generation of new energy levels of ZnO and SnO_2 on formation of composite nanostructure [8][12].

3.4. FTIR analysis

The FTIR spectra of using ATR (Attenuated total reflection) mode, for ZnO , SnO_2 , SnO_2/ZnO composites, the FTIR was recorded and various functional groups were obtained which can be confirmed using different absorption peaks as shown in the plots in

Fig. 6. Absorbance band around at 607.6 cm^{-1} could be for O-Sn-O, Sn-O lattice which is in accordance with the results obtained by. Further, the absorptions at 3433 cm^{-1} of infrared spectra assigned to hydroxyl groups of water molecules adsorbed on the outer surface of SnO_2 nanorods. Similarly, the absorption peak appearing at 3478 cm^{-1} in the IR spectrum of ZnO nanorods could be assigned to the O-H vibrations, which confirms the presence of hydroxyl (OH) groups on the external surface of ZnO nanorods. Additionally, the appearance of the absorbance band at 498.5 cm^{-1} confirms the formation of ZnO nanorods as reported in literature. The absorbance band at 3423 cm^{-1} appears in the IR spectrum of SnO_2/ZnO composite.

3.5. TEM analysis

The obtained Transmission Electron Microscopic (TEM) images of the synthesized samples are shown in Fig. 7 (a-i) at three different magnifications/scales. It can be observed that the morphology of the synthesized SnO_2 , and ZnO are nanobelts and nanorods, respectively. While the nanostructure of SnO_2/ZnO composites is a combination of nanobelts and nanorods as shown in Fig. 7 (g-i). Lattice resolved TEM fringes of surface of the SnO_2 , ZnO and SnO_2/ZnO composites are represented in Fig. 8 along with their SAED pattern. The spacing between two neighbouring fringes has been calculated from TEM image (a-c). The obtained average spac-

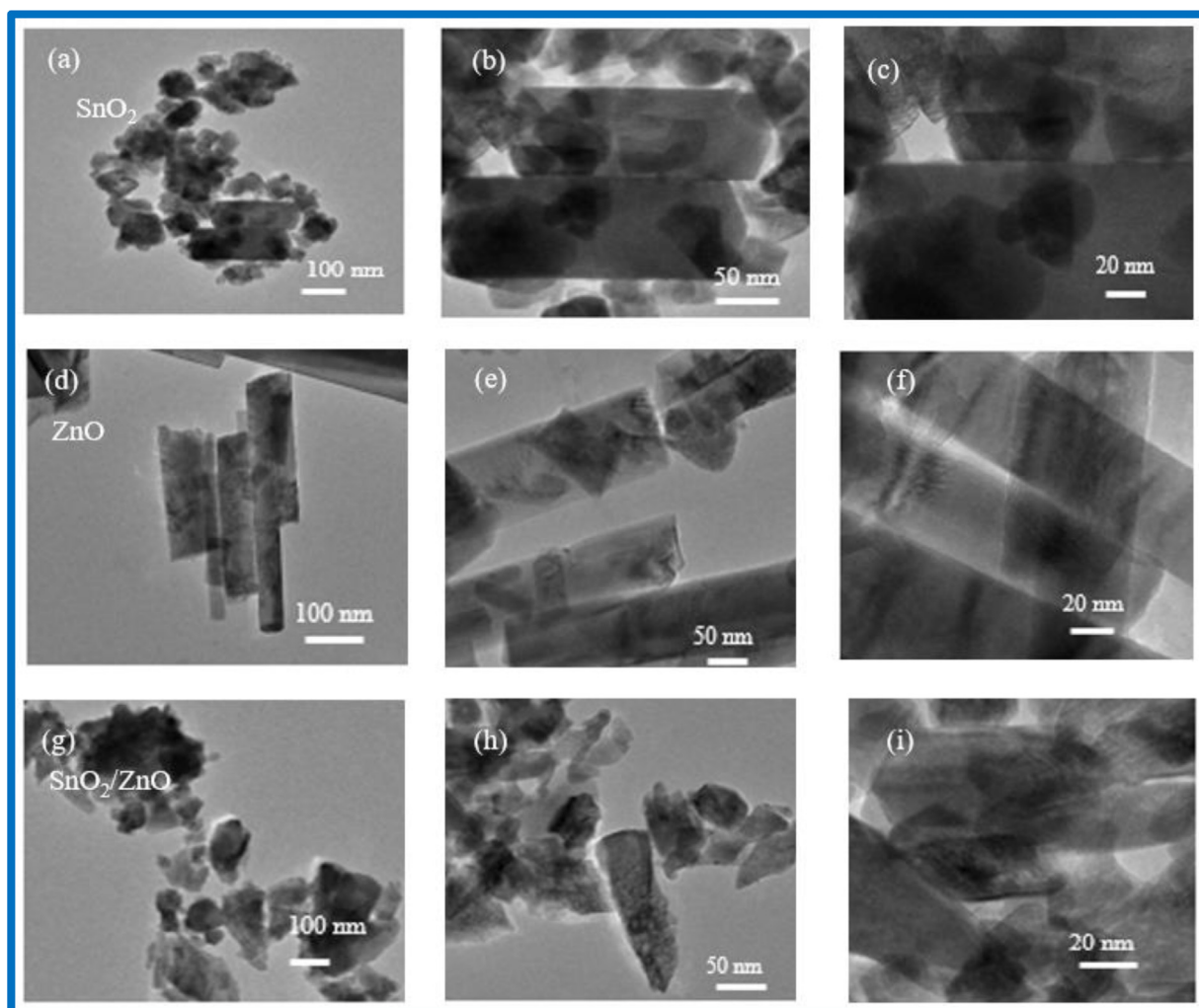


Fig. 7. A typical magnification TEM image of (a-c) SnO_2 (d-f) ZnO (g-i) SnO_2/ZnO .

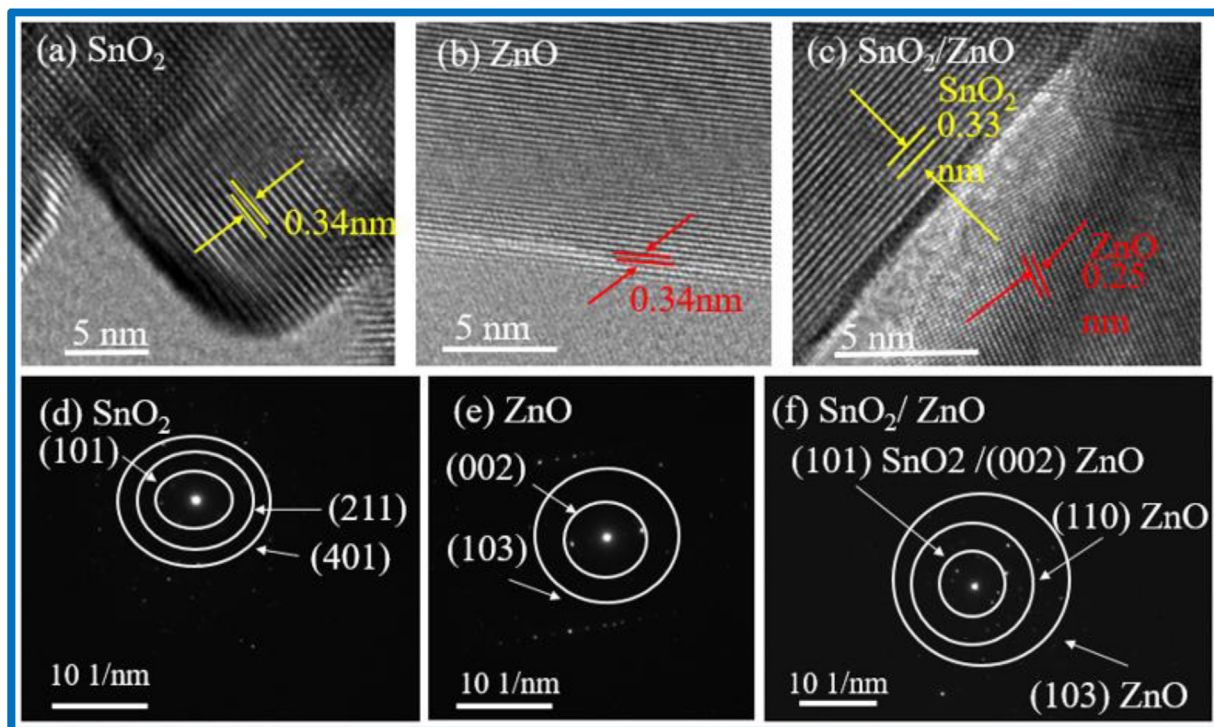


Fig. 8. (a-c) SAED pattern (d-f) Lattice-resolved TEM micrograph of the surface of the SnO₂, ZnO and SnO₂/ZnO composite.

ing for SnO₂ and ZnO fringes are 0.33 nm and 0.25 nm which correspond to (110) of SnO₂ and ZnO (002) crystal planes. The lattice fringes of SnO₂/ZnO composite with average interplanar spacing of 0.33 nm and 0.25 nm are also matched with (110) and (002) planes of SnO₂ and ZnO, respectively. This further confirms that the SnO₂/ZnO composite is a two-phase composite material composed of SnO₂ and ZnO.

The SAED, Fig. 8 (d-f), clearly shows the continuous ring diffraction pattern due to the SnO₂ containing a large number of small randomly distributed crystallites for different planes like (101), (211) and (401). ZnO have SAED pattern showing planes (002) and (103). In composite (002) and (101) planes of SnO₂ and ZnO are overlapping with each other while (110) and (103) planes of ZnO are clearly visible [1].

4. Conclusion

Pure SnO₂, ZnO and SnO₂/ZnO composite nanostructures were prepared by using hydrothermal methods. The various structural parameters of synthesized samples were obtained by using Scherrer and W-H Methods. The results indicate that the composites structure has very few additional defects and strain as compared to pure SnO₂ and ZnO. While the observed optical band gap of SnO₂/ZnO composite is smaller than pure SnO₂, and ZnO. Which might be due to generation of new energy levels in the composite structure. The FTIR spectrum further confirms the synthesis of pure SnO₂, pure ZnO and SnO₂/ZnO composites. This study suggests that on synthesis of SnO₂/ZnO composite the structural properties such as phase, lattice strain, defects did not change significantly while the UV-Visible and FTIR absorption peaks get shifted as compared to pure SnO₂ and ZnO. These results are found to be interesting for the application of SnO₂/ZnO composite in the optoelectronic and gas sensing devices.

CRediT authorship contribution statement

Deshraj Meena: Supervision. **Vineeta Yadav:** Conceptualization. **Nirmal Singh:** Conceptualization. **Shilpa Rana:** . **K.M. Komal:** .

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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